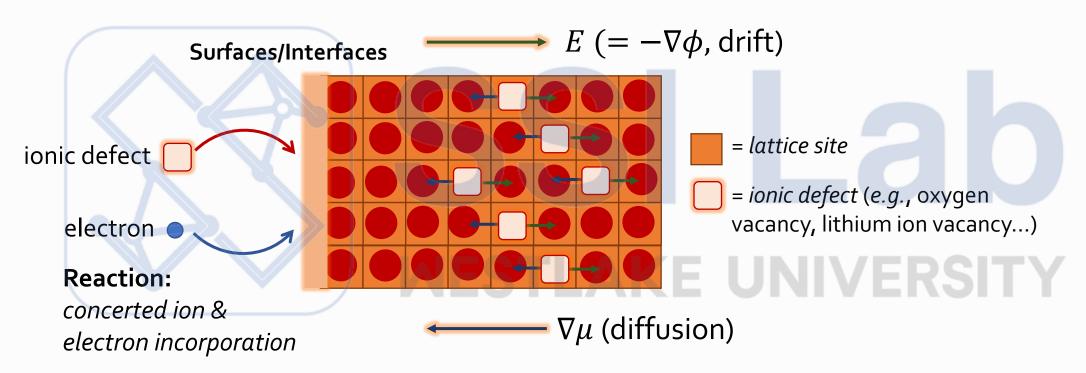




# Diffusion and reactions in solid states w/ the picture of ionic defects



**Ion motion**: *drift* + *diffusion* (similar to electrons/holes in semiconductor physics)



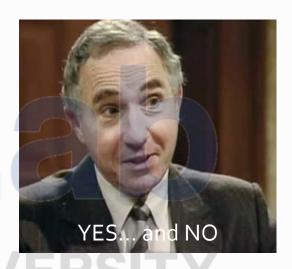
### Things we will discuss in this lecture

#### Type of diffusivities:

- What are the different types of diffusivities?
- What physical mechanism and concept does each diffusivity describe?

#### Chemical diffusivity:

- What physical process does the chemical diffusion describe?
- What are the key factors that govern the chemical diffusivity?



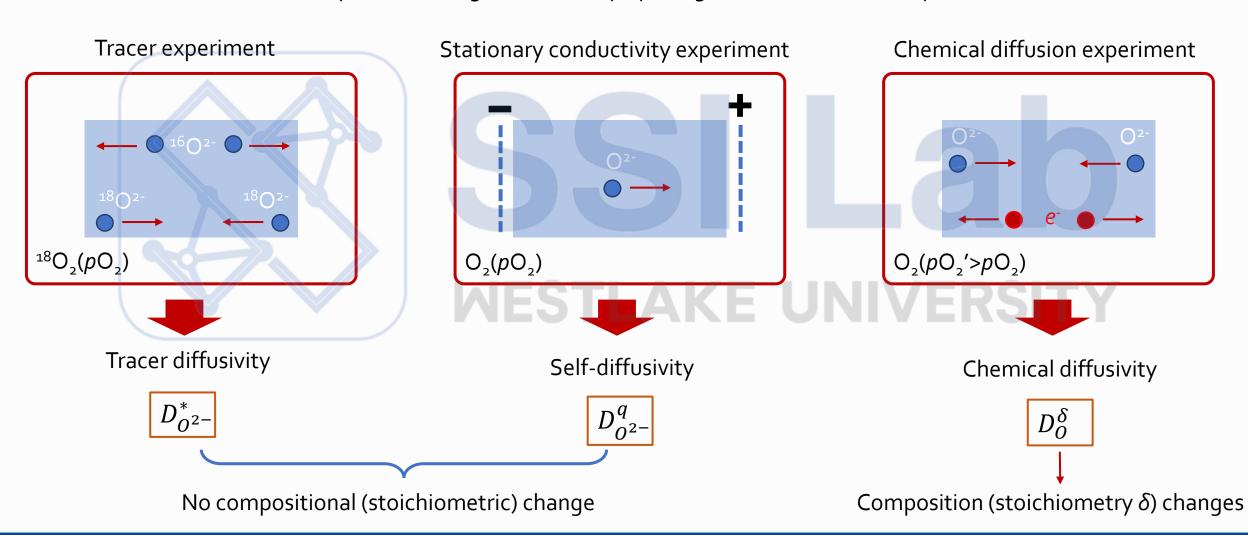
Goal of this lecture: you should be able to answer the questions above by the end of this lecture : )

A test on your intuition: Is the diffusion coefficient dependent on the concentration of ionic defects?



## Types of diffusivities (or diffusion coefficients)

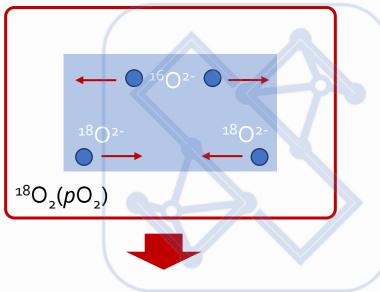
We can measure *phenomenological* diffusivity by using the three different experiments:





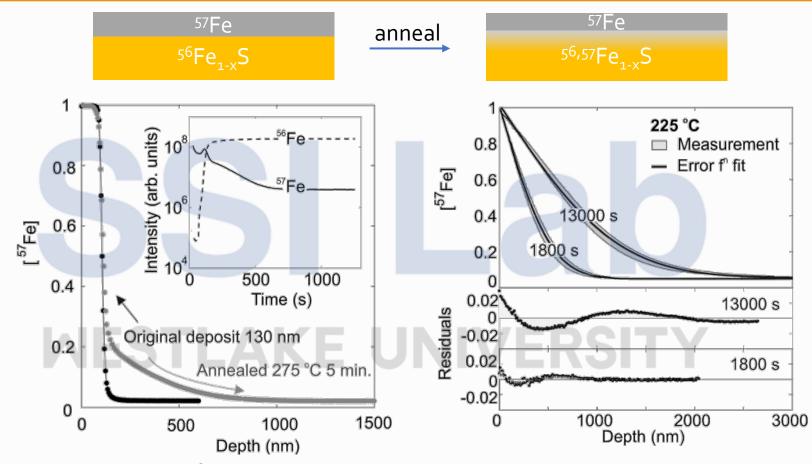
### Tracer diffusivity: isotope exchange profile

#### Tracer experiment



Tracer diffusivity

 $D_{O^{2-}}^{*}$ 



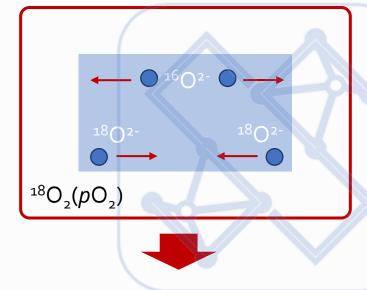
- 57Fe isotope profile measured by using secondary-ion mass spectrometry (SIMS)
- Concentration profile modeled by solving semi-infinite diffusion equation

$$c(x,t) = c_0(1 - \operatorname{erf}(\frac{x}{4D^*t}))$$



## Tracer diffusivity: isotope exchange profile

#### Tracer experiment

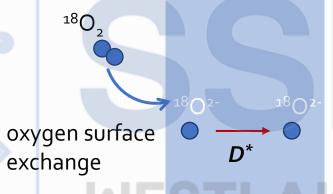


Tracer diffusivity

$$D_{0^{2-}}^{*}$$

#### Fick's 2<sup>nd</sup> Law:

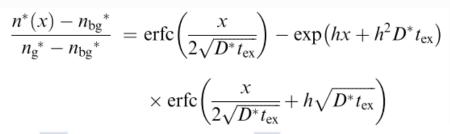
$$\frac{\partial c(x,t)}{\partial t} = D^* \frac{\partial^2 c(x,t)}{\partial x^2}$$

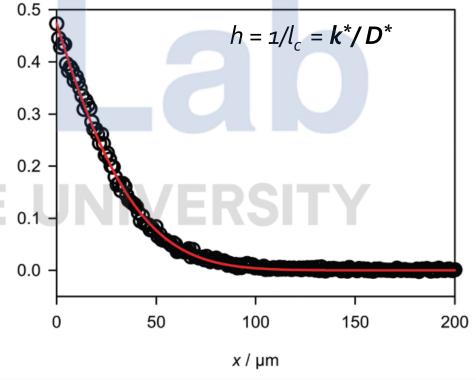


rate constant k\*

k\*: unit cm/sD\*: unit cm²/s

Critical length  $l_c = D^*/k^*$ 

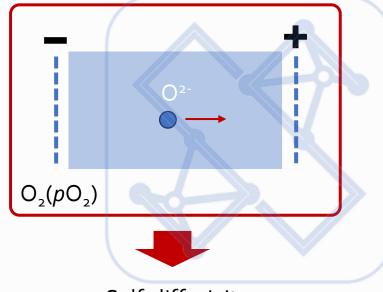






## 月 画湖大學 Self-diffusivity: steady-state conductivity measurement

#### Stationary conductivity experiment



Self-diffusivity

$$D_{O^{2-}}^{q}$$

Consider **Nernst-Einstein** relation

$$\frac{D}{M} = \frac{RT}{zF}$$

$$\sigma = c zF M = c zF D \frac{zF}{RT} = c \frac{z^2F^2}{RT}D$$

Therefore, we can get diffusivity by measuring ionic conductivity.

We have:

$$D = \frac{RT}{z^2 F^2} \frac{\sigma}{c}$$

Confusing question: which concentration should we plug in here?

$$D_{O^{2-}}^{q} = \frac{RT}{4F^2} \frac{\sigma_{O^{2-}}}{c_{O^{2-}}}$$

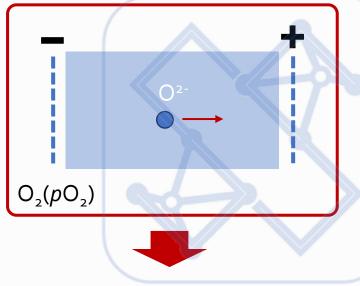
*q*: charge transport

Oxide ion  $(0^{2-})$  concentration (**NOT** defect concentration)



## 」 画湖大學 Self-diffusivity: steady-state conductivity measurement

#### Stationary conductivity experiment



Self-diffusivity

$$D_{O^{2-}}^{q}$$

$$D_{O^{2-}}^{q} = \frac{RT}{4F^2} \frac{\sigma_{O^{2-}}}{c_{O^{2-}}}$$

q: charge transport

Oxide ion ( $O^{2-}$ ) concentration (**NOT** defect concentration)

In reality, the  $O^{2-}$  conductivity is contributed by defects (either oxygen vacancies  $V_0^{"}$  or oxygen interstitials  $O_i^{"}$ ).

If we assume that oxygen interstitials  $O_i^{\prime\prime}$  are the major oxygen ionic defects,

we have:

$$\sigma_{O^{2-}} = \sigma_{O_i''} = 2Fc_{O_i''}M_{O_i''}$$

$$D_{O^{2-}}^{q} = \frac{RT}{4F^2} \frac{\sigma_{O^{2-}}}{c_{O^{2-}}} = \frac{RT}{4F^2} \frac{\sigma_{O_i^{\prime\prime}}}{c_{O_i^{\prime\prime}}} \frac{c_{O_i^{\prime\prime}}}{c_{O^{2-}}} = D_{O_i^{\prime\prime}} \frac{c_{O_i^{\prime\prime}}}{c_{O^{2-}}}$$

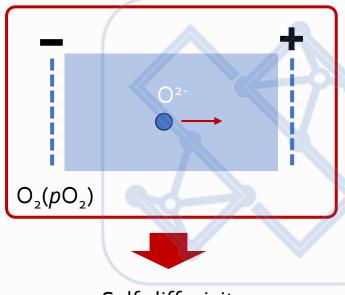
$$D_{O^{2-}}^q = D_{O_i^{\prime\prime}} \frac{c_{O_i^{\prime\prime}}}{c_{O^{2-}}} = x_{O_i^{\prime\prime}} D_{O_i^{\prime\prime}}$$

Fraction of oxygen interstitials  $O_i^{"}$ 



## Self-diffusivity: steady-state conductivity measurement

#### Stationary conductivity experiment



Self-diffusivity

$$D_{O^{2-}}^q$$

$$D_{O^{2-}}^q = D_{O_i^{\prime\prime}} \frac{c_{O_i^{\prime\prime}}}{c_{O^{2-}}} = x_{O_i^{\prime\prime}} D_{O_i^{\prime\prime}}$$

Fraction of oxygen interstitials  $O_i^{\prime\prime}$ 

Since 
$$x_{O_i^{\prime\prime}}=c_{O_i^{\prime\prime}}/c_{O^{2-}}$$
 is far less than 1  $\longrightarrow$   $D_{O_i^{\prime\prime}}\gg D_{O^{2-}}^q$ 

The self-diffusivity of defect species is *much higher* than that of ionic species.

Contain information on both ion migration and defect formation

$$D_{O^{2-}}^{q} = x_{O_i''} D_{O_i''}$$

Related to defect formation

$$\propto \exp\left(-\frac{\Delta H_f}{k_B T}\right)$$

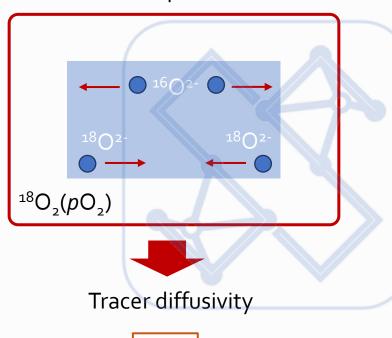
Reflect the "true" mechanism of ion migration

$$\propto \exp\left(-\frac{\Delta H_{mig}}{k_B T}\right)$$

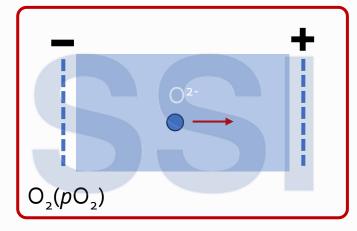


## Relationship between tracer diffusivity and self-diffusivity

#### Tracer experiment



Stationary conductivity experiment



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Self-diffusivity

$$D_{O^{2-}}^q$$

No compositional (stoichiometric) change

Tracer diffusivity and self-diffusivity are usually on the same order of magnitude, since they both describe the process **without** compositional changes.

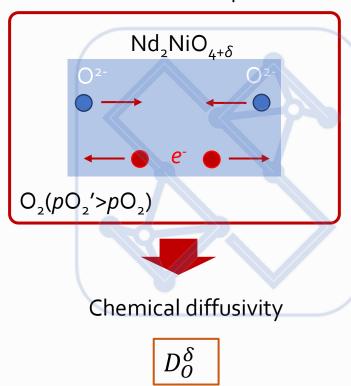
$$D_{O^{2-}}^* = H_{O^{2-}} D_{O^{2-}}^q$$

Haven ratio

- Haven ratio mainly describes the correlations between each hops.
- $H_{O^{2-}}$  is usually of the order of 1



#### Chemical diffusion experiment



Composition (stoichiometry  $\delta$ ) changes

Very different from tracer diffusivity and self-diffusivity, chemical diffusion describe the process that involves *compositional or stoichiometric changes*.

$$e.g.$$
,  $Nd_2NiO_{4+\delta} + \Delta\delta/2O_2 \rightarrow Nd_2NiO_{4+\delta+\Delta\delta}$ 

In order to change oxygen stoichiometry ( $\Delta \delta$ ), oxide ions  $O^{2-}$  and electrons  $e^{-}$  must diffuse to opposite directions, to maintain *charge neutrality*.

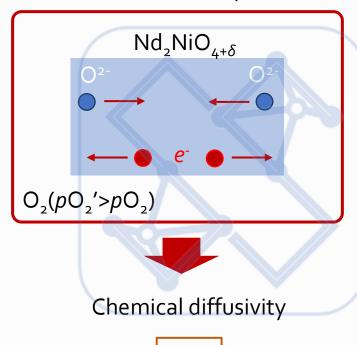
Therefore, chemical diffusivity is called *ambipolar diffusivity*, meaning that chemical diffusion involves the motion of *two charged species*.

In  $Nd_2NiO_{4+\delta}$  the change of oxygen stoichiometry can be expressed by the defect chemical reaction below:

$$\frac{1}{2}O_2 \rightleftharpoons O_0^{\times} \rightleftharpoons O_i^{\prime\prime} + 2h^{\cdot}$$

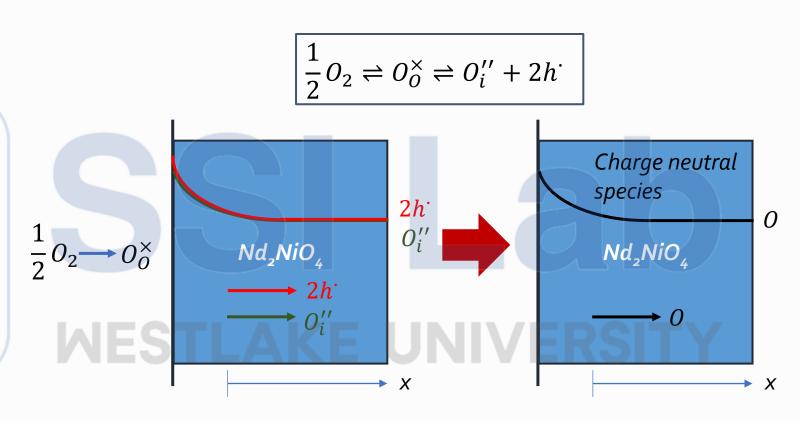


#### Chemical diffusion experiment





Composition (stoichiometry  $\delta$ ) changes



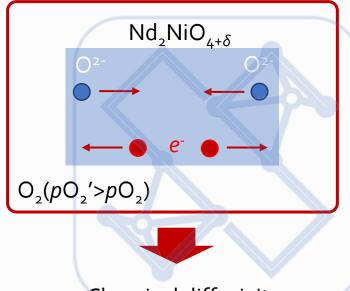
**Our goal:** find the expression of chemical diffusivity based on properties of defects ( $O_i'' \& h$ ') so that we can use Fick's law to predict the concentration profile, *i.e.*:

$$J_O = -D_O^{\delta} \nabla c_O$$
 or  $J_O = -D_O^{\delta} \frac{\partial c_O}{\partial x}$  (1D)



## 

#### Chemical diffusion experiment



Chemical diffusivity



Composition (stoichiometry  $\delta$ ) changes

$$\boxed{\frac{1}{2}O_2 \rightleftharpoons O_0^{\times} \rightleftharpoons O_i^{\prime\prime} + 2h^{\cdot}}$$

Flux of oxygen interstitials:

$$J_{O_{i}^{\prime\prime}} = -\frac{\sigma_{O_{i}^{\prime\prime}}}{4F^{2}} \nabla \tilde{\mu}_{O_{i}^{\prime\prime}} = -\frac{\sigma_{O_{i}^{\prime\prime}}}{4F^{2}} (\nabla \mu_{O_{i}^{\prime\prime}} - 2F \nabla \phi)$$

Flux of holes: 
$$J_{h^{\cdot}} = -\frac{\sigma_{h^{\cdot}}}{F^2} \nabla \tilde{\mu}_{h^{\cdot}} = -\frac{\sigma_{h^{\cdot}}}{F^2} (\nabla \mu_{h^{\cdot}} + F \nabla \phi)$$

To satisfy the local electro-neutral condition (we will discuss why it holds):

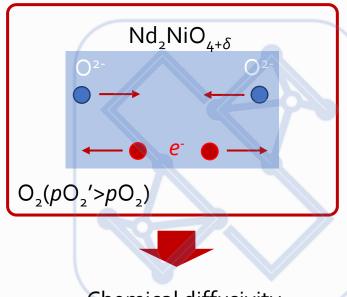
$$J_{O_i^{\prime\prime}} = \frac{1}{2} J_{h^{\cdot}}$$

$$\frac{\sigma_{O_i^{\prime\prime}}}{4F^2} \left( \nabla \mu_{O_i^{\prime\prime}} - 2F \nabla \phi \right) = \frac{\sigma_{h^{\cdot}}}{2F^2} \left( \nabla \mu_{h^{\cdot}} + F \nabla \phi \right) \longrightarrow \begin{bmatrix} \sigma_{O_i^{\prime\prime}} \\ 4F^2 \\ \end{bmatrix} \nabla \mu_{O_i^{\prime\prime}} - \frac{\sigma_{h^{\cdot}}}{2F^2} \nabla \mu_{h^{\cdot}} \\ = \frac{\sigma_{O_i^{\prime\prime}} + \sigma_{h^{\cdot}}}{2F} \nabla \phi$$



## 

#### Chemical diffusion experiment



Chemical diffusivity

$$D_O^{\delta}$$

Composition (stoichiometry  $\delta$ ) changes

$$\frac{1}{2}O_2 \rightleftharpoons O_0^{\times} \rightleftharpoons O_i^{\prime\prime} + 2h^{\cdot}$$

$$\frac{\sigma_{O_i^{\prime\prime}}}{4F^2}\nabla\mu_{O_i^{\prime\prime}} - \frac{\sigma_{h^{\cdot}}}{2F^2}\nabla\mu_{h^{\cdot}} = \frac{\sigma_{O_i^{\prime\prime}} + \sigma_{h^{\cdot}}}{2F}\nabla\phi$$

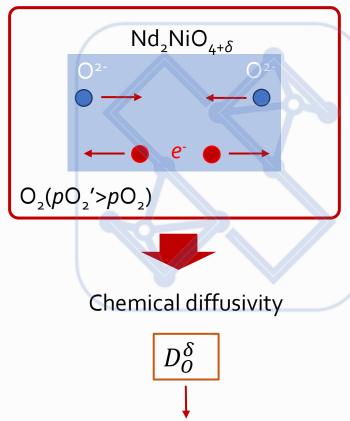
$$\begin{split} J_{o} &= J_{o_{i}''} = -\frac{\sigma_{o_{i}''}}{4F^{2}} \Big( \nabla \mu_{o_{i}''} - 2F \nabla \phi \Big) = -\frac{\sigma_{o_{i}''}}{4F^{2}} \nabla \mu_{o_{i}''} + \frac{\sigma_{o_{i}''}}{2F} \nabla \phi \\ &= -\frac{\sigma_{o_{i}''}}{4F^{2}} \nabla \mu_{o_{i}''} + \frac{\sigma_{o_{i}''}}{\sigma_{o_{i}''} + \sigma_{h}} (\frac{\sigma_{o_{i}''}}{4F^{2}} \nabla \mu_{o_{i}''} - \frac{\sigma_{h}}{2F^{2}} \nabla \mu_{h}) \end{split}$$

$$J_{0} = -\frac{1}{4F^{2}} \frac{\sigma_{h} \cdot \sigma_{O_{i}^{"}}}{\sigma_{O_{i}^{"}} + \sigma_{h}} (\nabla \mu_{O_{i}^{"}} + 2\nabla \mu_{h}) \qquad \text{If we denote: } \mu_{O} = \mu_{O_{i}^{"}} + 2\mu_{h}$$

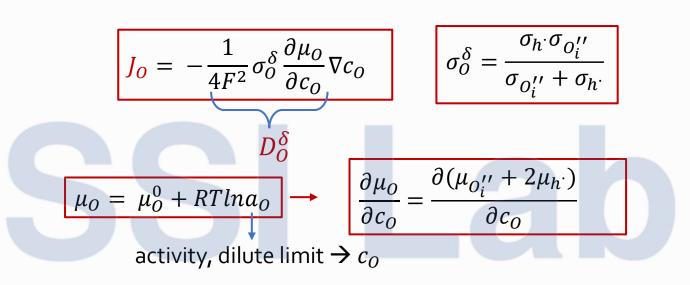
$$J_{O} = -\frac{1}{4F^{2}} \frac{\sigma_{h} \cdot \sigma_{O_{i}^{\prime\prime}}}{\sigma_{O_{i}^{\prime\prime}} + \sigma_{h}} \nabla \mu_{O} = -\frac{1}{4F^{2}} \sigma_{O}^{\delta} \frac{\partial \mu_{O}}{\partial c_{O}} \nabla c_{O} \qquad \longleftarrow \qquad \sigma_{O}^{\delta} = \frac{\sigma_{h} \cdot \sigma_{O_{i}^{\prime\prime}}}{\sigma_{O_{i}^{\prime\prime}} + \sigma_{h}}$$



#### Chemical diffusion experiment



Composition (stoichiometry  $\delta$ ) changes

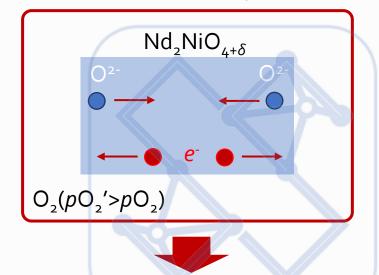


$$D_{O}^{\delta} = \frac{1}{4F^{2}} \sigma_{O}^{\delta} \frac{\partial \mu_{O}}{\partial c_{O}} = \frac{RT}{4F^{2}} \frac{\sigma_{h} \cdot \sigma_{O_{i}^{\prime\prime}}}{\sigma_{O_{i}^{\prime\prime}} + \sigma_{h} \cdot} \left(\frac{1}{c_{O_{i}^{\prime\prime}}} + \frac{4}{c_{h} \cdot}\right)$$

$$(\partial c_{O} = \partial c_{O_{i}^{\prime\prime}} = \frac{1}{2} \partial c_{h} \cdot)$$
"electrical" "ionic"



#### Chemical diffusion experiment



Chemical diffusivity

$$D_O^{\delta}$$

Composition (stoichiometry  $\delta$ ) changes

$$D_{O}^{\delta} = \frac{1}{4F^{2}} \sigma_{O}^{\delta} \frac{\partial \mu_{O}}{\partial c_{O}} = \frac{1}{4F^{2}} \frac{\sigma_{h} \cdot \sigma_{O_{i}^{"}}}{\sigma_{O_{i}^{"}} + \sigma_{h}} (\frac{1}{c_{O_{i}^{"}}} + \frac{4}{c_{h}})$$

(valid at dilute limit)

"electrical" ".

"ionic"

$$D_O^{\delta} = \frac{RT}{4F^2} \frac{\sigma_O^{\delta}}{c_O^{\delta}}$$

"electrical": harmonic mean conductivity

"ionic": harmonic mean concentration

$$\frac{1}{\sigma_O^{\delta}} = \frac{1}{\sigma_{O_i^{\prime\prime}}} + \frac{1}{\sigma_{h}}$$

$$\frac{1}{c_O^{\delta}} = \frac{1}{c_{O_i''}} + \frac{2^2}{c_{h'}}$$

$$\sigma_{o_{i}^{\prime\prime}} = 2c_{o_{i}^{\prime\prime}}FM_{o_{i}^{\prime\prime}} = \frac{RT}{4F^{2}}c_{o_{i}^{\prime\prime}}D_{o_{i}^{\prime\prime}}$$

$$\sigma_{h^{\cdot}} = c_{h^{\cdot}} F M_{h^{\cdot}} = \frac{RT}{F^2} c_{h^{\cdot}} D_{h^{\cdot}}$$

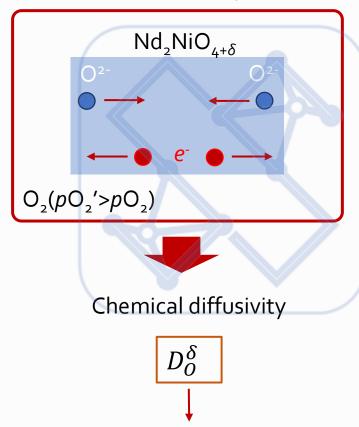
$$D_O^{\delta} = t_{h'} D_{O_i^{\prime\prime}} + t_{O_i^{\prime\prime}} D_{h'}$$

$$t_{h'} = \frac{\sigma_{h'}}{\sigma_{0''_i} + \sigma_{h'}}$$

Transference number



#### Chemical diffusion experiment



Composition (stoichiometry  $\delta$ ) changes

$$D_O^{\delta} = t_{h'} D_{O_i^{\prime\prime}} + t_{O_i^{\prime\prime}} D_{h'}$$

$$t_{h'} = \frac{\sigma_{h'}}{\sigma_{O_i''} + \sigma_{h'}}$$

$$t_{O_i^{\prime\prime}} = \frac{\sigma_{O_i^{\prime\prime}}}{\sigma_{O_i^{\prime\prime}} + \sigma_{h}}.$$

Transference number

If 
$$\sigma_{h^{\cdot}} \gg \sigma_{O_i^{\prime\prime}}$$
, then  $t_{h^{\cdot}} \rightarrow 1$  and if  $D_{h^{\cdot}}$  is not too large  $\longrightarrow$ 

#### Note:

- In this scenario,  $D_0^\delta \approx D_{o_i''} \gg D_{o^{2-}}^q$  (recall  $D_{o^{2-}}^q = x_{o_i''}D_{o_i''}$ , while usually  $x_{o_i''} \ll 1$ )
- If the system deviate from the dilute limit, then:

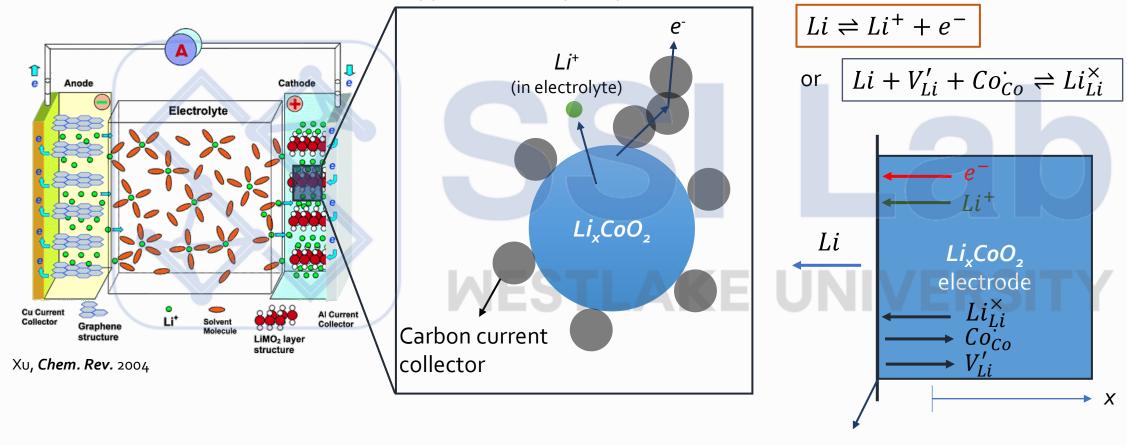
$$D_O^{\delta} = \Gamma_{h'} t_{h'} D_{O_i''} + \Gamma_{O_i''} t_{O_i''} D_{h'}$$

Γ: thermodynamic factor



### Microscopic picture: how is Li intercalated into cathodes?

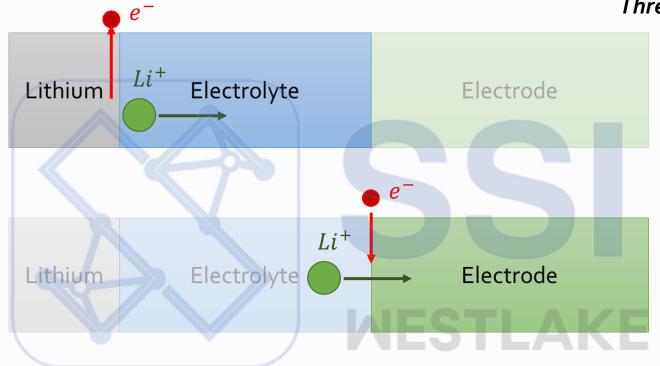
**Question:** how does *Li*-intercalation happen microscopically?



*Li<sub>x</sub>CoO<sub>2</sub>*/carbon/electrolyte interface



### Kinetics and transport in Li-ion batteries



Three decisive mechanistic steps of a Li-ion battery

#### Ion transport

(through electrolyte)

 $\tau \sim ns$ 

Ion transfer

(at electrolyte/electrode interface)

 $\sim \mu s$ 

Lithium Electrolyte Li<sup>+</sup> Electrode

Maier, Journal of Power Sources 174 (2007) 569-574

Chemical diffusion

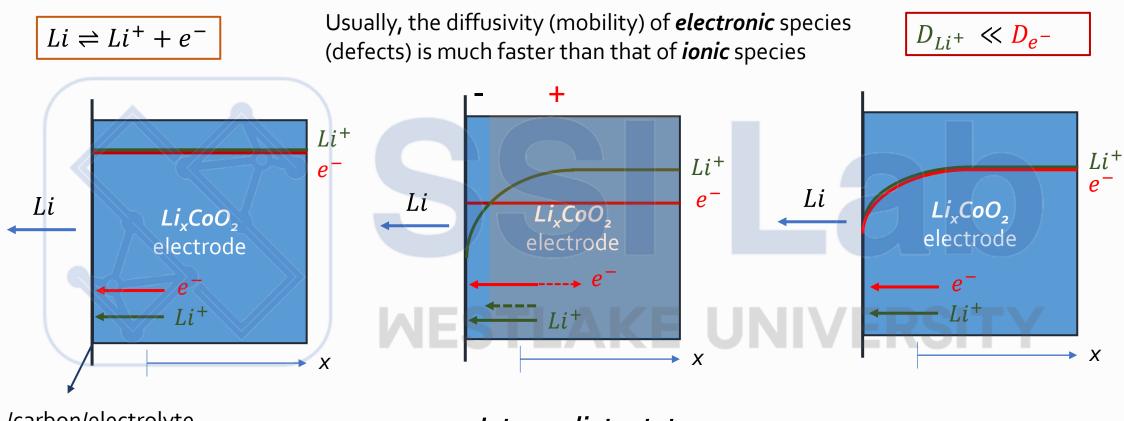
(in electrode bulk)

 $\tau \sim \min, hr, yr$ 

Depend on the size of electrodes!



# Ambipolar diffusion: diffusion process involving both ions and electrons



Li<sub>x</sub>CoO₂/carbon/electrolyte interface *Initial state* 

Local charge neutral

Intermediate state (imaginary)

Local charge ≠ o

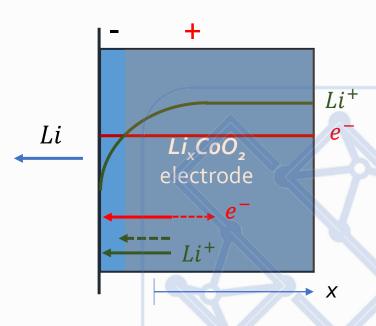
Final state

Local charge neutral

The electric field **slows down** the fast-diffusing species and **speeds up** the slow diffusing species.



## Note: the bulk of the sample must be charge neutral



Intermediate state (imaginary)

Local charge ≠ o

This picture on the left is an exaggeration to help you to understand the role of accelerating/decelerating electric field;

In a real world, the *charge neutrality* still holds (except for the interfaces)

We can do a simple back-on-envelope calculation:

$$E = -\frac{\partial \phi}{\partial x} \rightarrow \text{Electrostatic potential} \quad \frac{\partial E}{\partial x} = \frac{\partial^2 \phi}{\partial x^2} = -\frac{\rho}{\varepsilon_0 \varepsilon_r} \rightarrow \text{Charge density} \\ \varepsilon_0 : \text{vacuum permittivity} \\ \varepsilon_r : \text{relative permittivity}$$

Electric field

**Poisson's equation**: connecting charge with potential

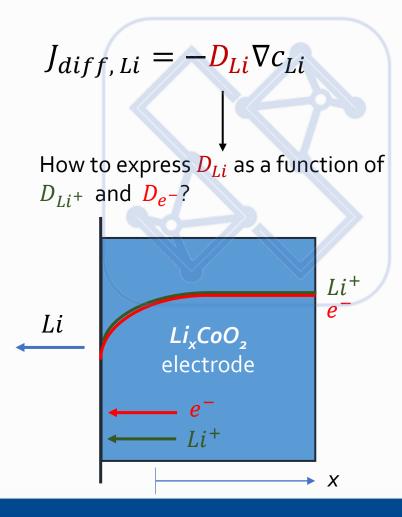
$$\frac{\Delta E}{\Delta x} = -\frac{\rho}{\varepsilon_0 \varepsilon_r} = -\frac{Fc}{\varepsilon_0 \varepsilon_r} = -\frac{96500C/mol \times 1mol/L}{8.8 \times 10^{-12} F/m \times 10} = 1 \times 10^{18} V/m^2 \qquad (c = 1 M)$$

$$\Delta E = \frac{\Delta V}{\Delta x} = 1 \times 10^{18} \ V/m^2 \Delta x$$
 If  $\Delta x = 1 \ \text{nm} \longrightarrow \Delta V = 1 \ V \ (!) \longrightarrow$  Unrealistic



## How to solve for the diffusivity?

#### Fick's first law



$$Li \rightleftharpoons Li^+ + e^-$$

Key equation: local equilibrium  $|\mu_{Li} = \tilde{\mu}_{e^-} + \tilde{\mu}_{Li^+}|$ 

$$\mu_{Li} = \tilde{\mu}_{e^-} + \tilde{\mu}_{Li^+}$$

Flux: 
$$J_i = -\frac{\sigma_i}{z_i^2 F^2} \nabla \widetilde{\boldsymbol{\mu}}_i$$

Ionic flow: 
$$J_{Li^+} = -\frac{\sigma_{Li^+}}{F^2} \nabla \tilde{\mu}_{Li^+}$$

$$\frac{D_i c_i}{RT} = \frac{\sigma_i}{z_i^2 F^2}$$

Electronic flow: 
$$J_{e^-} = -\frac{\sigma_{e^-}}{F^2} \nabla \tilde{\mu}_{e^-}$$

At steady state:  $J_{Li} = J_{Li^+} = J_{e^-}$ 

$$\mu_{Li} = \tilde{\mu}_{e^{-}} + \tilde{\mu}_{Li^{+}} \longrightarrow \nabla \mu_{Li} = \nabla \tilde{\mu}_{e^{-}} + \nabla \tilde{\mu}_{Li^{+}}$$

$$J_{Li^{+}} = J_{e^{-}} \longrightarrow \sigma_{Li^{+}} \nabla \tilde{\mu}_{Li^{+}} = \sigma_{e^{-}} \nabla \tilde{\mu}_{e^{-}}$$

$$\nabla \mu_{Li} = (1 + \frac{\sigma_{e^{-}}}{\sigma_{Li^{+}}}) \nabla \tilde{\mu}_{e^{-}}$$

$$\nabla \mu_{Li} = \left(1 + \frac{\sigma_{e^-}}{\sigma_{Li^+}}\right) \nabla \tilde{\mu}_{e^-}$$

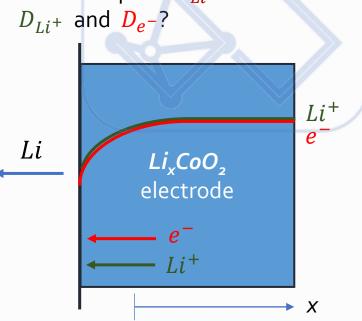


## J 西湖大學 How to solve for the diffusivity?

#### Fick's first law

$$J_{diff,Li} = -D_{Li} \nabla c_{Li}$$
How to express  $D_{Li}$  as a function of

 $D_{Li}^+$  and  $D_e^-$ ?



$$\nabla \mu_{Li} = \left(1 + \frac{\sigma_{e^-}}{\sigma_{Li^+}}\right) \nabla \tilde{\mu}_{e^-}$$

$$J_{Li} = J_{e^-} = -\frac{\sigma_{e^-}}{F^2} \nabla \tilde{\mu}_{e^-} = -\frac{1}{F^2} \frac{\sigma_{e^-} \sigma_{Li^+}}{\sigma_{e^-} + \sigma_{Li^+}} \nabla \mu_{Li}$$

$$J_{Li} = -rac{1}{F^2} rac{\sigma_e - \sigma_{Li^+}}{\sigma_{e^-} + \sigma_{Li^+}} 
abla \mu_{Li}$$
 Compare:  $J_i = -rac{\sigma_i}{{z_i}^2 F^2} 
abla \widetilde{\mu}_i$ 

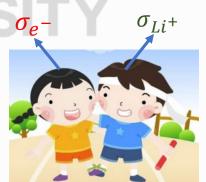
$$J_i = -\frac{\sigma_i}{z_i^2 F^2} \nabla \widetilde{\boldsymbol{\mu}}_i$$

We have reached a simple yet somewhat expected result:

"
$$\sigma_{Li}$$
" =  $\frac{\sigma_e - \sigma_{Li^+}}{\sigma_e - + \sigma_{Li^+}} = \frac{1}{1/\sigma_e - + 1/\sigma_{Li^+}}$ 

Chemical diffusivity is limited by the species that move *more slowly* 

If 
$$\sigma_e$$
 -  $\gg \sigma_{Li}$  +, then " $\sigma_{Li}$ "  $\approx \sigma_{Li}$  +





## リ 亜 湖 大 學 How to solve for chemical diffusivity?

#### Fick's first law

$$J_{diff, Li} = -D_{Li} \nabla c_{Li}$$

How to express  $D_{Li}$  as a function of  $D_{Li^+}$  and  $D_{e^-}$ ?

 $Li$ 
 $Li_x CoO_2$ 
electrode
 $e^ Li^+$ 

$$J_{Li} = -\frac{1}{F^2} \frac{\sigma_e - \sigma_{Li^+}}{\sigma_e - + \sigma_{Li^+}} (\nabla \tilde{\mu}_e - + \nabla \tilde{\mu}_{Li^+}) \qquad \nabla c_e - = \nabla c_{Li^+} = \nabla c_{Li}$$

$$\nabla c_{e^{-}} = \nabla c_{Li^{+}} = \nabla c_{Li}$$

$$= -\frac{RT}{F^2} \frac{\sigma_e - \sigma_{Li^+}}{\sigma_{e^-} + \sigma_{Li^+}} \left(\frac{1}{c_{e^-}} + \frac{1}{c_{Li^+}}\right) \nabla c_{Li} \quad \text{(Dilute limit)}$$

 $\delta$  means it describes the change of *non-stoichiometry* 

$$D_{Li}^{\delta} = \frac{RT}{F^2} \frac{\sigma_e - \sigma_{Li^+}}{\sigma_{e^-} + \sigma_{Li^+}} \left( \frac{1}{c_{e^-}} + \frac{1}{c_{Li^+}} \right)$$
 So-called "chemical diffusivity"
$$1/R^{\delta} \qquad 1/C^{\delta}$$
 For  $D^{\delta} = 10^{-10}$  cm<sup>2</sup>/s

For 
$$D^{\delta} = 10^{-10} \text{ cm}^2/\text{s}$$

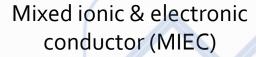
"Chemical "Chemical Resistance" Capacitance"

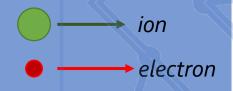
Relaxation time  $\tau^{\delta} = R^{\delta} C^{\delta} \propto \frac{L^2}{R^{\delta}}$ 

L	τ
10 mm	300 years
10 µm	2 hours
10 nm	0.01 S



## Job-sharing synergistic mass transport

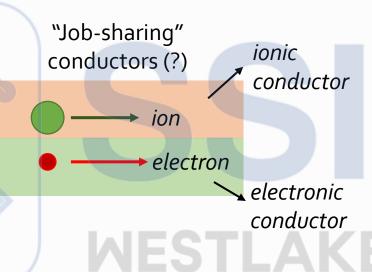




Chemical diffusion

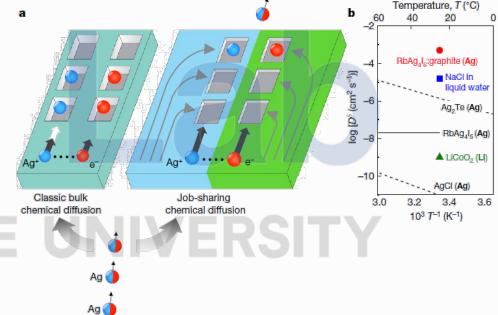
$$\frac{1}{R^{\delta}} = \frac{\sigma_{eon}\sigma_{ion}}{\sigma_{eon} + \sigma_{ion}}$$

$$D^{\delta} \propto \frac{1}{R^{\delta}}$$



$$\frac{1}{R^{\delta}} = ???$$

$$D^{\delta'} \gg D^{\delta}$$
???



Chia-Chin Chen, Lijun Fu & Joachim Maier, *Nature* (2016)



### Things we have discussed in this lecture

#### Type of diffusivities:

- What are the different types of diffusivities?
- What physical mechanism and concept does each diffusivity describe?

#### Chemical diffusivity:

- What physical process does the chemical diffusion describe?
- What are the key factors that govern the chemical diffusivity?

Goal of this lecture: you should be able to answer the questions above now (hopefully):)

