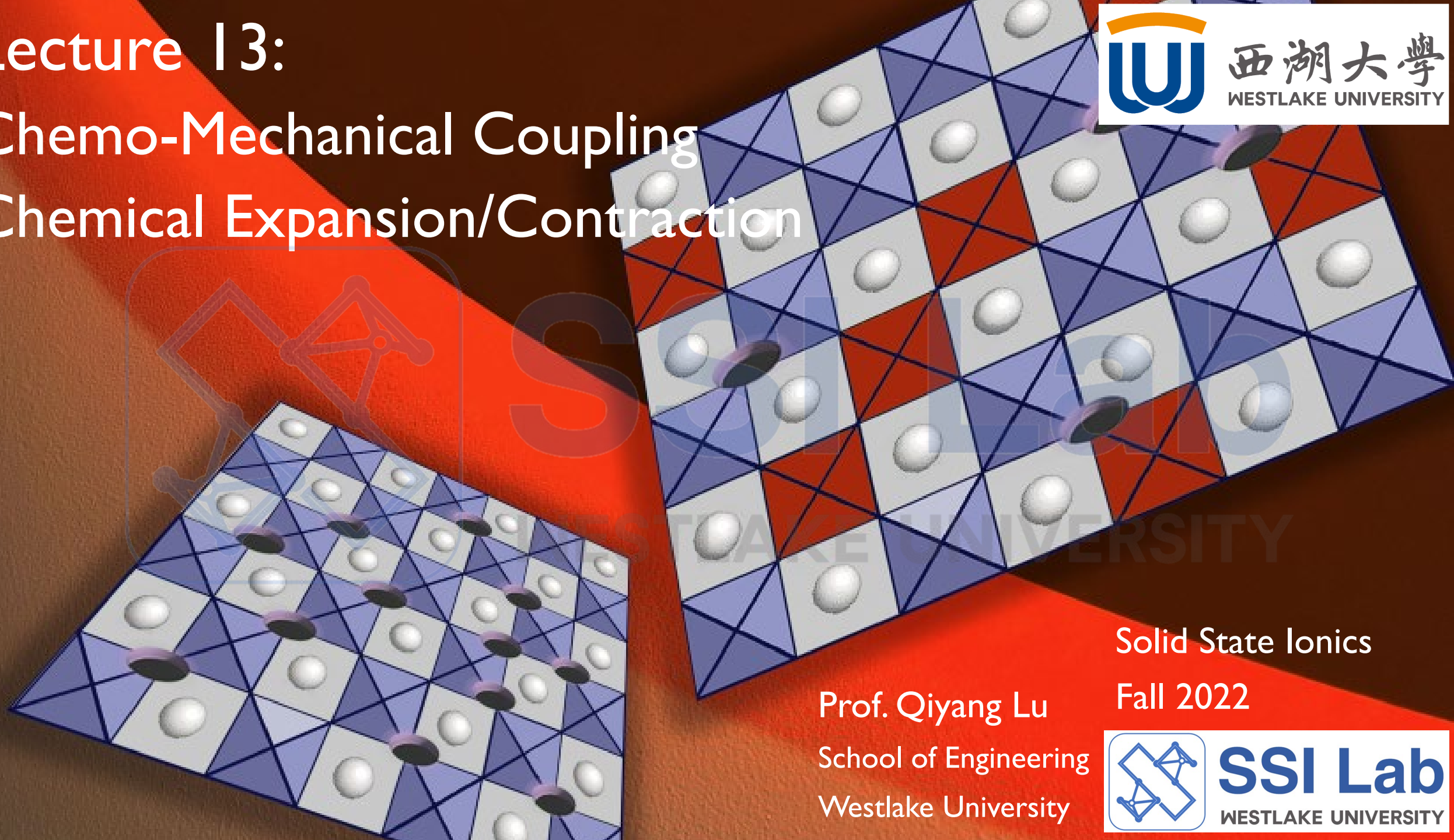


# Lecture 13:

## Chemo-Mechanical Coupling

### Chemical Expansion/Contraction



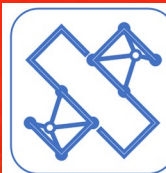
Solid State Ionics

Fall 2022

Prof. Qiyang Lu

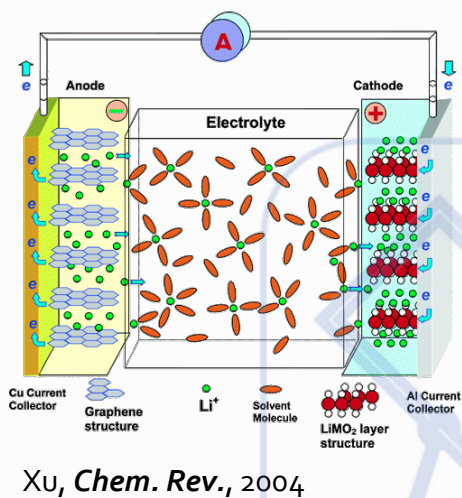
School of Engineering

Westlake University

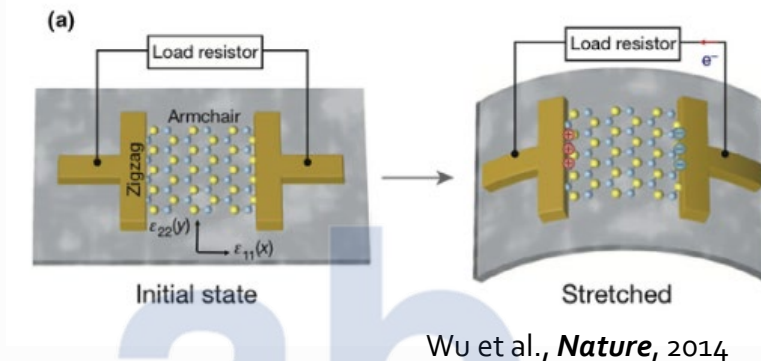


**SSI Lab**  
WESTLAKE UNIVERSITY

# Electro-chemo-mechanical coupling



Electrochemistry  
(e.g., Batteries)

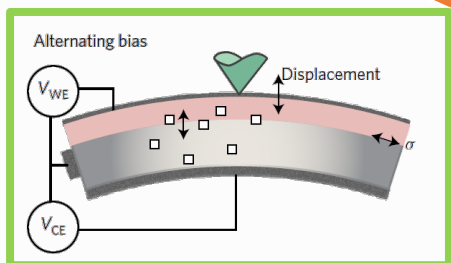


Electro-mechanics  
(e.g., Piezoelectric)

Chemical

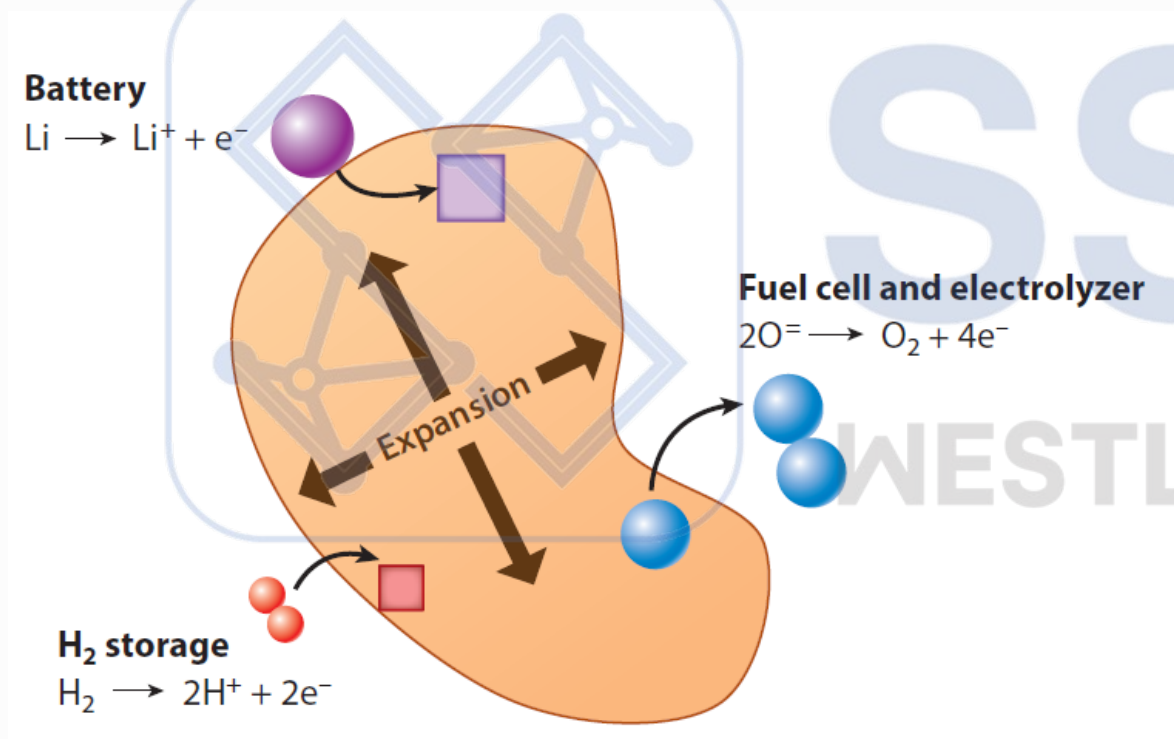
Chemo-mechanics  
(e.g., Actuators)

Mechanical

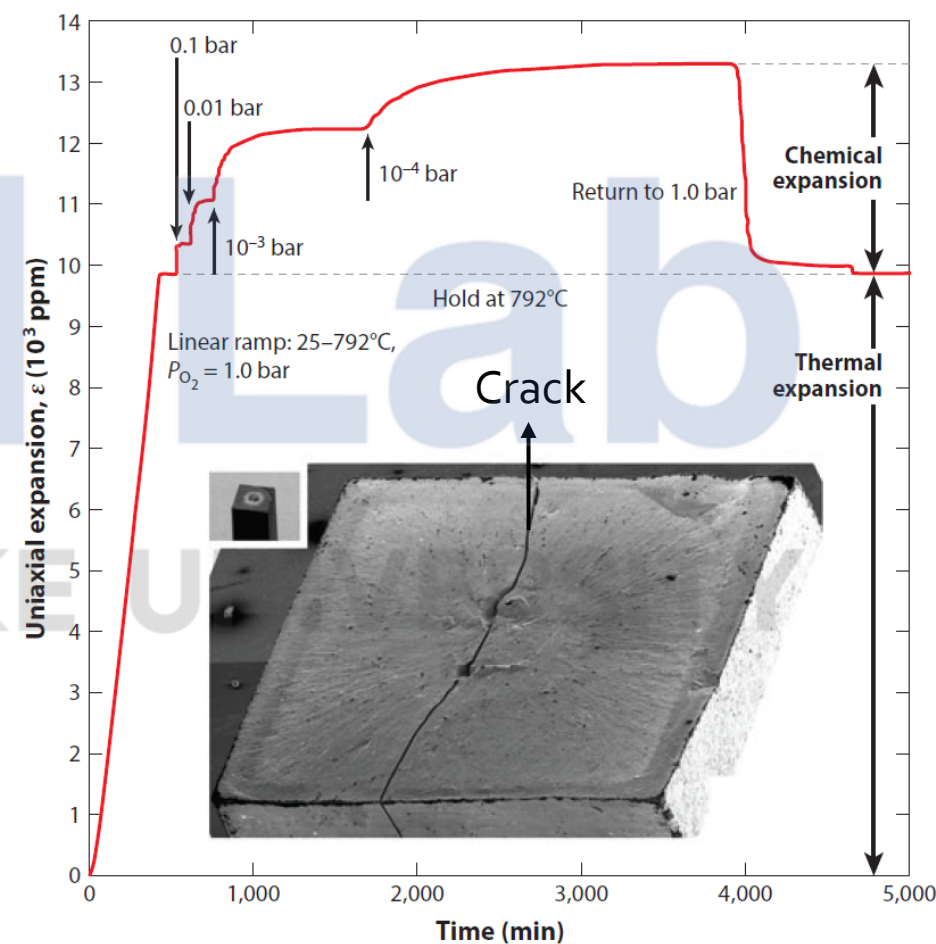


# Chemical expansion: definition and consequence

Chemical  
(i.e., point defects)  $\longleftrightarrow$  Mechanical  
(i.e., strain, lattice constant)

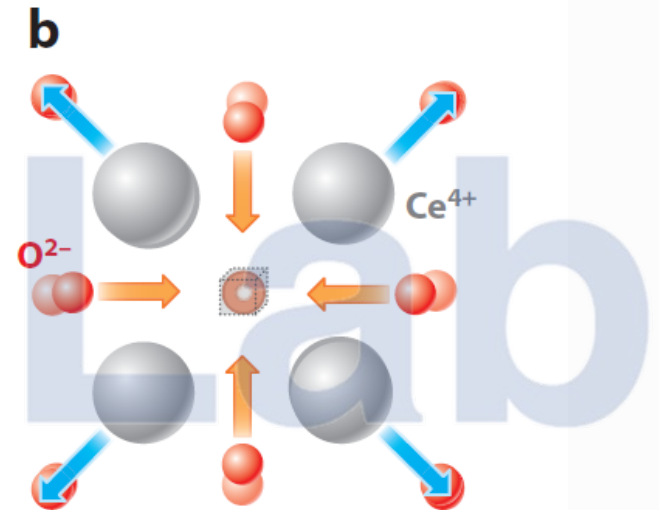
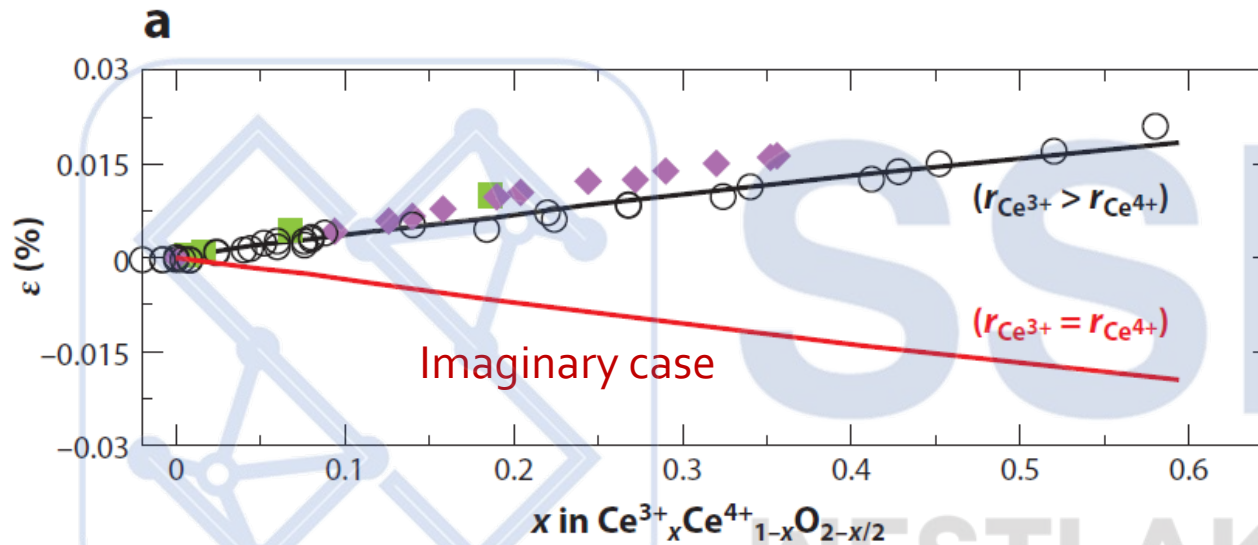


Bishop et al., *Annu. Rev. Mater. Res.*, 2014

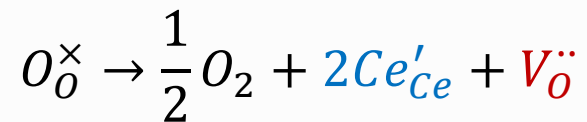


Adler, *J. Am. Ceram. Soc.*, 2001

# CeO<sub>2-δ</sub> as a model system to study chemical expansion



Defect chemical reaction in  $\text{CeO}_{2-x}$



Different effect on lattice constant:

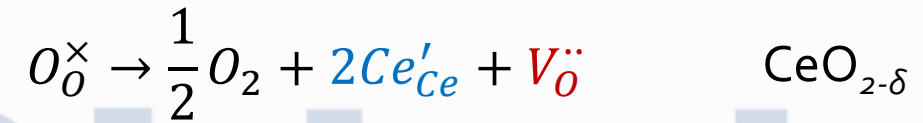
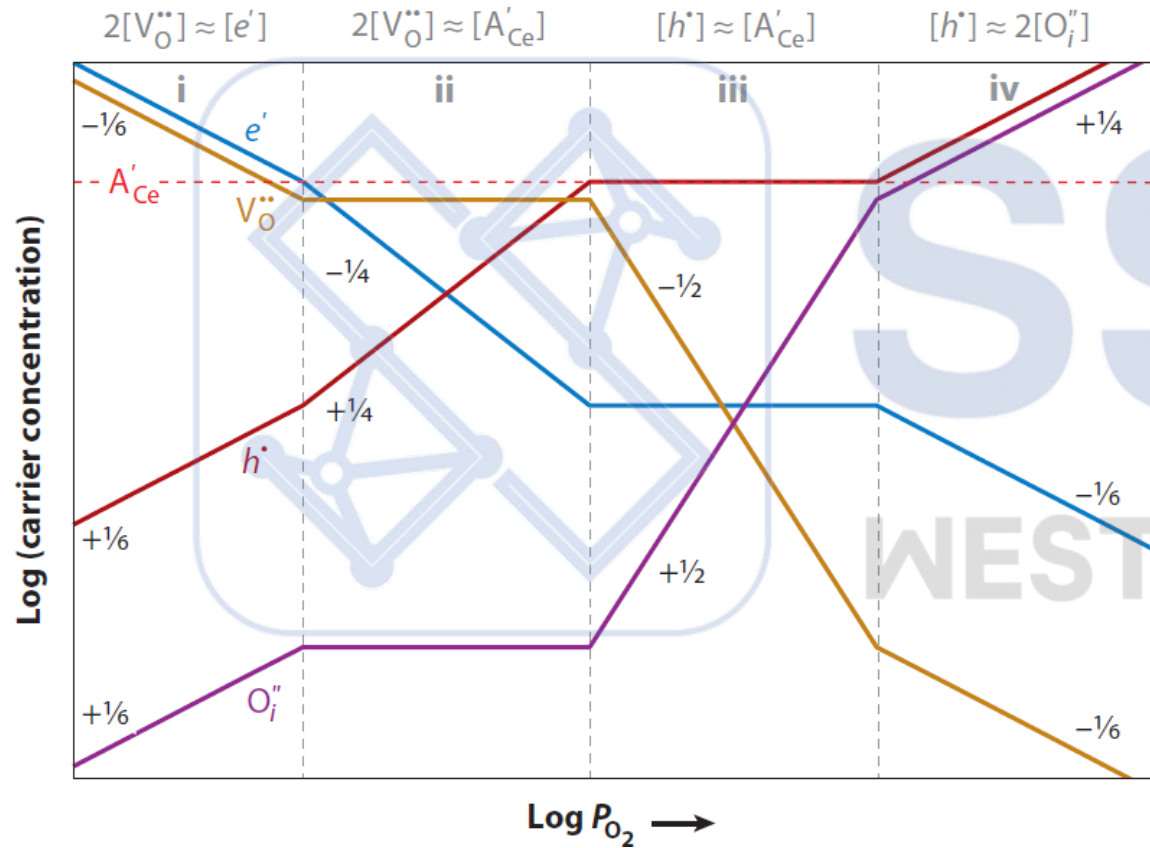
$\text{Ce}'_{\text{Ce}}$  expansion (ionic radius  $\text{Ce}^{3+} > \text{Ce}^{4+}$ )

$\text{V}_\text{O}^{\bullet\bullet}$  contraction (vacancies = voids)

Overall, increase of  $x$  in  $\text{CeO}_{2-x}$  *increases* lattice constant



# How to give a quantitative description on chemical expansion?



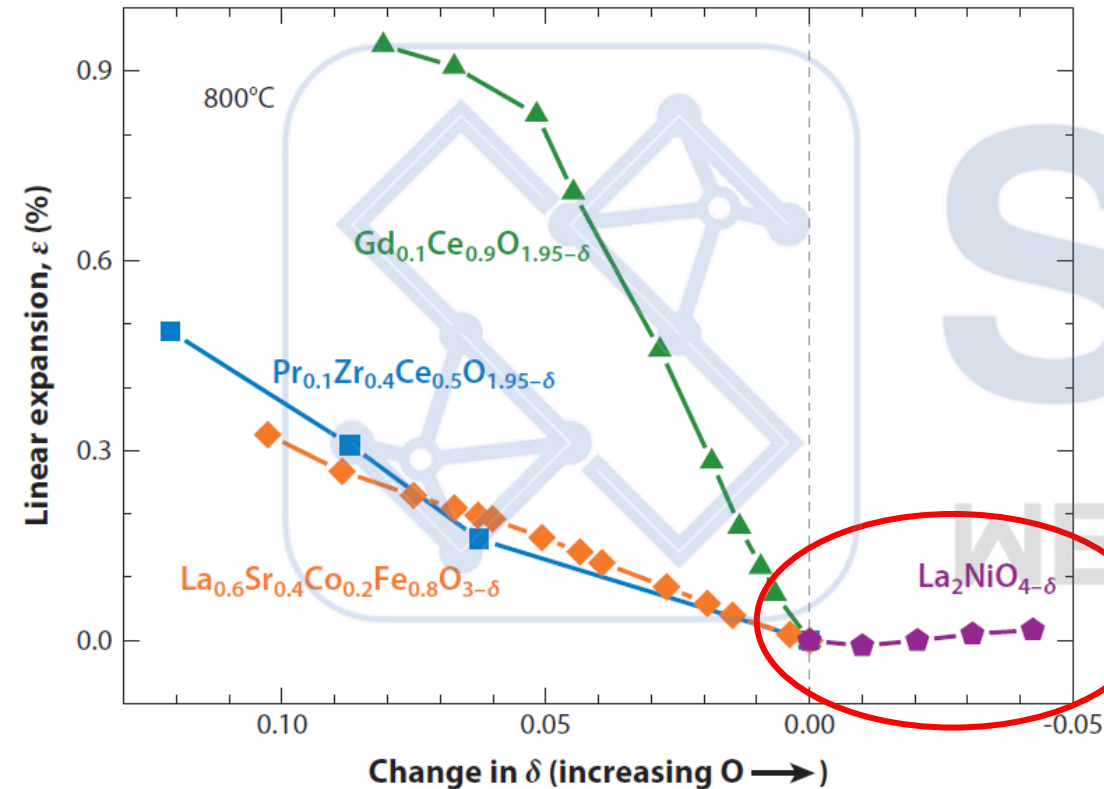
$$\varepsilon = \alpha_C [i]$$

Strain  
def:  $\Delta L/L$

Chemical  
expansion  
coefficient

Concentration  
of defects  
Unit:  $\#/cm^3$

Note: concentration of defects expressed by using  $\#/cm^3$  is different from  $\delta$



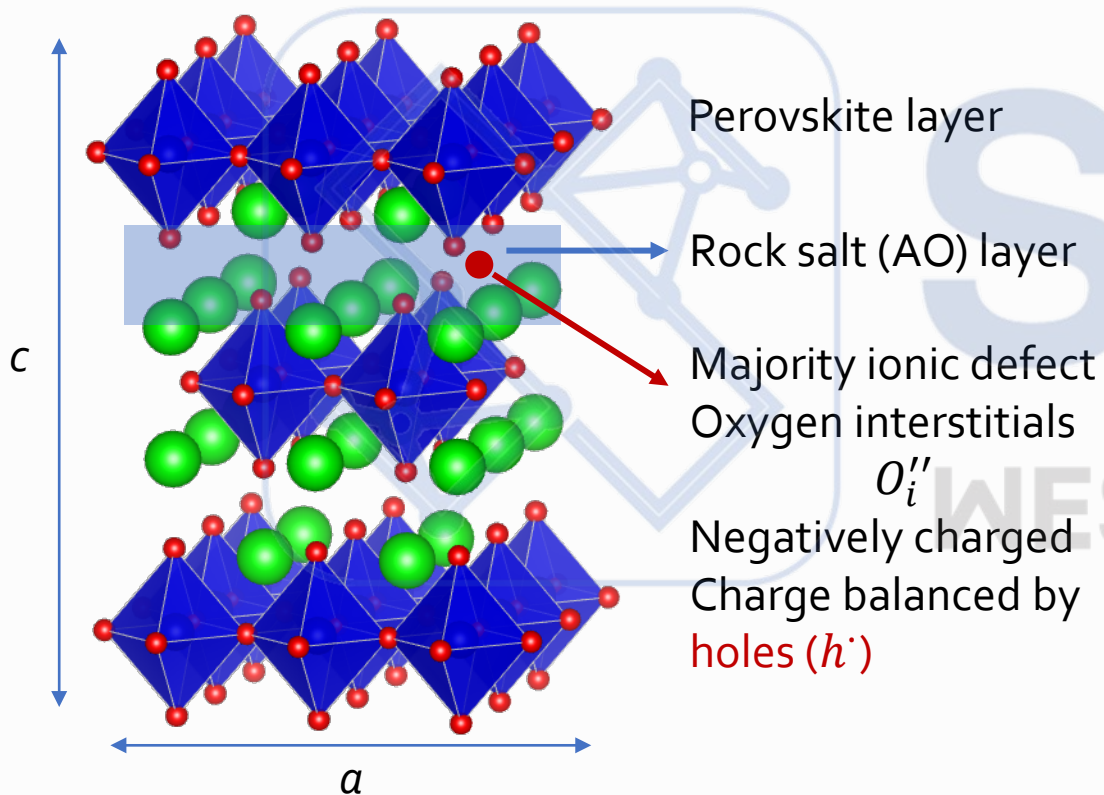
Note: concentration of defects expressed by using  $\#/\text{cm}^3$  is different from  $\delta$

Table 1 Experimentally determined stoichiometric expansion coefficients and corresponding oxygen vacancy radii for different materials together with the DFT values obtained in this work. The chemical expansion coefficients and  $r_V$  for A-site and B-site doped  $\text{LaGaO}_3$  were obtained from averaging several experimental values (which we report in the ESI) taken from ref. 11 and 50

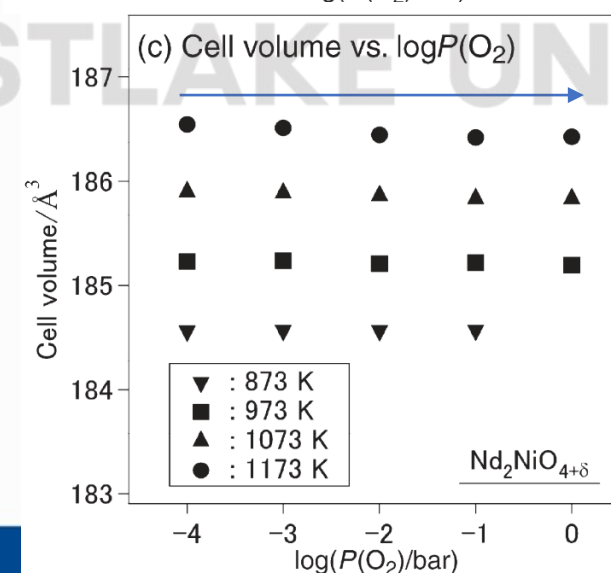
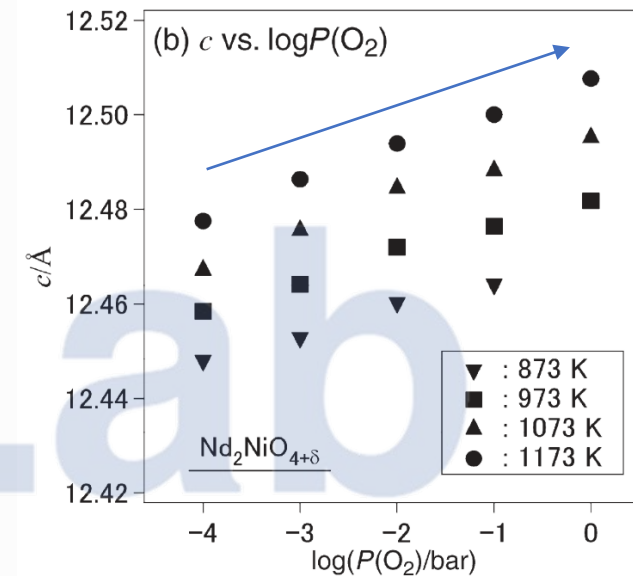
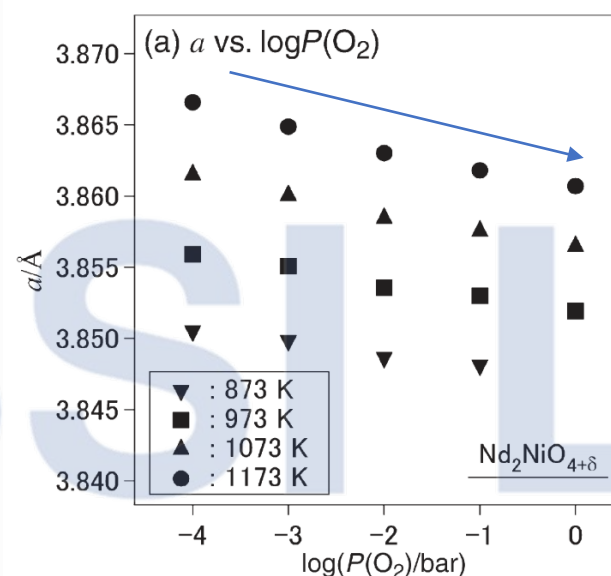
Material	$\alpha_S$	$\alpha_S^{\text{DFT}} [\text{cm}^3 \times 10^{24}]$	$r_V [\text{\AA}]$	Ref.
<b>Fixed valence substitution</b>				
$\text{La}(\text{Ga}, \text{Mg})\text{O}_3$ (Sr A-site substitution)	0.008	0.48	1.30	11 and 50
$\text{La}_{1-x}\text{Sr}_x\text{Ga}_{0.8}\text{Mg}_{0.2}\text{O}_{3-x/2}$ (DFT)	$0.026 \pm 0.004$	1.6	1.41	This work
$(\text{La}, \text{Sr})\text{GaO}_3$ (Mg B-site substitution)	0.056	3.3	1.29	11 and 50
$\text{La}_{0.8}\text{Ga}_{1-x}\text{Mg}_x\text{O}_{3-x/2}$ (DFT)	$0.057 \pm 0.008$	3.4	1.30	This work
$\text{La}_{1-x}\text{Sr}_x\text{AlO}_3$	0.017	0.93	1.36	53
$\text{La}_{1-x}\text{Sr}_x\text{AlO}_3$ (DFT)	$0.018 \pm 0.014$	0.98	1.38	This work
$\text{SrTi}_{1-x}\text{Ga}_x\text{O}_3$ (DFT)	$0.040 \pm 0.021$	2.4	1.57	This work
<b>Multivalent cation</b>				
$\text{CeO}_{2-\delta}$ <sup>a</sup>	0.10	4.0	1.169	20
$\text{Pr}_{0.1}\text{Zr}_{0.4}\text{Ce}_{0.5}\text{O}_{1.95-\delta}$ <sup>a</sup>	0.046	1.7	0.966	19
$\text{La}_{0.9}\text{Sr}_{0.1}\text{Ga}_{0.95}\text{Ni}_{0.05}\text{O}_{3-\delta}$ (800–900 °C, oxidizing conditions)	0.040–0.049	2.4–2.9	1.24–1.30	32
$\text{SrTi}_{0.65}\text{Fe}_{0.35}\text{O}_{3-\delta}$ (700–1000 °C, oxidizing conditions)	0.040–0.049	2.38–2.92	1.38–1.43	54

<sup>a</sup> These compounds have the fluorite structure.

Marrocchelli *et al.*, *PCCP*, 2015

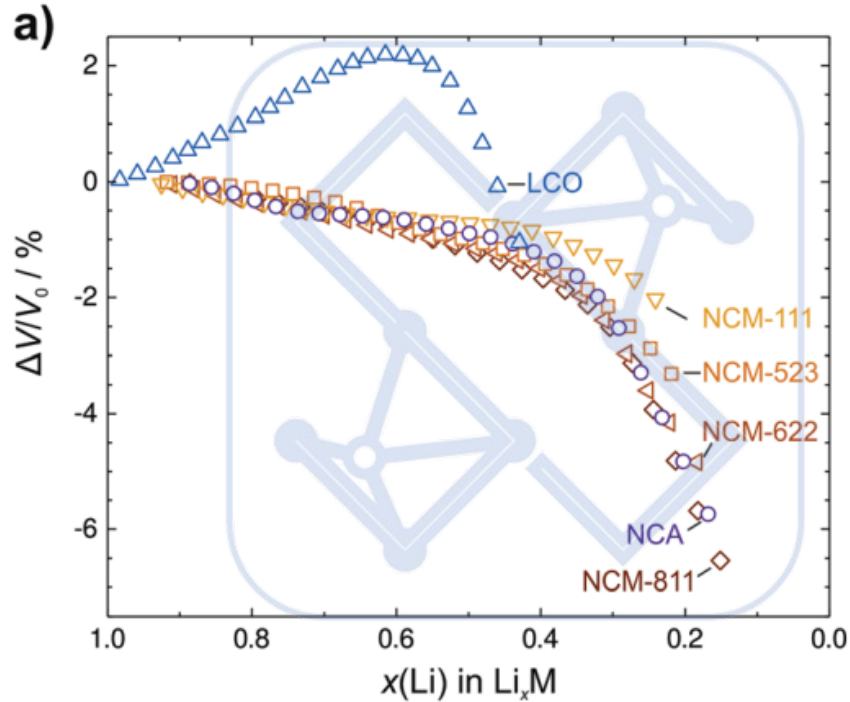


Ruddlesden-Popper phase  
 $A_2BO_4$  e.g.,  $Nd_2NiO_4$

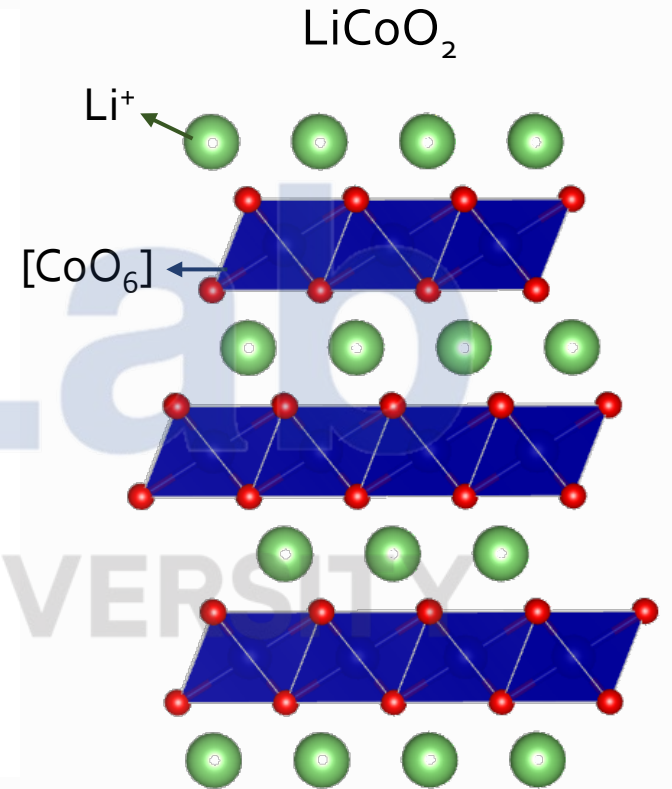
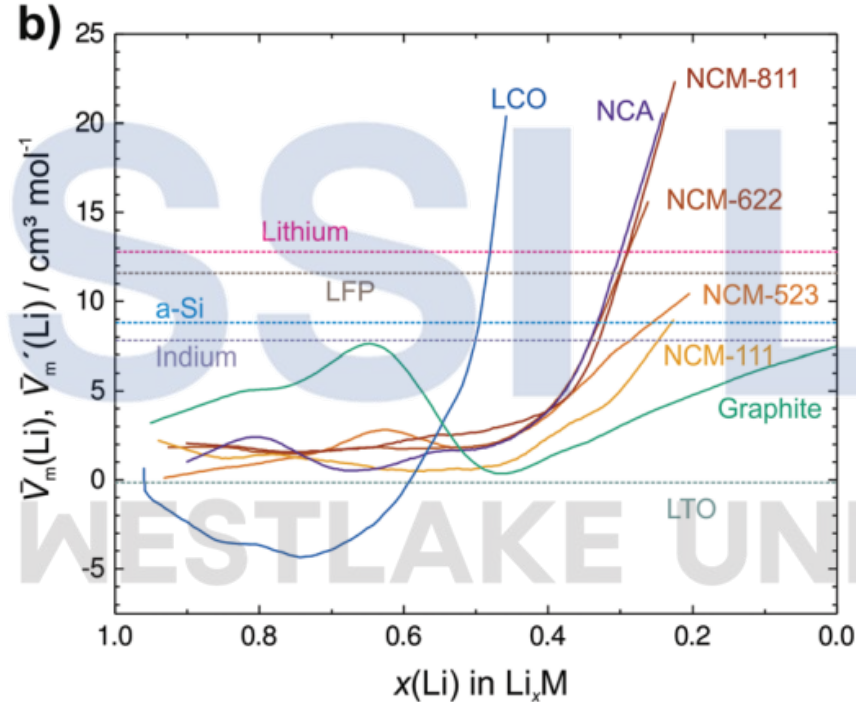


Oxygen interstitials affect  $a$  and  $c$  differently.

$$pO_2 \rightarrow \delta \rightarrow a, c \rightarrow V$$



LCO =  $\text{Li}_x\text{CoO}_2$   
 NCMxyz =  $\text{Li}_\delta(\text{Ni}_x\text{Co}_y\text{Mn}_z)\text{O}_2$   
 NCA =  $\text{Li}_\delta(\text{Ni}_x\text{Co}_y\text{Al}_z)\text{O}_2$

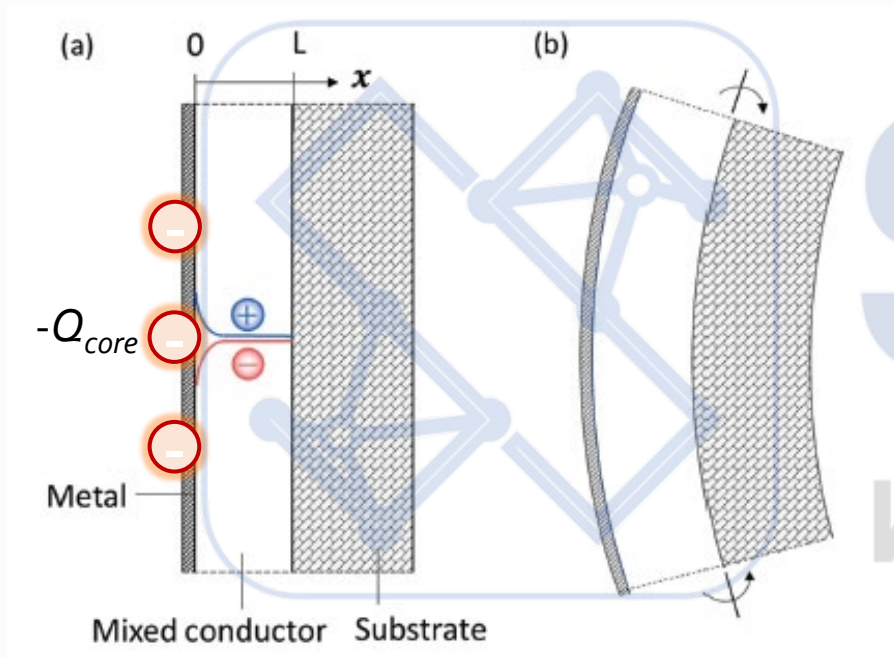


$$\bar{V}_m(\text{Li}) = \left( \frac{\partial V}{\partial n_{\text{Li}}} \right)_{T,p,n_i \neq n_{\text{Li}}} \quad \text{Partial molar volume of Li}$$



# How to model the equilibrium of charged ionic defects under strain/stress?

## Space charge layer + stress field



Equilibrium condition:  $\tilde{\mu}_i$  is uniform  $\tilde{\mu}_i = \mu_i + z_i F \phi$

$$\tilde{\mu}_i = \mu_i + z_i F \phi = \mu_i^0 + \underbrace{PV_{m,i}}_{\text{hydrostatic pressure}} + \underbrace{RT \ln a_i}_{\text{partial molar volume}} + z_i F \phi$$

In dilute limit:

$$\frac{c_i(x)}{c_{i,\text{ref}}} = \exp\left[-\frac{1}{RT} \left[ z_i F (\phi(x) - \phi_{\text{ref}}) + V_{m,i} (P(x) - P_{\text{ref}}) \right] \right]$$

Electro-chemical  
coupling

Chemo-mechanical  
coupling

# How to model the equilibrium of charged ionic defects under strain/stress?

In dilute limit:

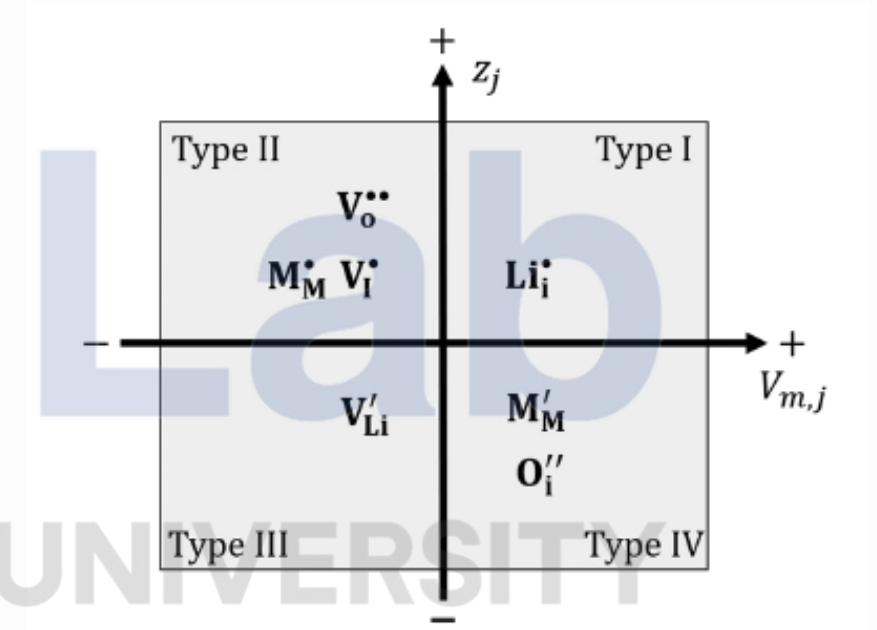
$$\frac{c_i(x)}{c_{i,\text{ref}}} = \exp\left[-\frac{1}{RT} \left[ z_i F (\phi(x) - \phi_{\text{ref}}) + V_{m,i} (P(x) - P_{\text{ref}}) \right] \right]$$

Electro-chemical  
coupling

Chemo-mechanical  
coupling

**Note:**

- In this picture, electrons and holes are treated as localized charge (polarons) → reduced cations (electrons) lead to expansion while oxidized cations (holes) lead to contraction;
- At equilibrium, the concentration of ionic defects are determined collectively by both electrochemical and chemomechanical coupling.



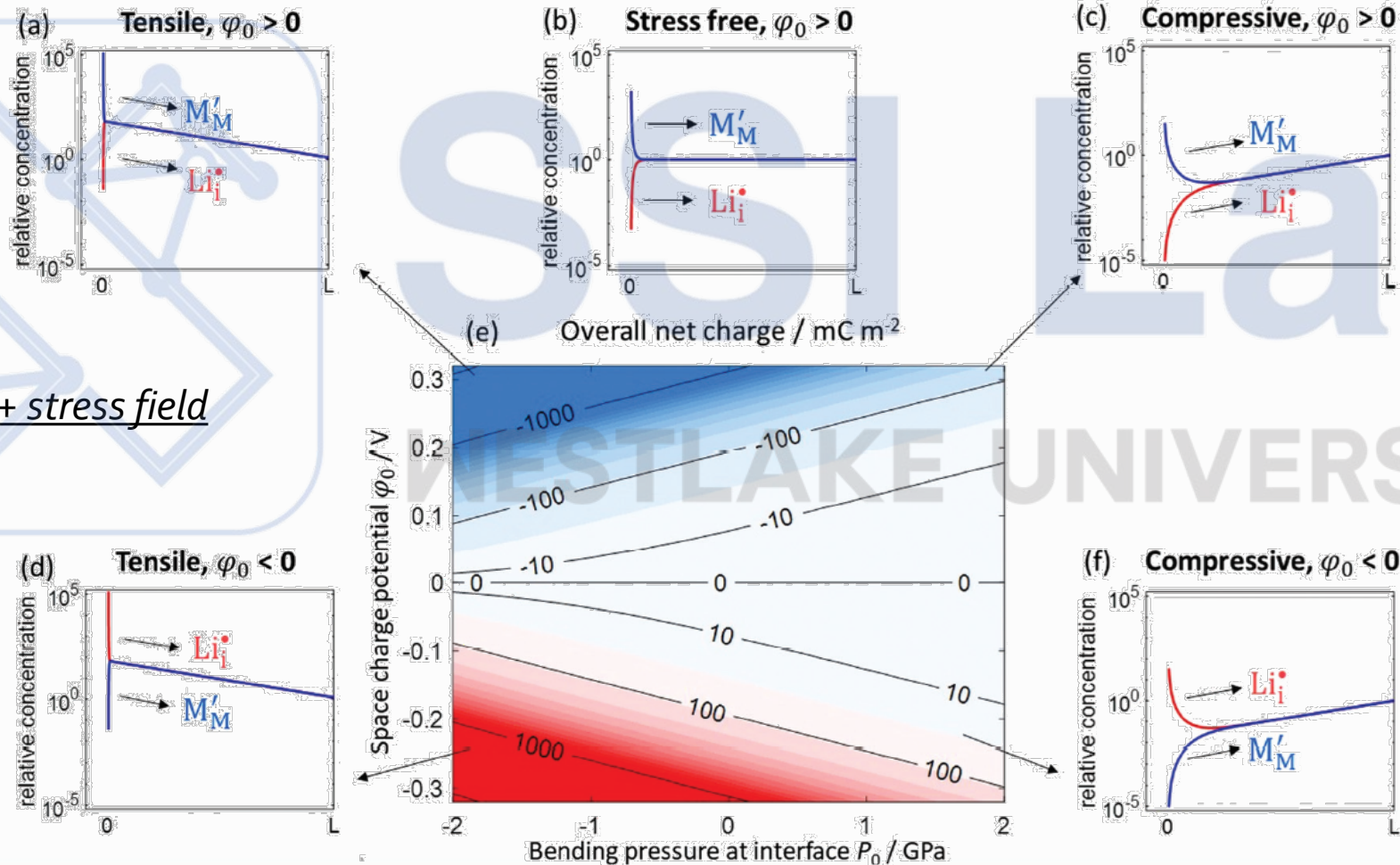
Vacancy:  $V_{m,i} < 0$

Interstitial:  $V_{m,i} > 0$

# How to model the equilibrium of charged ionic defects under strain/stress?

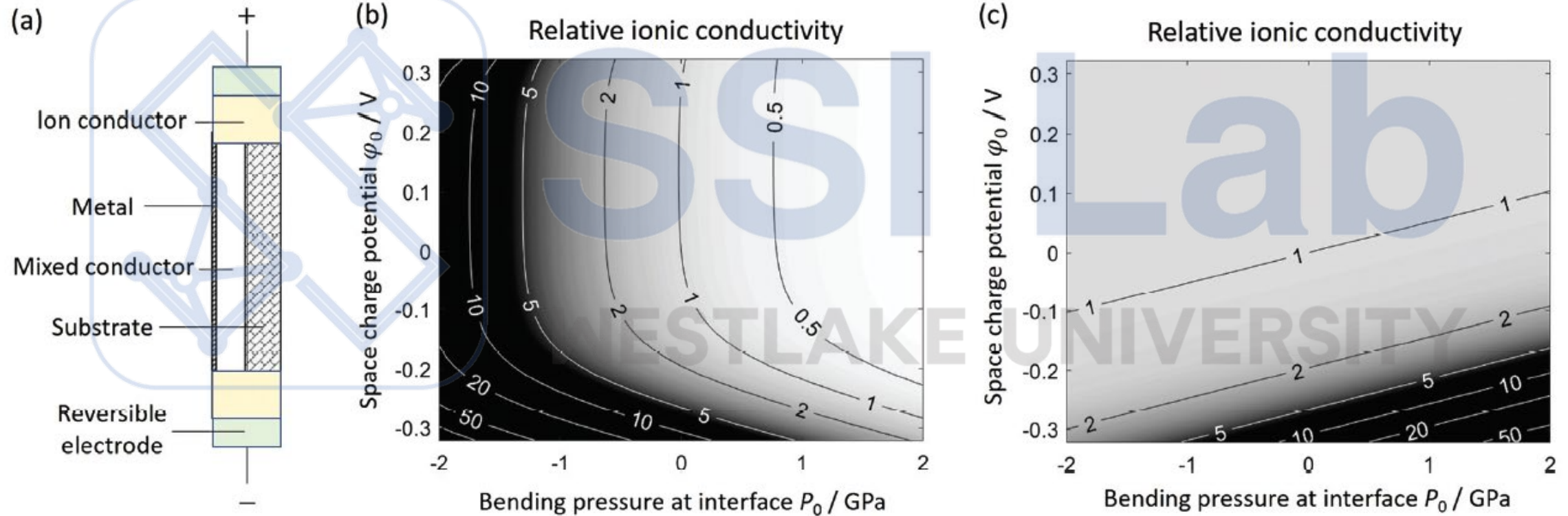
$$\frac{c_i(x)}{c_{i,\text{ref}}} = \exp\left[-\frac{1}{RT} \left[ z_i F (\phi(x) - \phi_{\text{ref}}) + V_{m,i} (P(x) - P_{\text{ref}}) \right] \right]$$

Space charge layer + stress field



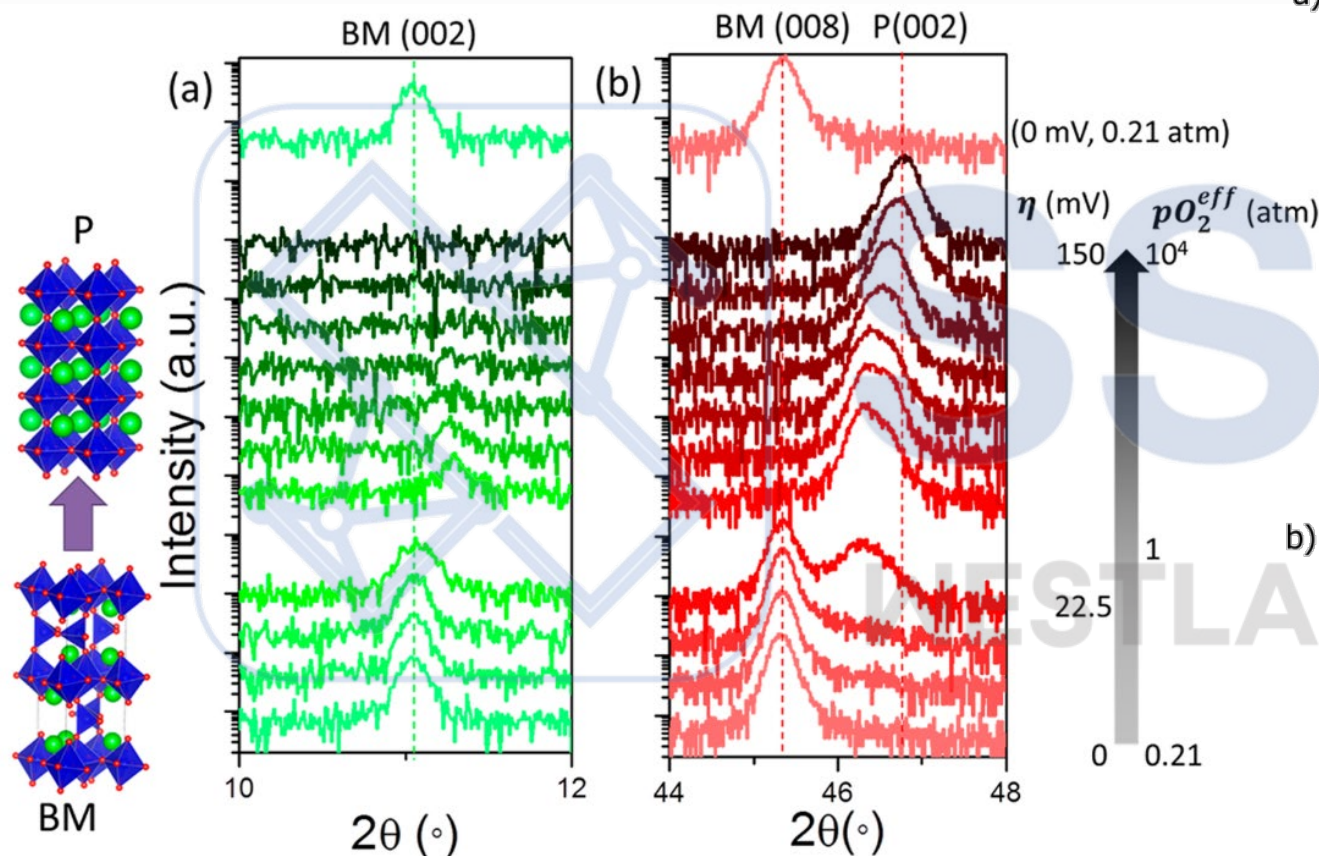
# How to model the equilibrium of charged ionic defects under strain/stress?

$$\frac{c_i(x)}{c_{i,\text{ref}}} = \exp\left[-\frac{1}{RT} \left[ z_i F (\phi(x) - \phi_{\text{ref}}) + V_{m,i} (P(x) - P_{\text{ref}}) \right] \right]$$

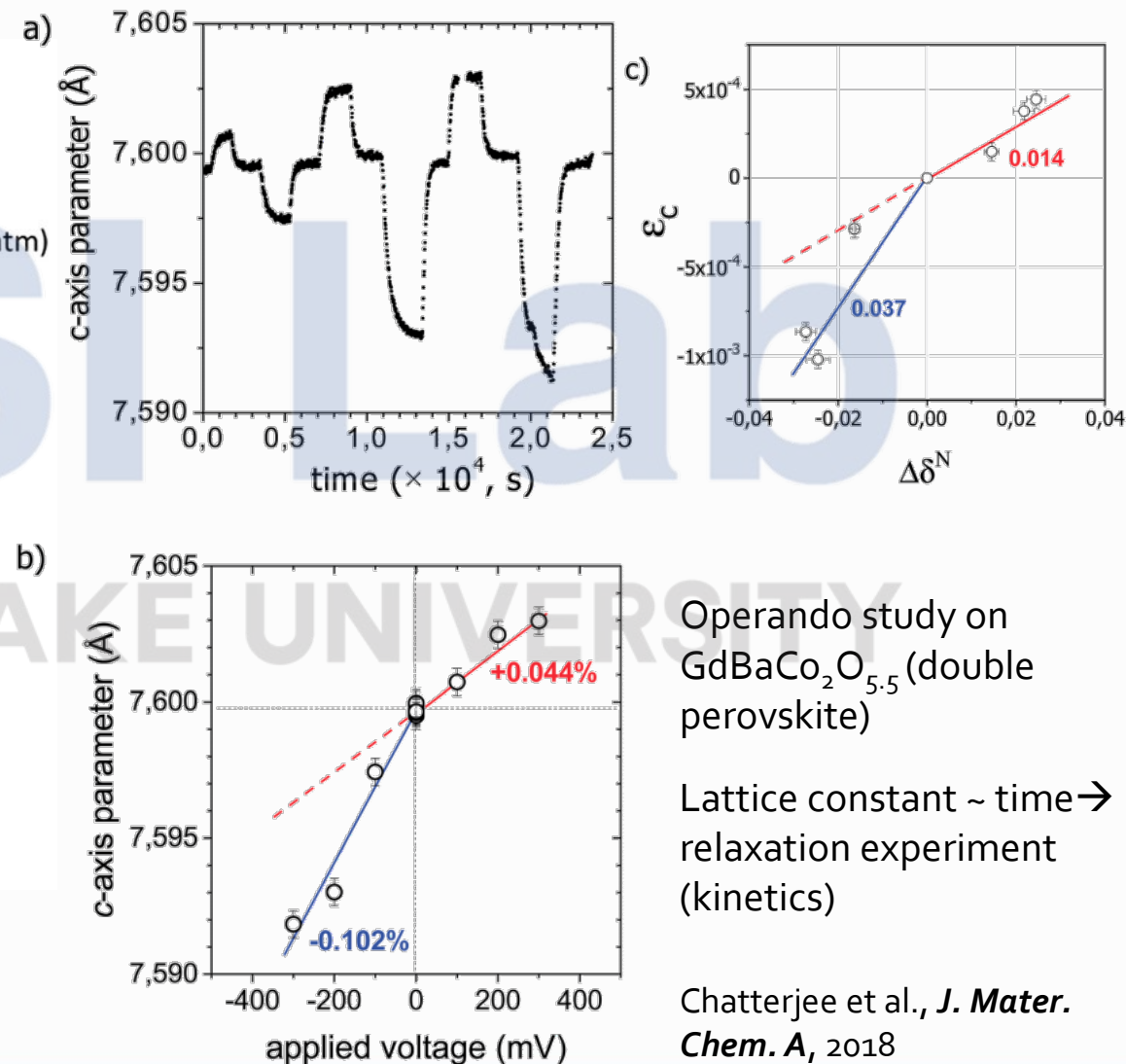




# How to probe chemical expansion: X-ray diffraction (XRD)

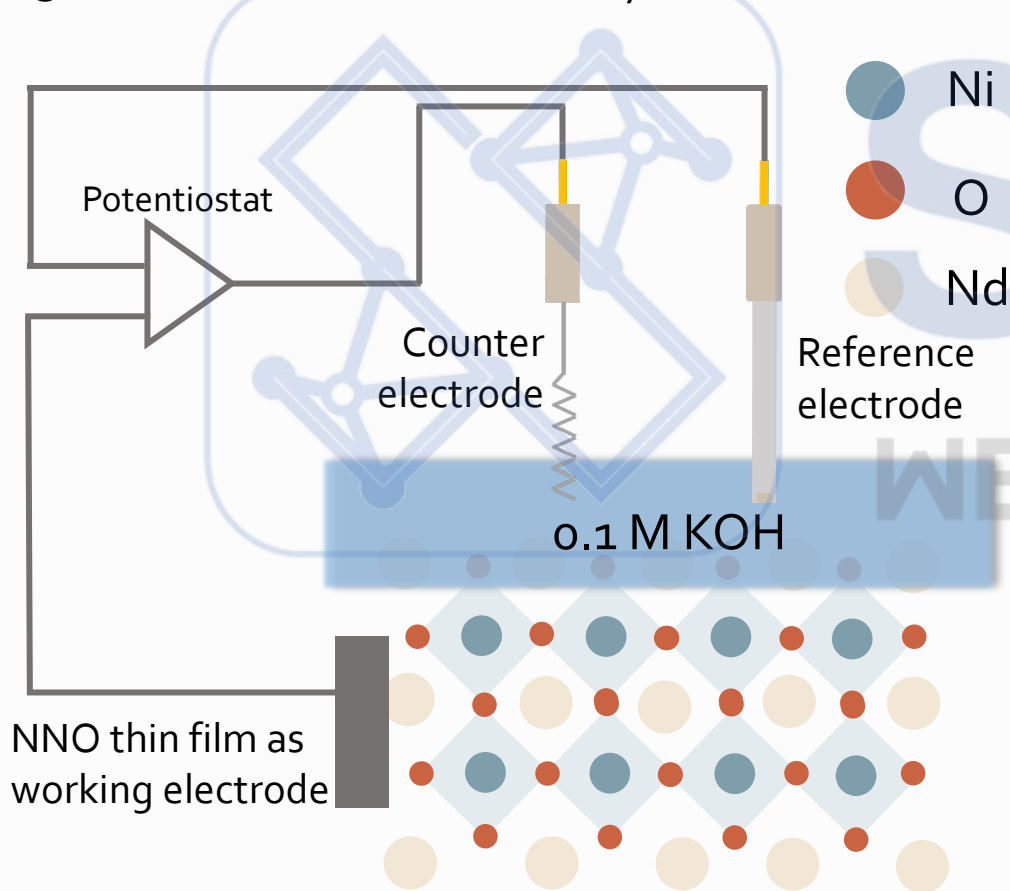


*In situ* XRD to monitor chemical expansion in  $SrCoO_x$

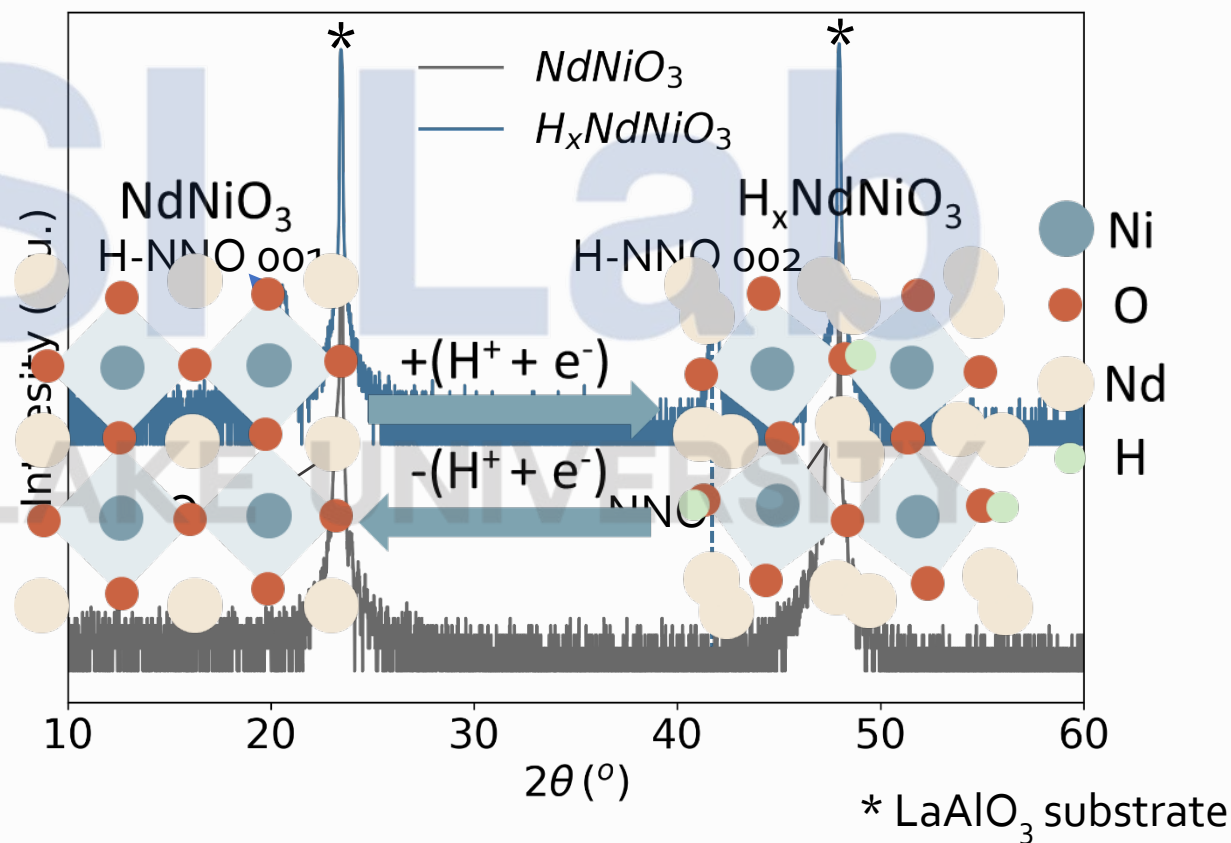


# How to probe chemical expansion: X-ray diffraction (XRD)

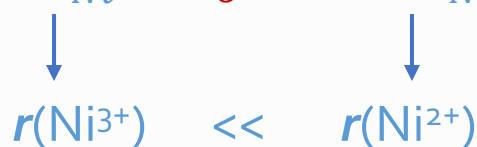
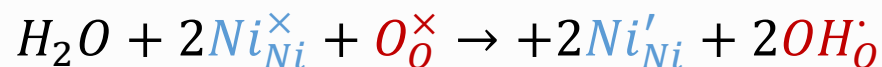
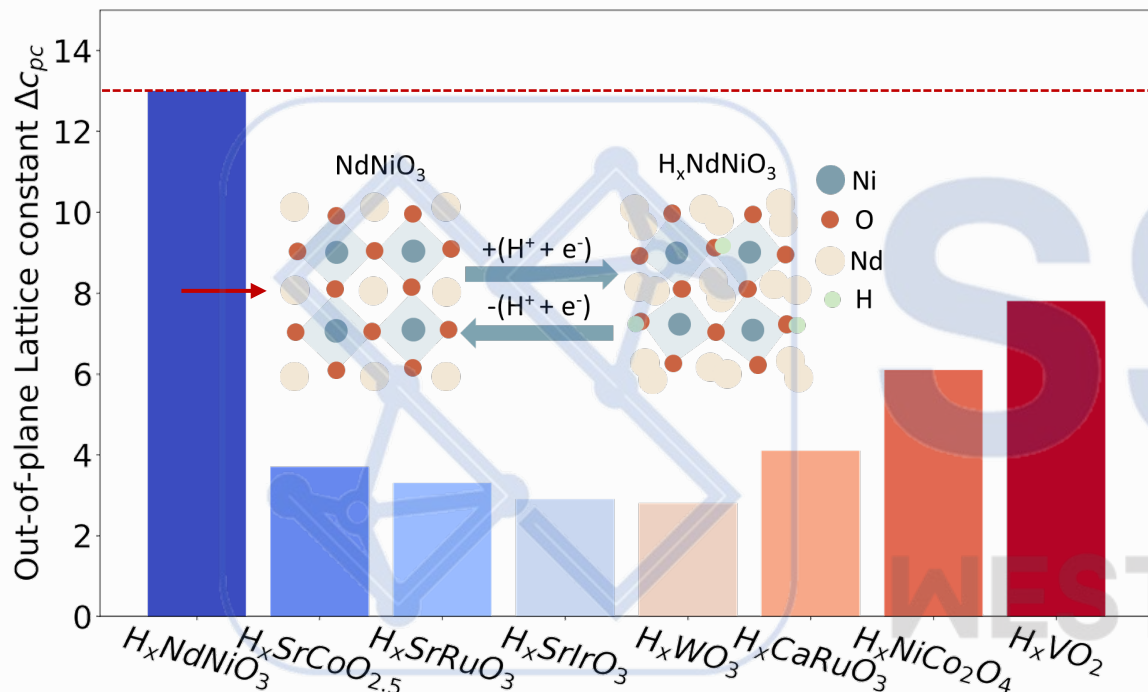
Hydrogenation of  $\text{NdNiO}_3$  thin films w/ a **3-electrode** electrochemical system



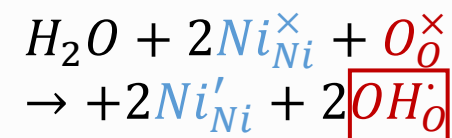
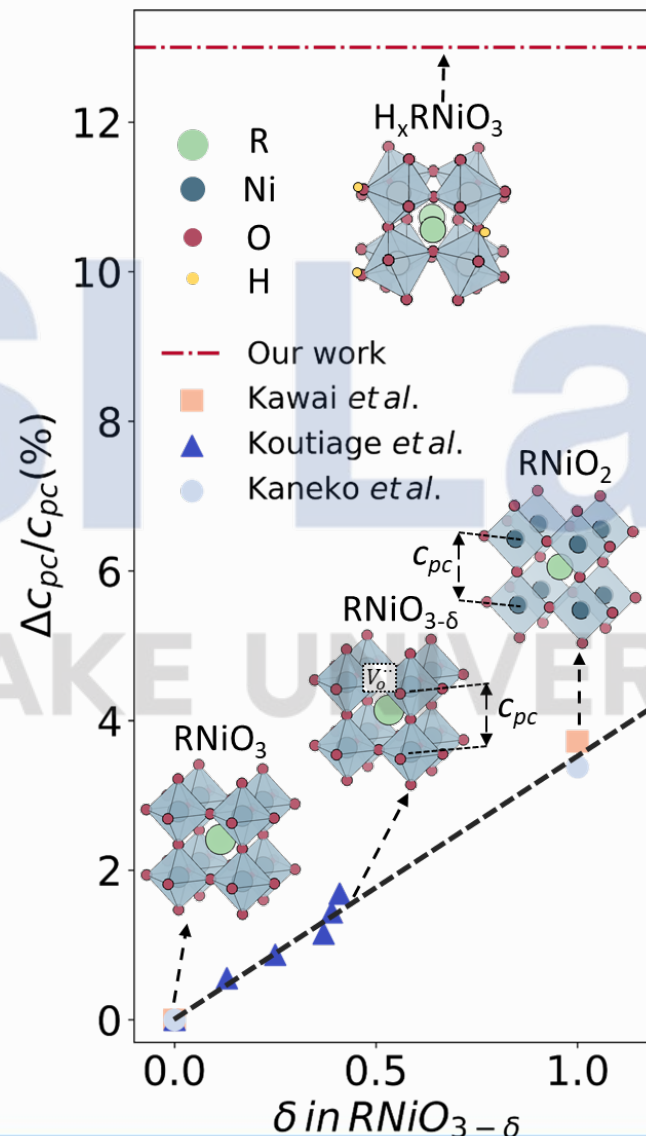
Pristine  $\text{NdNiO}_3$  (NNO)  $c_{pc} = 3.816 \text{ \AA}$   
 Hydrogenated  $\text{NdNiO}_3$  (H-NNO)  $c_{pc} = 4.312 \text{ \AA}$  }  $\Delta c/c = 13\% (!)$



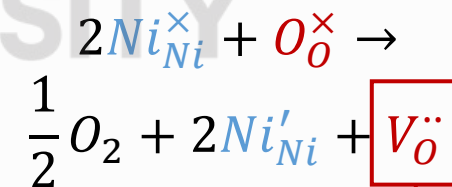
# The chemical expansion in $H_x\text{NdNiO}_3$ is by far largest in oxide thin films reported in the literature



Chemical expansion expected w/ reduced cations  
(although not as large as 13%)



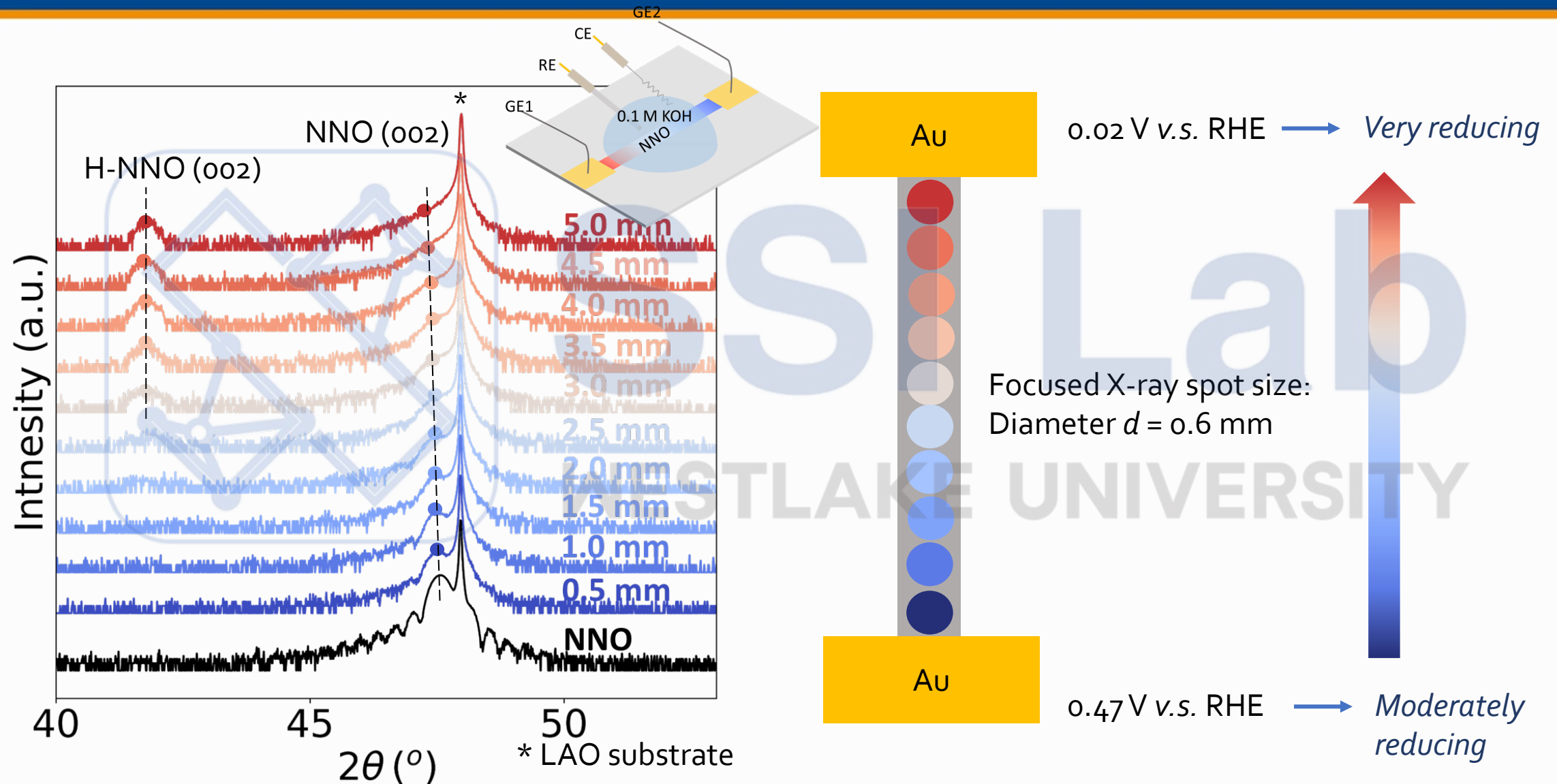
Filled oxygen vacancy  
→ **Increased expansion**



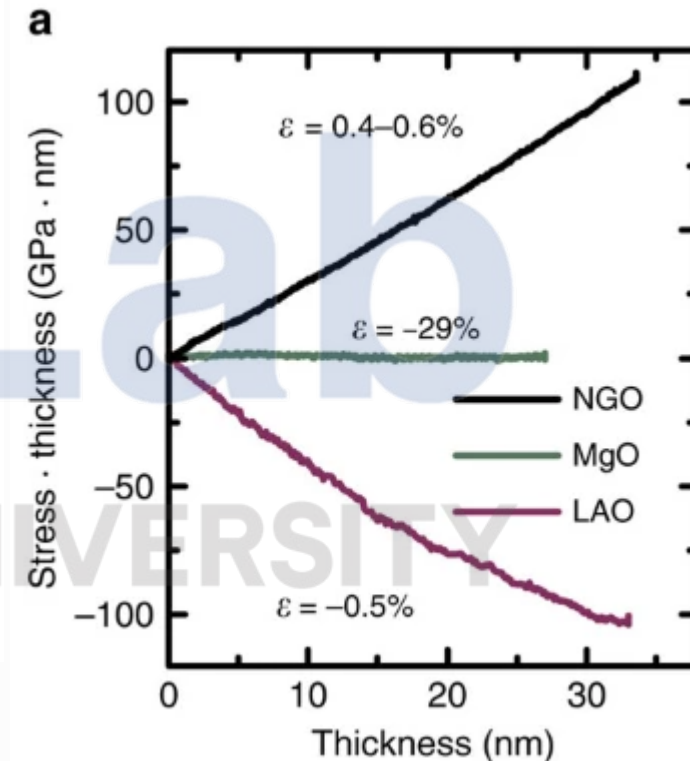
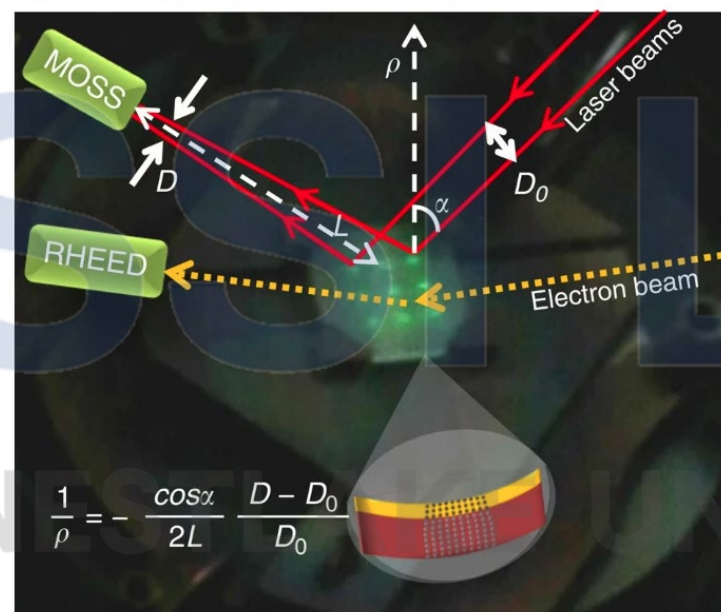
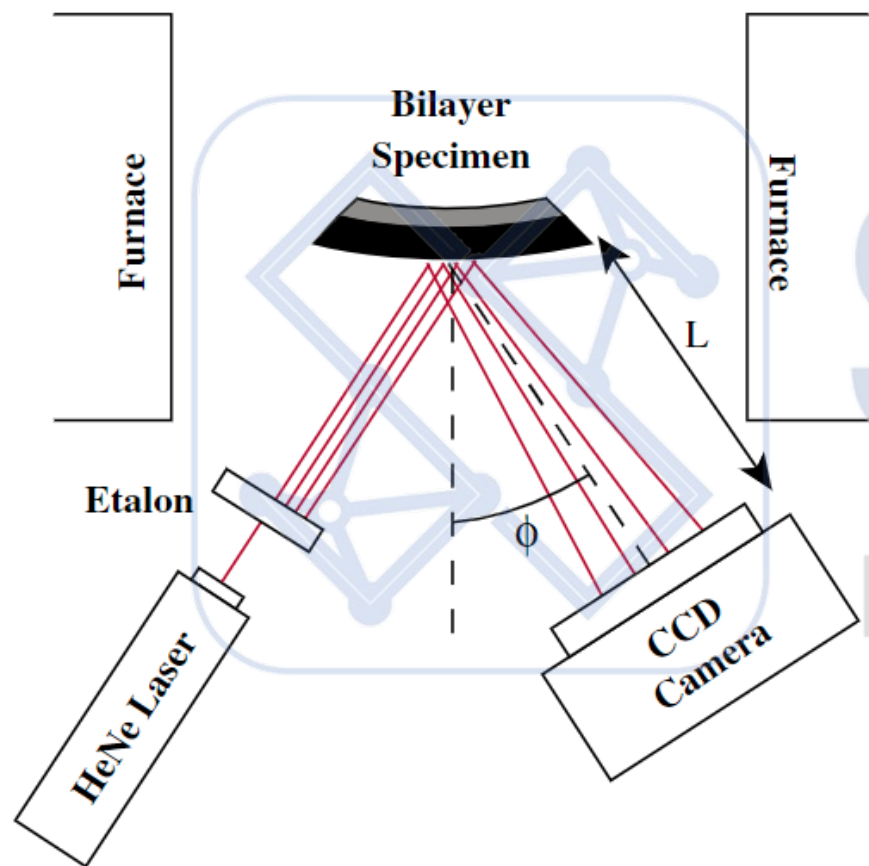
Oxygen vacancy  
→ **Contraction**



# Concentration gradient of protons in NdNiO<sub>3</sub> thin films enables efficient characterizations

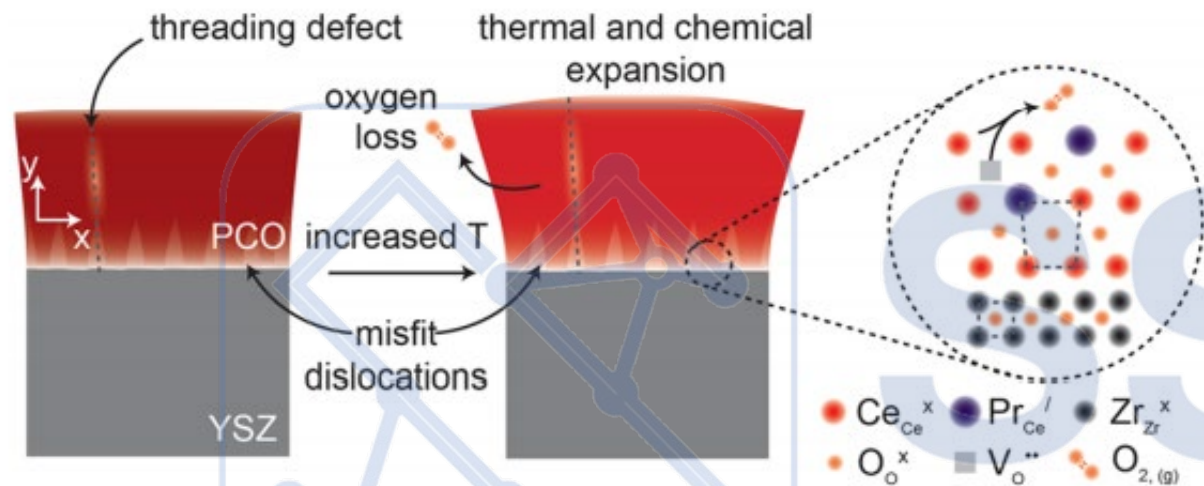




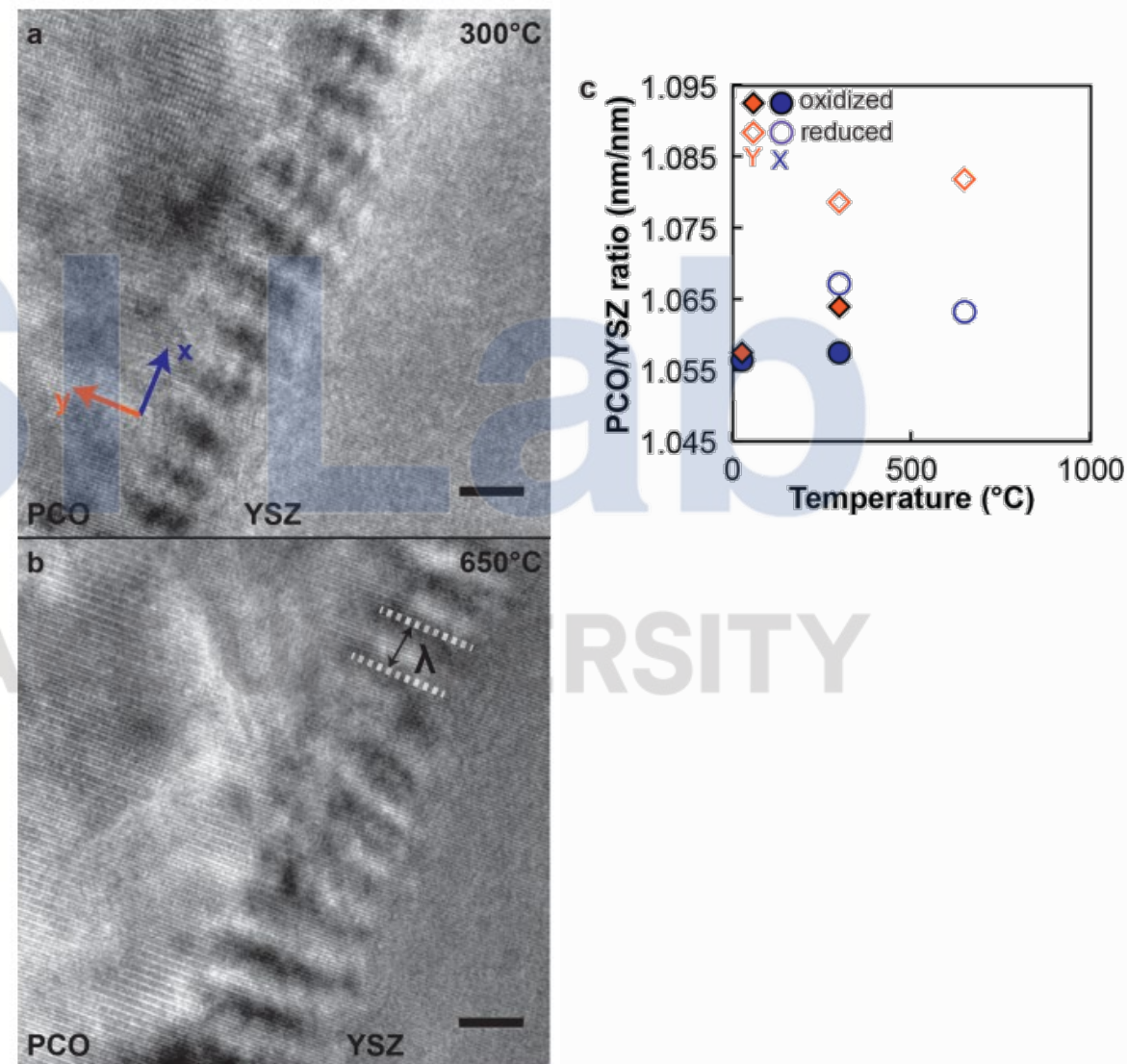


## *In situ* monitoring of stress in oxide thin films during PLD deposition

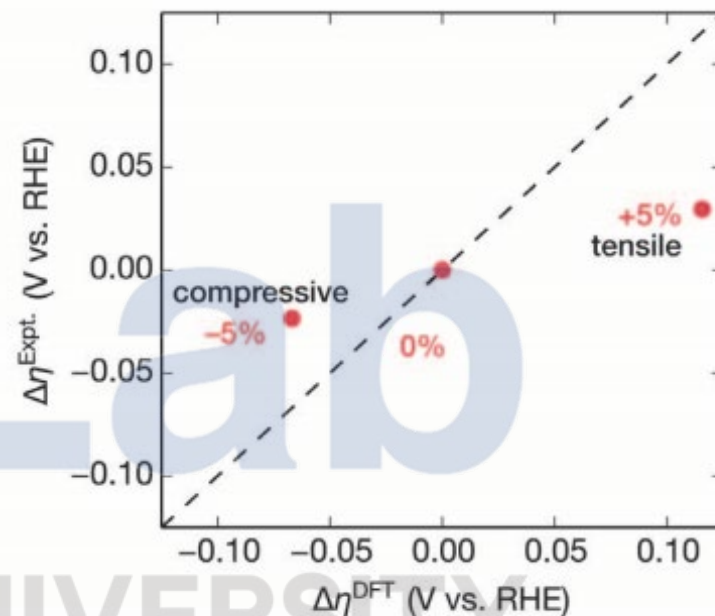
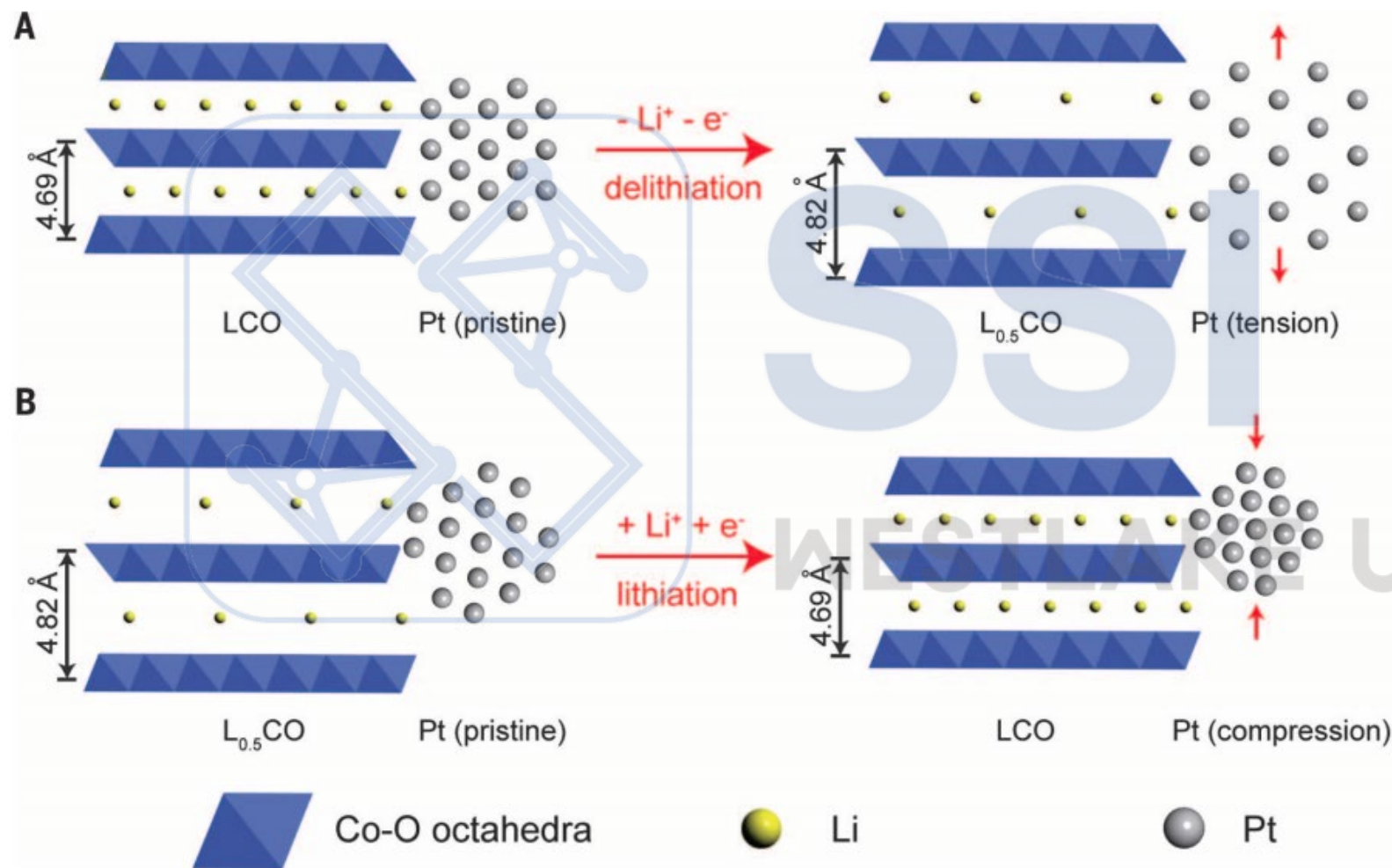
# How to probe chemical expansion: electron microscopy



- *In situ* monitoring of strain in  $(\text{Pr},\text{Ce})\text{O}_{2-x}$  thin films by using *in situ* TEM
- Lattice mismatch between substrate (YSZ) and thin film (PCO) introduces interfacial misfit dislocations
- Chemical expansion can be observed by tracking lattice parameter ratio between PCO and YSZ.



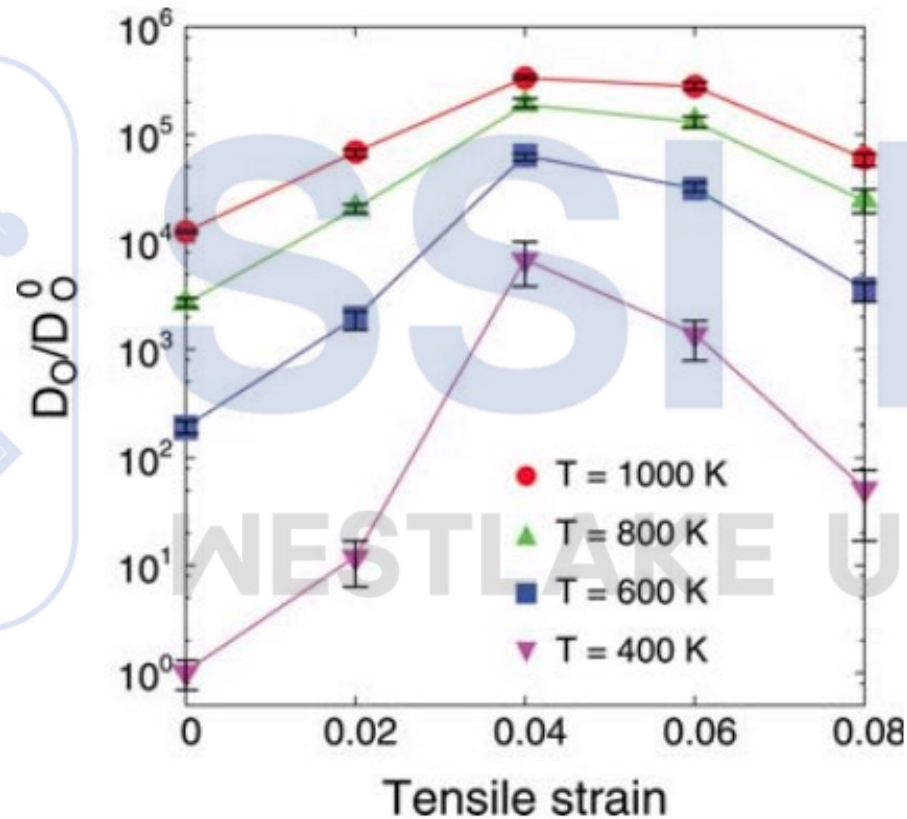
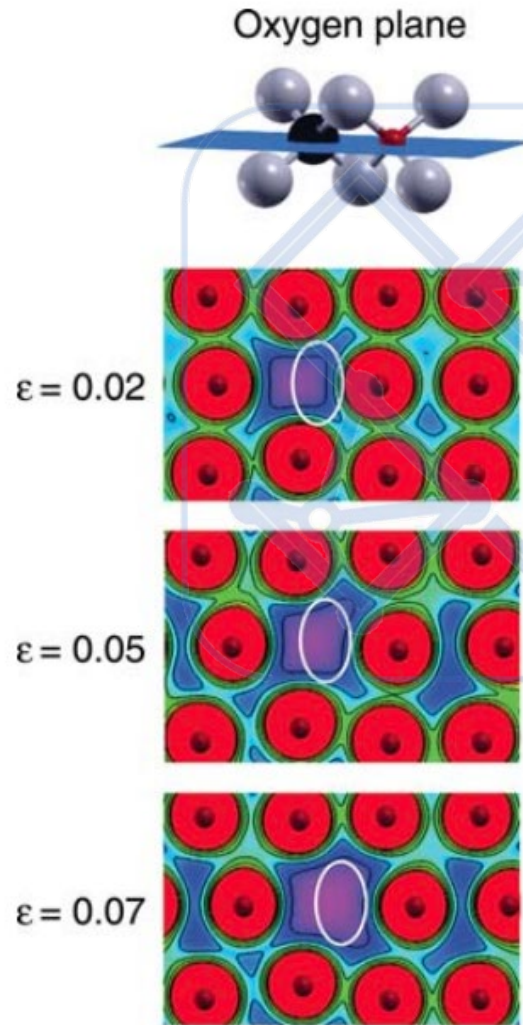
# Effect of chemical expansion on electrocatalytic activity



- Strain introduced into Pt by chemically expanding/contracting  $\text{Li}_x\text{CoO}_2$  lattice;
- Introduced strain affects the ORR activity of Pt.



# Effect of strain on ion migration



Diffusivity of oxide ion in  $(Y,Zr)O_2$  (YSZ) increases with tensile strain (up to  $\sim 10^3$ ).

$$D = D_0 \exp\left(-\frac{\Delta H_{mig}}{RT}\right)$$

w/ strain:  $\Delta H_{mig} = \Delta H_{mig}^0 + \sigma \cdot \varepsilon$

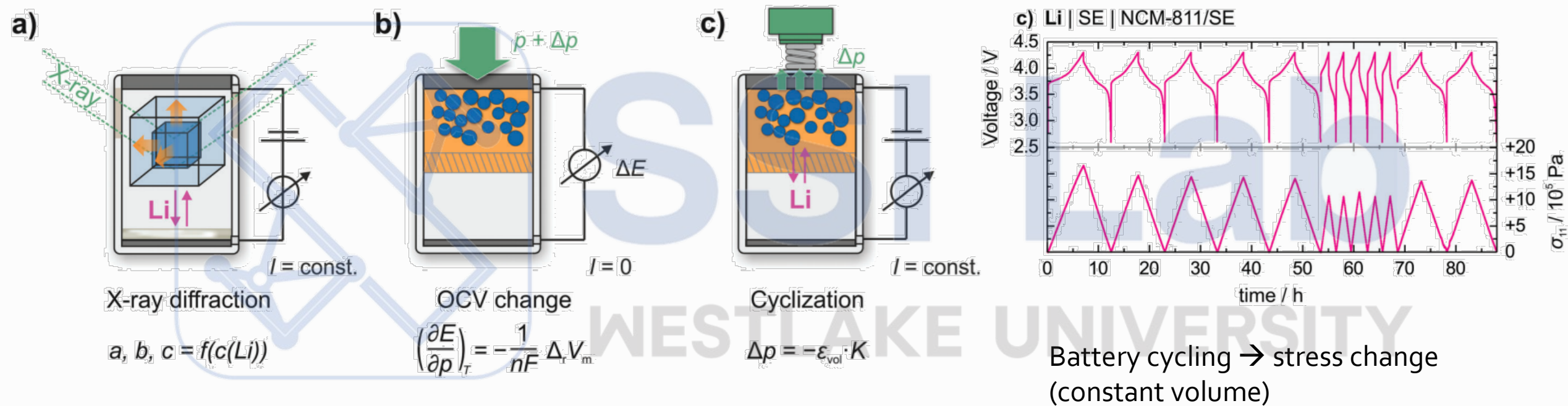
$\swarrow$  stress  
 $\searrow$  strain

$$\sigma = E \varepsilon$$

$\downarrow$   
 Young's modulus  
 (1D case)



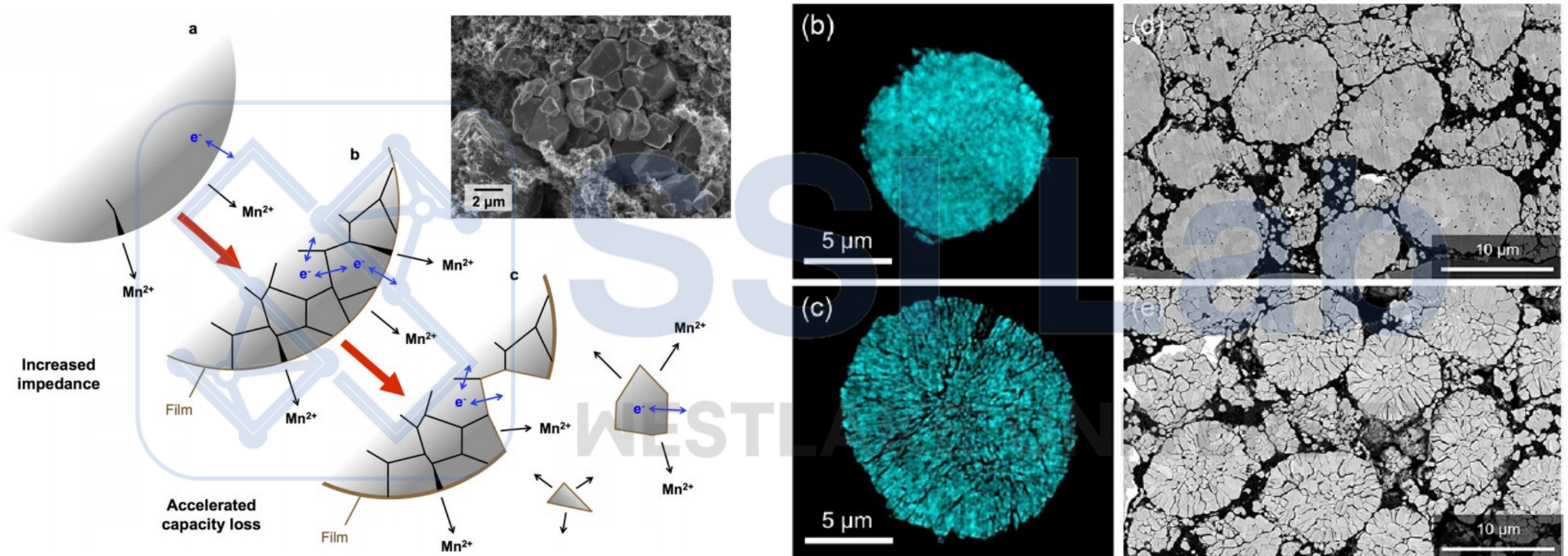
Solid state battery (SSB) → very large volume change due to (de-)intercalation of  $\text{Li}^+$



$$\mu_{\text{Li}} = \mu_i^0 + PV_{m,i} + RT \ln a_i \longrightarrow \text{OCV changes as a function of pressure } P (!)$$

hydrostatic pressure

partial molar volume



Fracture of battery cathode materials (e.g.,  $\text{Li}_x\text{Mn}_2\text{O}_4$ )

→ Capacity loss due to disconnected primary particles.

## Chemo-mechanical coupling (chemical expansion/contraction):

- What is the microscopic mechanism of lattice expansion/contraction induced by point defects?
- How to describe and quantify the change of lattice constant induced by point defects?
- What are the consequences of chemical expansion/contraction in the context of energy storage and conversion devices?

**Goal of this lecture:** you should be able to answer the questions above now (hopefully) : )



# End of Lecture 13

## Solid State Ionics Fall 2022

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