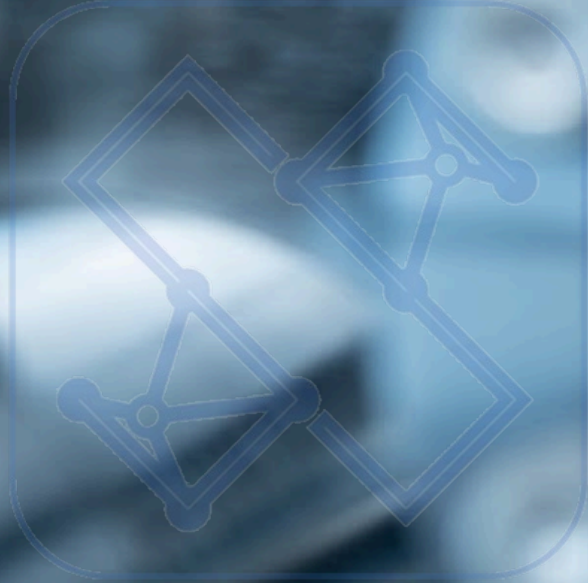


# Lecture 2:

## Electronic Defects

### The Brouwer Diagram



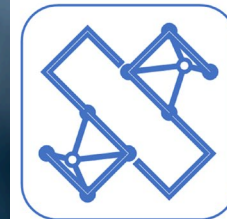
SSI Lab

WESTLAKE UNIVERSITY

Prof. Qiyang Lu

Solid State Ionics (SSI) Laboratory

School of Engineering, Westlake University



## Electronic defects:

- What is the difference between electronic and ionic defects?
- A brief review on basic semiconductor physics: how to write the charge neutrality equation by using Kröger-Vink notations?

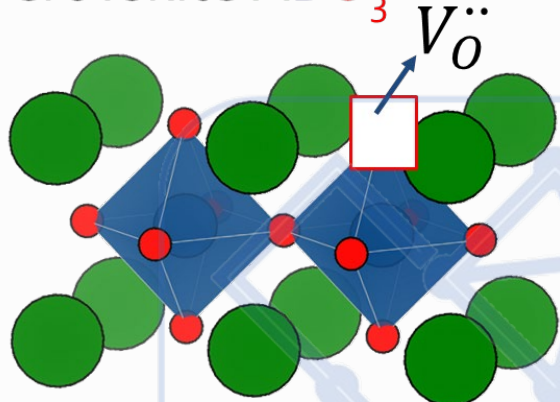
## Brouwer diagrams:

- What question are we trying to solve by using the Brouwer diagram?
- What is the key assumption of Brouwer diagram (so-called Brouwer condition)?
- How does a typical Brouwer diagram for metal oxides look like? How can it be used to predict/understand the conductivity of materials?

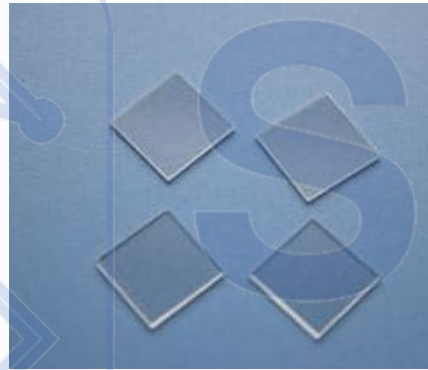
**Goal of this lecture:** you should be able to answer the questions above by the end of this lecture : )

# Ionic vs. Electronic defects

Perovskite  $ABO_3$



$SrTiO_{3-\delta}$  (STO)



Perovskite  $ABO_3$  structure; Cubic (ideal tolerance factor);  
("Si for perovskite oxides")

At RT (300 K):  $\mu_V = 1.43 \times 10^{-13} \text{ cm}^2 \text{ V}^{-1} \text{ s}^{-1}$  (*almost negligible*);  
 $\mu_n = 1.60 \text{ cm}^2 \text{ V}^{-1} \text{ s}^{-1}$ ;  $\mu_p = 1.27 \text{ cm}^2 \text{ V}^{-1} \text{ s}^{-1}$

At 500°C (773 K):  $\mu_V = 3.19 \times 10^{-5} \text{ cm}^2 \text{ V}^{-1} \text{ s}^{-1}$  (*still very low, but can sustain ion motion*);  
 $\mu_n = 0.20 \text{ cm}^2 \text{ V}^{-1} \text{ s}^{-1}$ ;  $\mu_p = 0.14 \text{ cm}^2 \text{ V}^{-1} \text{ s}^{-1}$

Compare **mobilities** of ionic and electronic defects

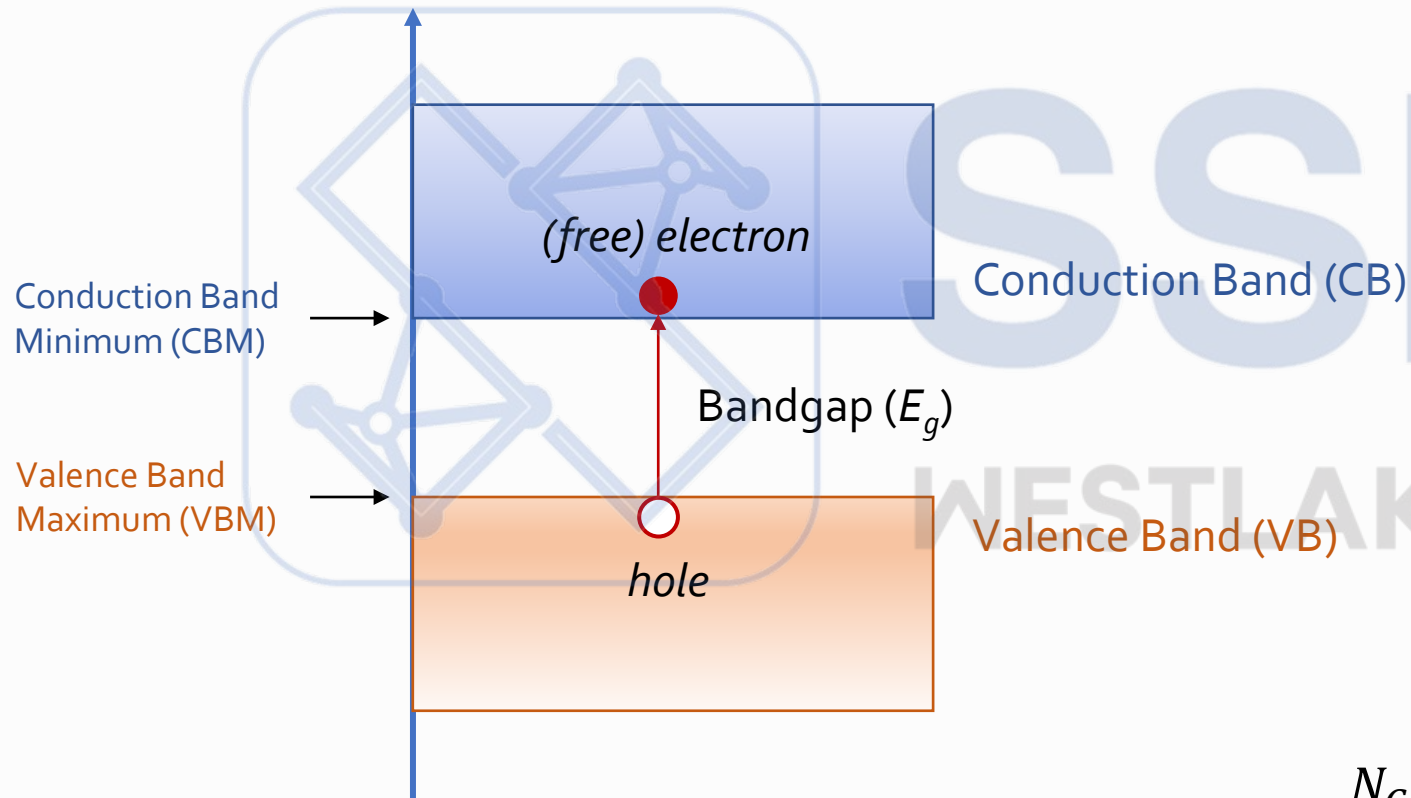
<b>Ionic</b> (oxygen vacancy)	$\mu_V = 1.0 \times 10^4 (T^{-1}/K^{-1})$ $\times \exp\left(-\frac{0.86 \text{ eV}}{k_B T}\right) \text{ cm}^2 \text{ V}^{-1} \text{ s}^{-1}$
<b>Electronic</b> (electrons & holes)	$\mu_n = 4.5 \times 10^5 (T/K)^{-2.2} \text{ cm}^2 \text{ V}^{-1} \text{ s}^{-1}$ $\mu_p = 8.9 \times 10^5 (T/K)^{-2.36} \text{ cm}^2 \text{ V}^{-1} \text{ s}^{-1}$

Guo, Fleig & Maier, *SSI*, 2002



# Review of semiconductor physics: intrinsic semiconductors

Electron Energy



"Band-to-band" excitation:

$$\text{null} \rightleftharpoons e' + h'$$

(free) electron      hole

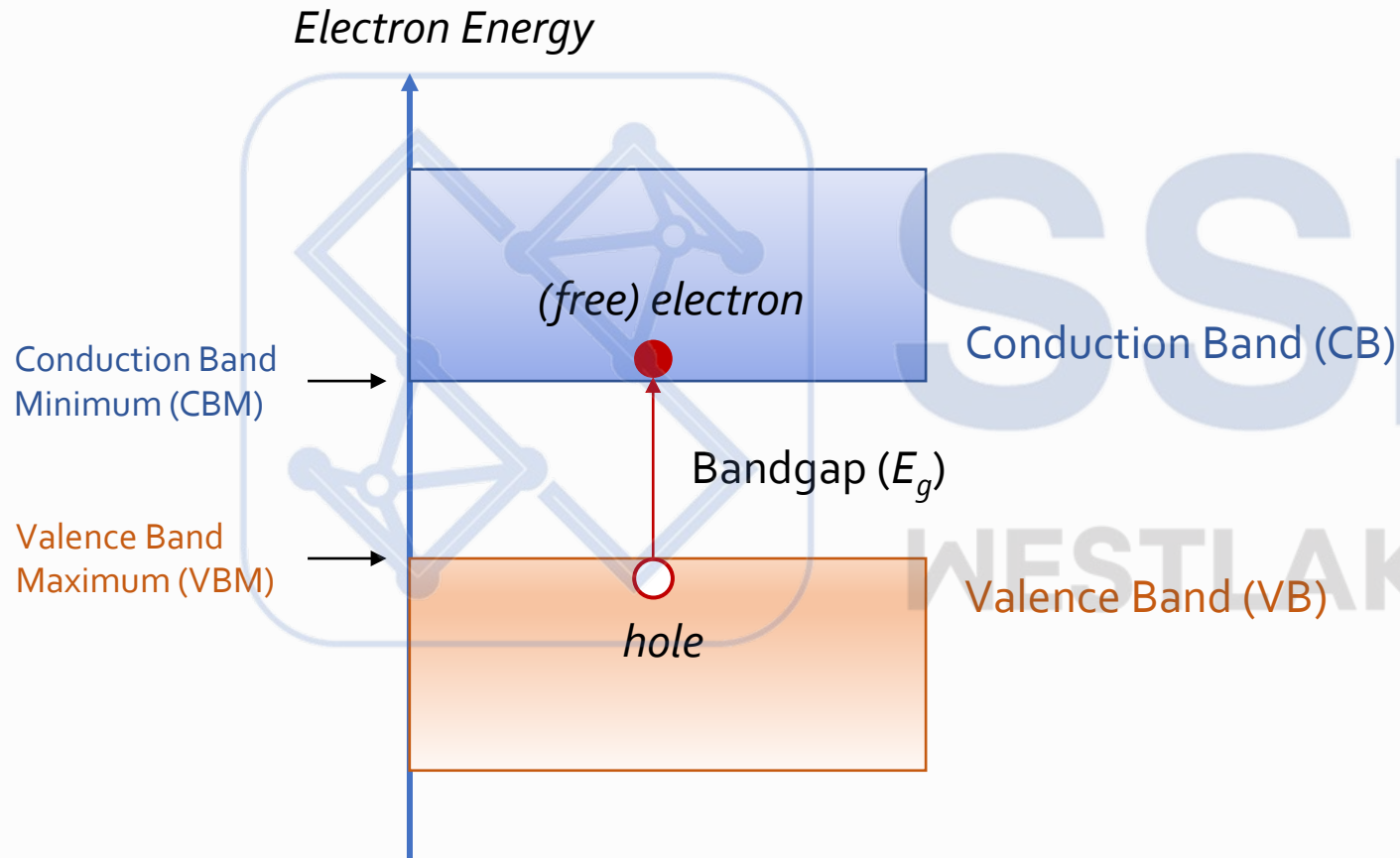
$$K_{el} = [e'] [h'] = n p$$

$$K_{el} = N_C N_V \exp\left(-\frac{E_g}{k_B T}\right)$$

$N_C$ : effective density of state (DOS) at CBM

$N_V$ : effective density of state (DOS) at VBM

# Review of semiconductor physics: intrinsic semiconductors, cont'd



"Band-to-band" excitation:

$$\text{null} \rightleftharpoons e' + h'$$

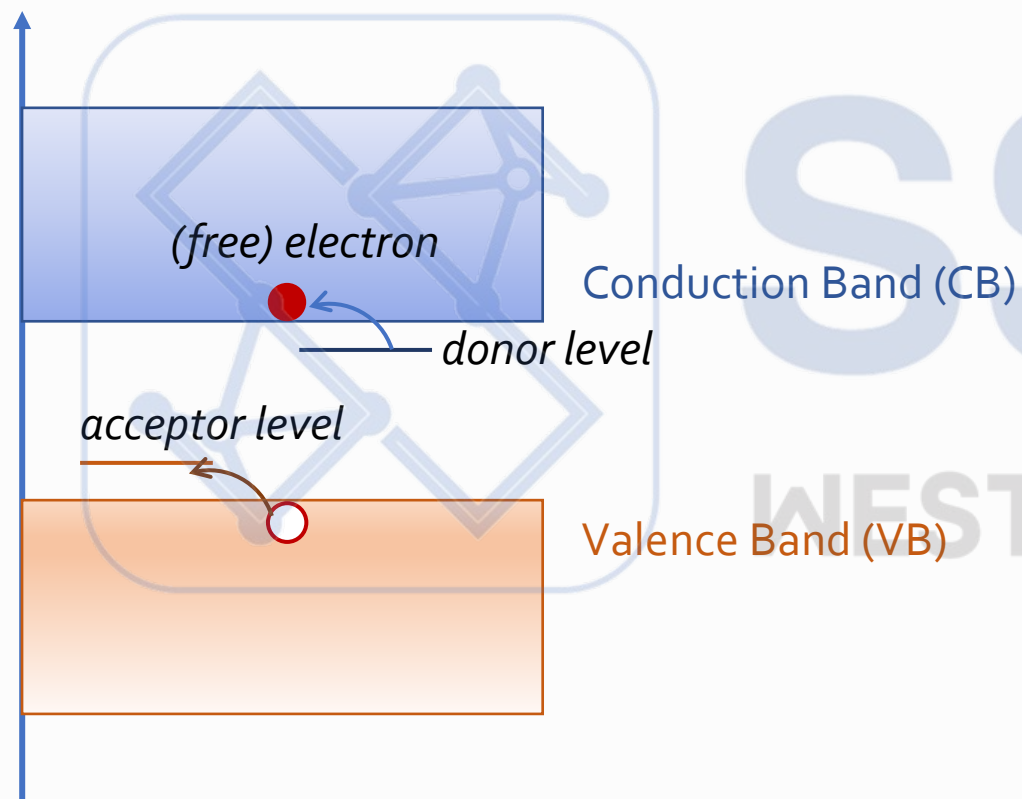
(free) electron      hole

$$K_{el} = [e'] [h'] = n p$$

Intrinsic semiconductors:

$$[e'] = [h'] = \sqrt{K_{el}} \propto \exp\left(-\frac{E_g}{2k_B T}\right)$$

Electron Energy



**Doping**—charge neutrality condition:

$$[A'] + [e'] = [D'] + [h']$$

$D'$ : donor, e.g.,  $P_{Si}$

$A'$ : acceptor, e.g.,  $B_{Si}$

*Intrinsic semiconductors:*

$$[e'] \approx [h'] \gg [A'] \ (\gg [D'])$$

*p-type semiconductors:*

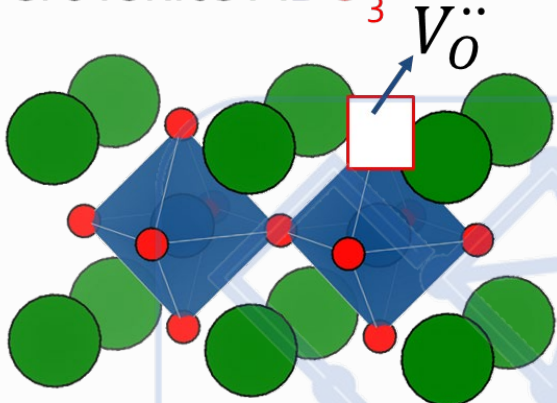
$$[A'] \approx [h'] \gg [e'] \ ([D'] \approx 0)$$

*n-type semiconductors:*

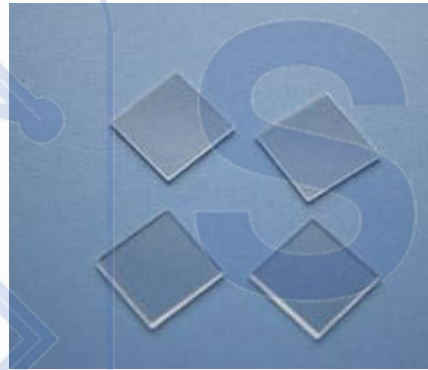
$$[D'] \approx [e'] \gg [h'] \ ([A'] \approx 0)$$

# Charge neutrality considering both ionic and electronic defects

Perovskite  $ABO_3$



$SrTiO_{3-\delta}$  (STO)



**Note:** even un-doped “pure”  $SrTiO_3$  single crystal contains *acceptor dopants* such as Mn, Al, Fe, Mg etc. (usually on the level of ~100 ppm, e.g.  $Al'_{Ti}$ ,  $Mg''_{Ti}$ )

Electronic defects:  $[e']$ ,  $[h\cdot]$

(Intrinsic) ionic defects:  $[V_O^{..}]$ ,  $[O_i^{..}]$  (uncommon),  $[Ti_i^{...}]$  (uncommon),  $[V_{Sr}^{..}]$ ,  $[Sr_{Ti}^{..}]$  (anti-site defects)

(Extrinsic) ionic defects:  $[Al'_{Ti}]$ ,  $[Mg''_{Ti}]$

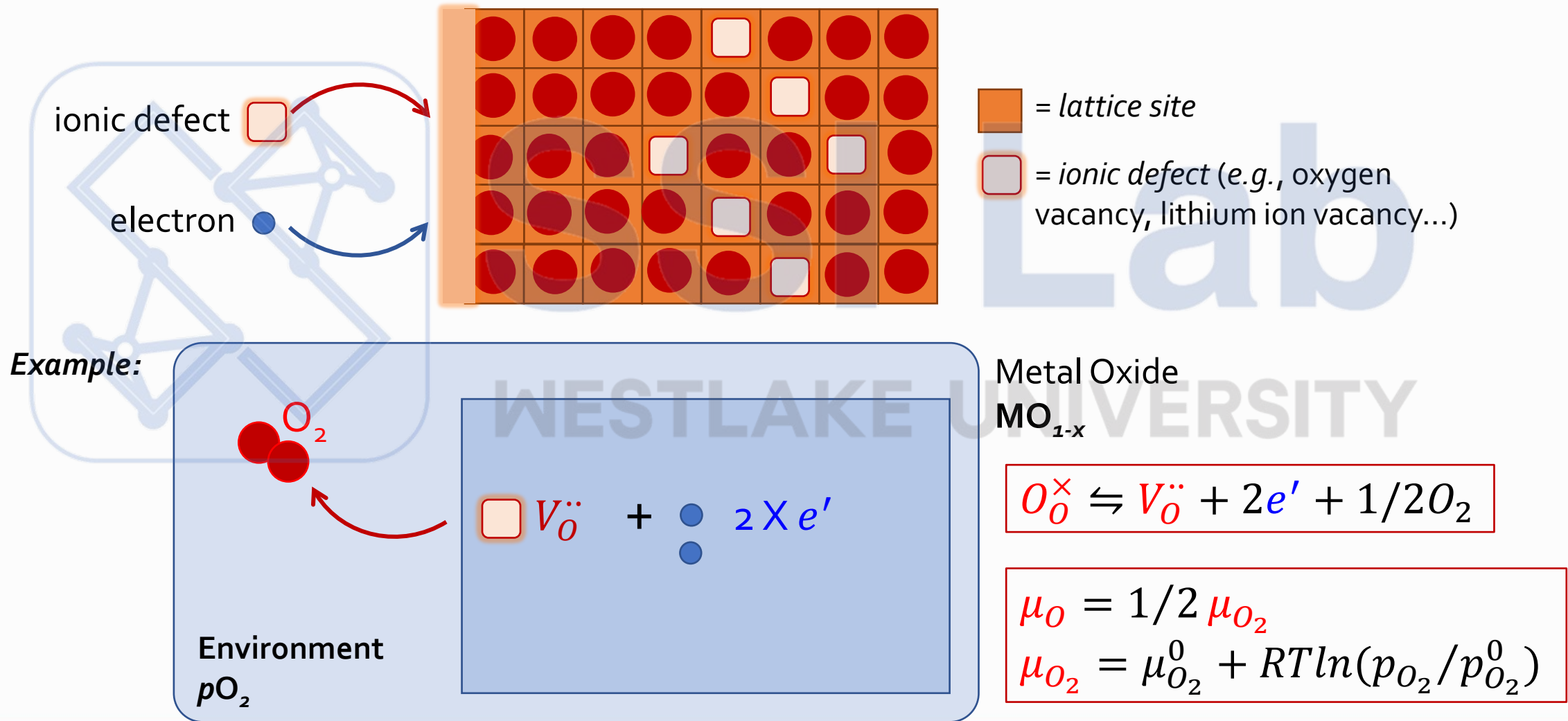
Charge neutrality considering both ionic and electronic defects:

$$[e'] + 2[V_{Sr}^{..}] + 2[Sr_{Ti}^{..}] + 2[O_i^{..}] = [h\cdot] + 2[V_O^{..}] + 4[Ti_i^{...}]$$

- Usually, it is very difficult to solve the equation w/ these many unknowns.
- The common practice is to ignore some defects with very low concentrations by dividing the chemical potential range into different regimes.

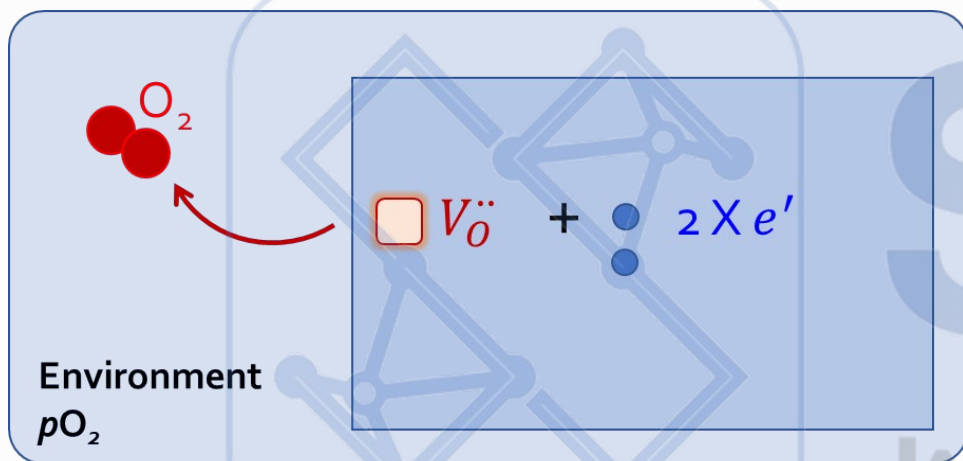
# How to tune defect concentration?

## Surfaces/Interfaces





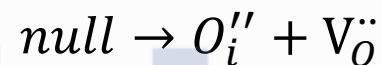
**Example:** Metal Oxide  $\text{MO}_{1-x}$



**Assumption:** metal cations (**M**) are immobile (usually true, since the migration of cations is much more difficult compared with anions)

## Defect reactions and mass action laws:

Frenkel pair



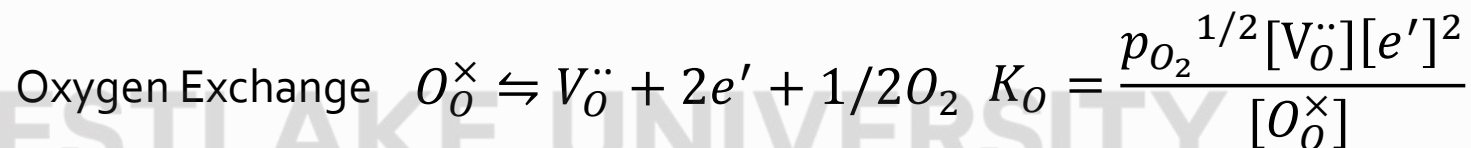
$$K_F = [\text{O}_i''][\text{V}_\text{O}]$$

"Band-to-band"



$$K_{el} = [e'][h']$$

Oxygen Exchange



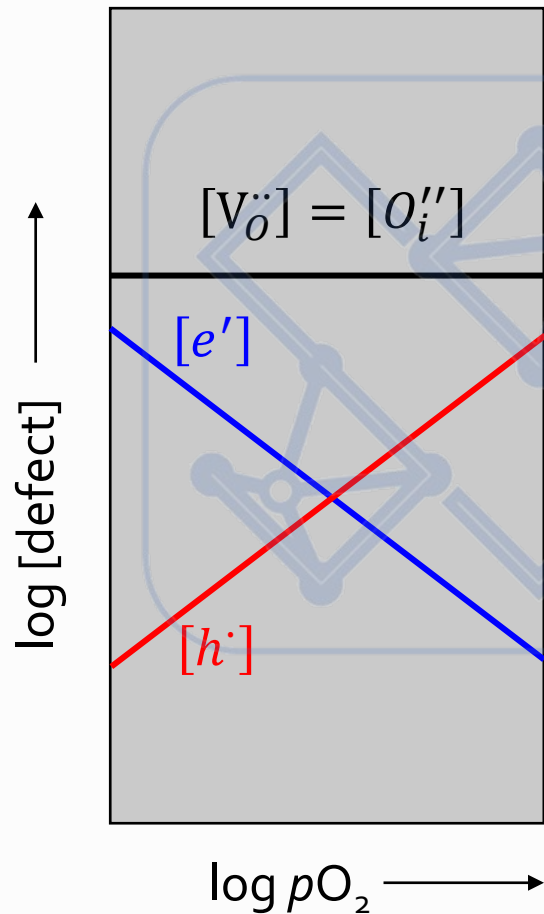
**Electroneutrality also applies**

$$[e'] + 2[\text{O}_i''] = [h'] + 2[\text{V}_\text{O}]$$

In principle, one can solve for four unknowns using these four equations, but it is much easier to **ignore the minority defects** with low enough concentration.

**Brouwer condition:** *only two majority defects with opposite charges matter.*

# The Brouwer diagram: the $\log[\ ] \sim \log pO_2$ plot



Regime I: "Intrinsic" regime

*The majority charge carriers are ionic defects*

$$[V_o] \gg [h^{\cdot}], [O_i''] \gg [e^{\cdot}]$$

Electroneutrality

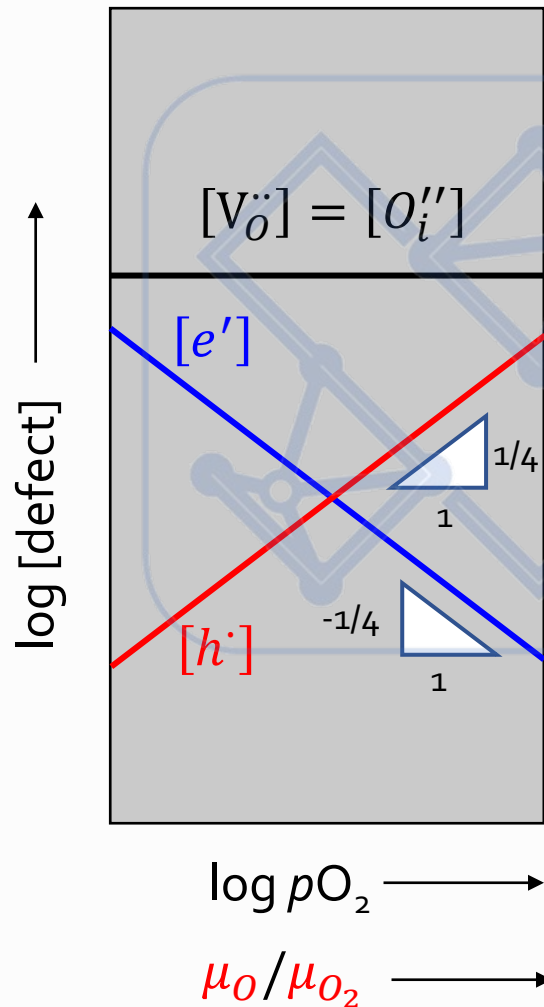
$$[e^{\cdot}] + 2[O_i''] = [h^{\cdot}] + 2[V_o]$$

$$\text{null} \rightarrow O_i'' + V_o, K_F = [O_i''] [V_o] \longrightarrow [O_i''] = [V_o] = \sqrt{K_F}$$

$$O_o^{\times} \rightleftharpoons V_o + 2e^{\cdot} + 1/2 O_2 \quad K_O = \frac{p_{O_2}^{1/2} [V_o] [e^{\cdot}]^2}{[O_o^{\times}]}$$

$$[e^{\cdot}] = \sqrt{\frac{K_O}{p_{O_2}^{1/2} [V_o]}} = p_{O_2}^{-1/4} K_O^{1/2} K_F^{-1/4} \propto p_{O_2}^{-1/4}$$

# Regime I: “Intrinsic” regime



## Regime I: “Intrinsic” regime

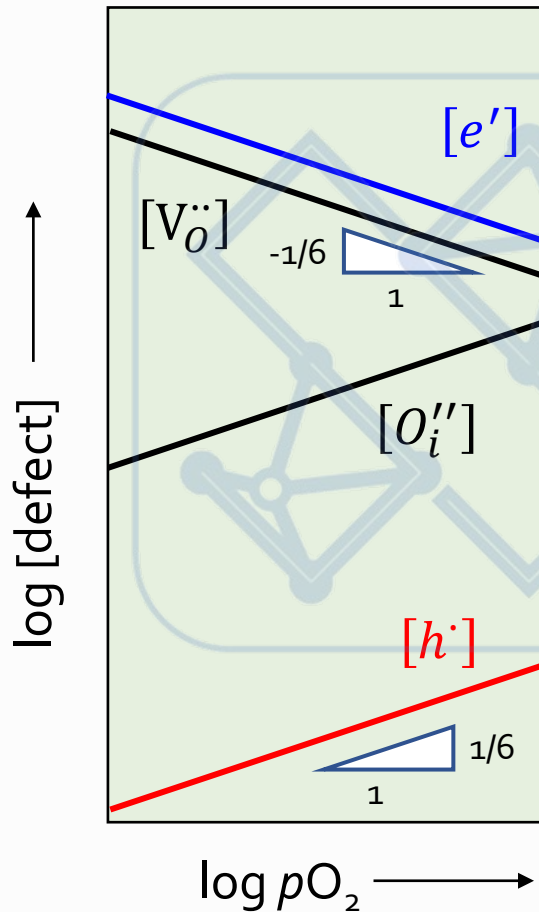
*The majority charge carriers are ionic defects*

$$[e'] = \sqrt{\frac{K_O}{p_{O_2}^{1/2} [V_{\ddot{O}}]}} = p_{O_2}^{-1/4} K_O^{1/2} K_F^{-1/4} \propto p_{O_2}^{-1/4}$$

$$\text{null} \rightarrow e' + h\cdot, K_{el} = [e'] [h\cdot] \longrightarrow [h\cdot] \propto p_{O_2}^{1/4}$$

### Notes:

- A “small” change of oxygen chemical potential can lead to **a large change in the concentration of electronic defects**;
- **n-type to p-type** transition can be triggered by changing pO<sub>2</sub>



## Regime II: "oxygen-poor" regime

The majority charge carriers are  $[V_{\ddot{O}}]$  &  $[e']$

$$[V_{\ddot{O}}] \gg [h^{\cdot}], [e'] \gg [O_i'']$$

Electroneutrality

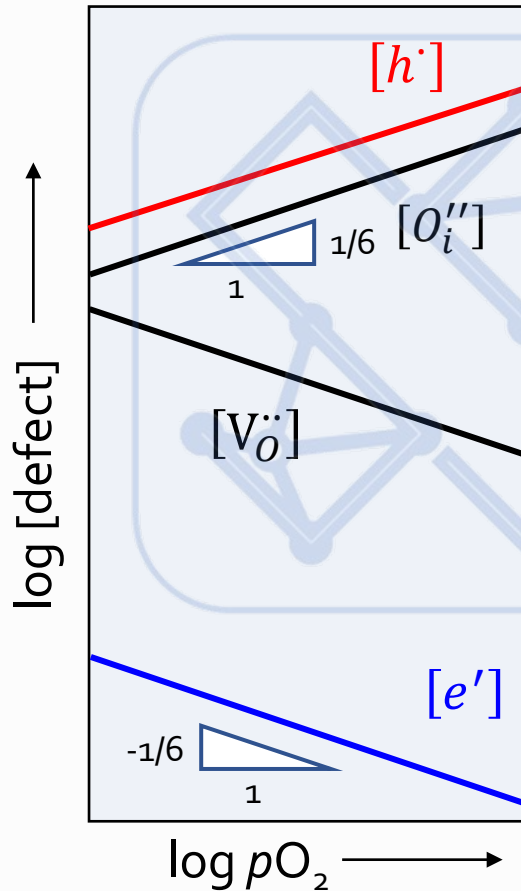
$$[e'] + 2[O_i''] = [h^{\cdot}] + 2[V_{\ddot{O}}]$$

$$K_O = \frac{p_{O_2}^{1/2} [V_{\ddot{O}}] [e']^2}{[O_O^{\times}]} \Rightarrow [V_{\ddot{O}}] \propto p_{O_2}^{-1/6}, [e'] \propto p_{O_2}^{-1/6}$$

$$K_{el} = [e'] [h^{\cdot}] \Rightarrow [h^{\cdot}] \propto p_{O_2}^{1/6}$$

$$K_F = [O_i''] [V_{\ddot{O}}] \Rightarrow [O_i''] \propto p_{O_2}^{1/6}$$





## Regime III: "oxygen-rich" regime

The majority charge carriers are  $[O_i'']$  &  $[h\cdot]$

$$[h\cdot] \gg [V_{\ddot{O}}], [O_i''] \gg [e']$$

Electroneutrality

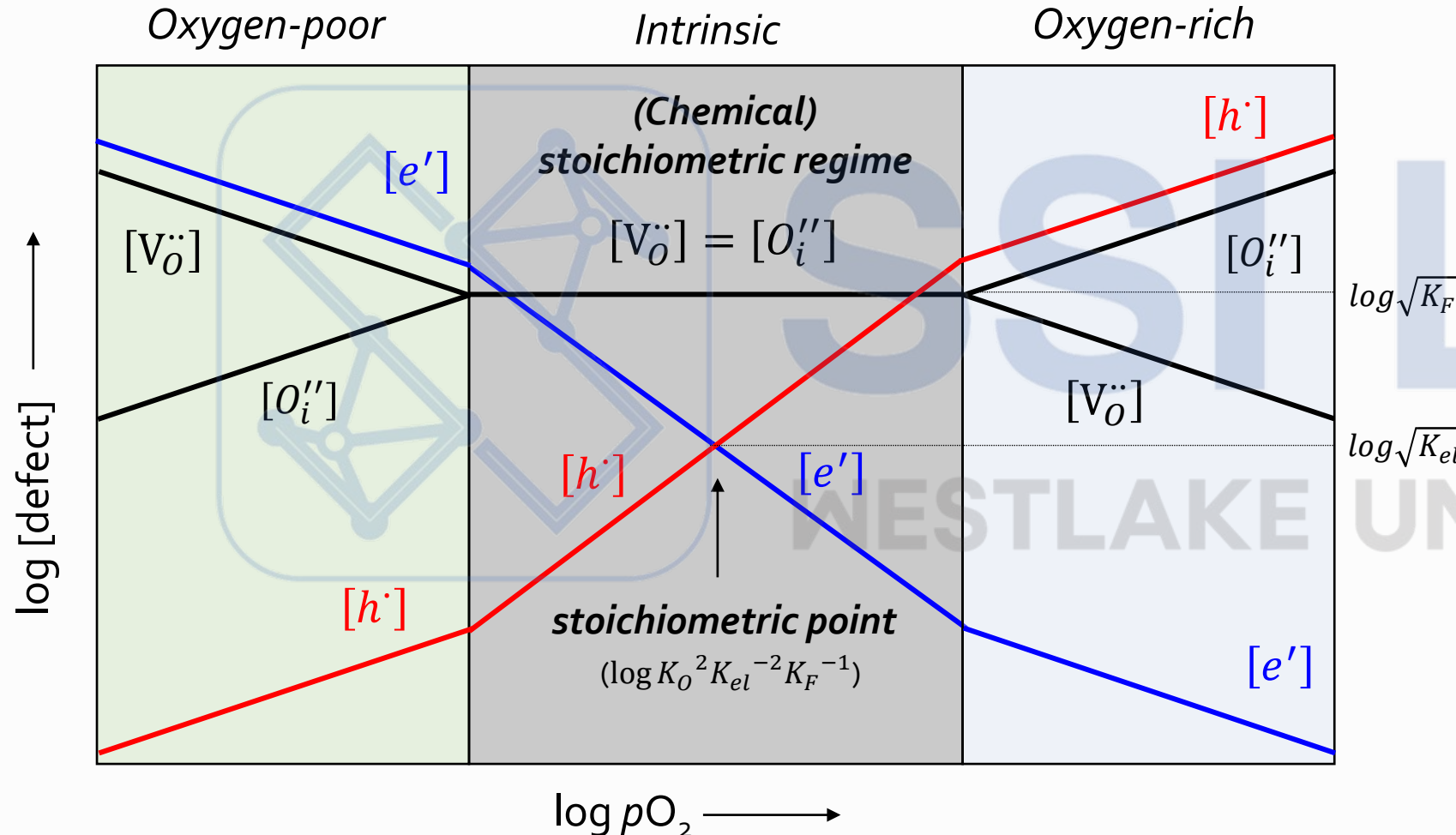
$$[e'] + 2[O_i''] = [h\cdot] + 2[V_{\ddot{O}}]$$

It is more convenient to rewrite the oxygen incorporation reaction in this case:

$$\boxed{\frac{1}{2} O_2 \rightleftharpoons O_i'' + 2h\cdot} \quad K_O = \frac{[O_i''] [h\cdot]^2}{p_{O_2}^{1/2}} \Rightarrow [O_i''] \propto p_{O_2}^{1/6} \quad [h\cdot] \propto p_{O_2}^{1/6}$$

$$K_{el} = [e'] [h\cdot] \Rightarrow [e'] \propto p_{O_2}^{-1/6} \quad K_F = [O_i''] [V_{\ddot{O}}] \Rightarrow [V_{\ddot{O}}] \propto p_{O_2}^{-1/6}$$

# Combine the three regimes together: a prototypical Brouwer diagram of metal oxides



## Note:

- We only considered intrinsic Frenkel pairs  $\rightarrow$  can be extended to **extrinsic ionic defects** or **Schottky pairs** ( $V_{\text{O}} + V_M''$ );
- The x-axis can be expressed by either  $\log pO_2$  or chemical potential of oxygen gas/neutral oxygen;
- The existence of a **stoichiometric point**.

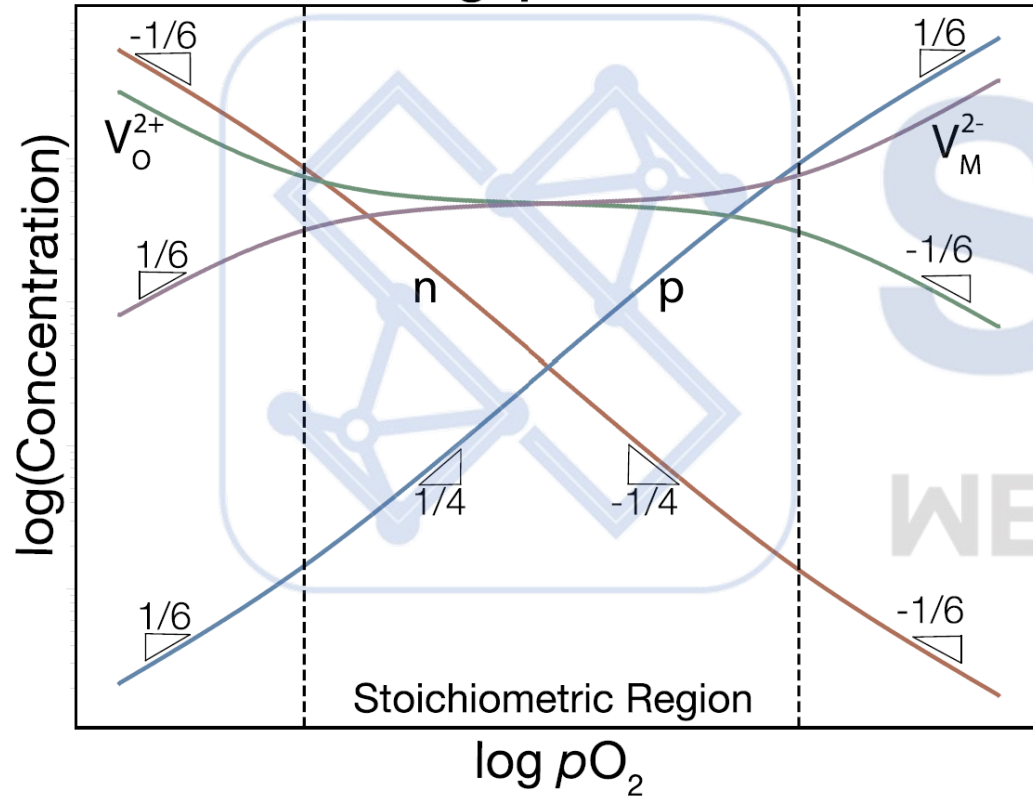
Metal Oxide  $MO_{1+\delta}$

$$\delta = [O_i''] - [V_{\text{O}}]$$

# More “realistic” Brouwer diagram w/ Schottky pairs or extrinsic ionic defects

$$K_{el} \ll K_S (E_g > \Delta G_{Sch}^0)$$

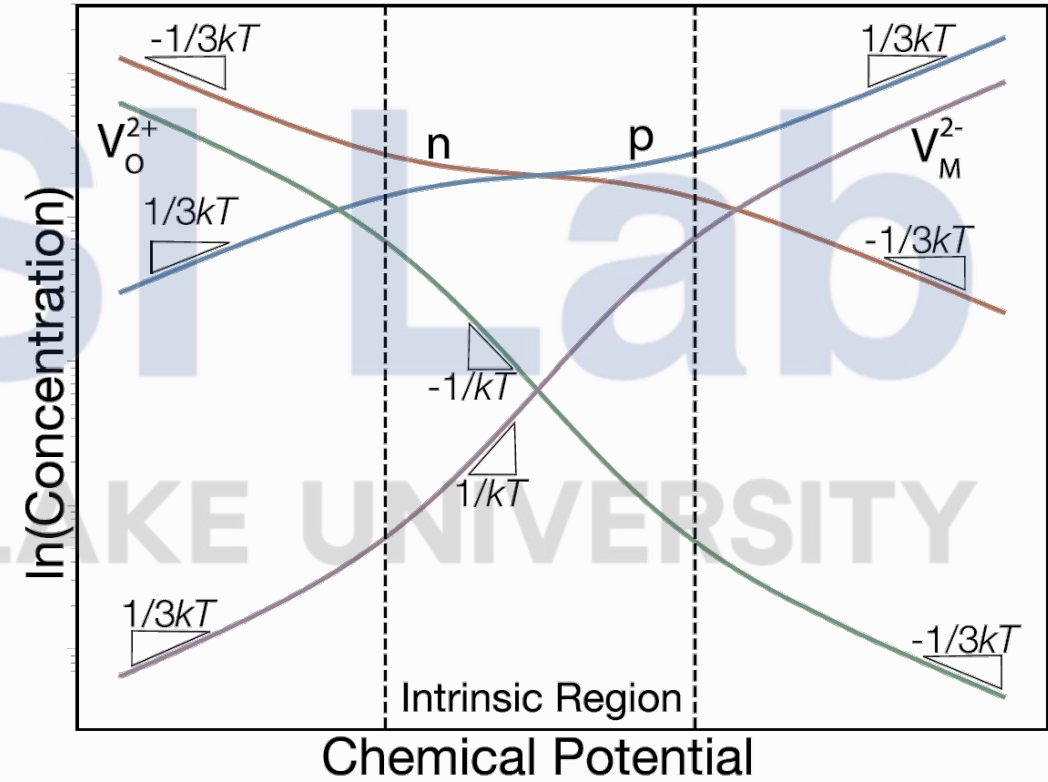
## Wide Band-gap Ion Conductor



Anand, Haile, Snyder et al., *Acc. Mater. Res.*, 2022

$$K_{el} \gg K_S (E_g < \Delta G_{Sch}^0)$$

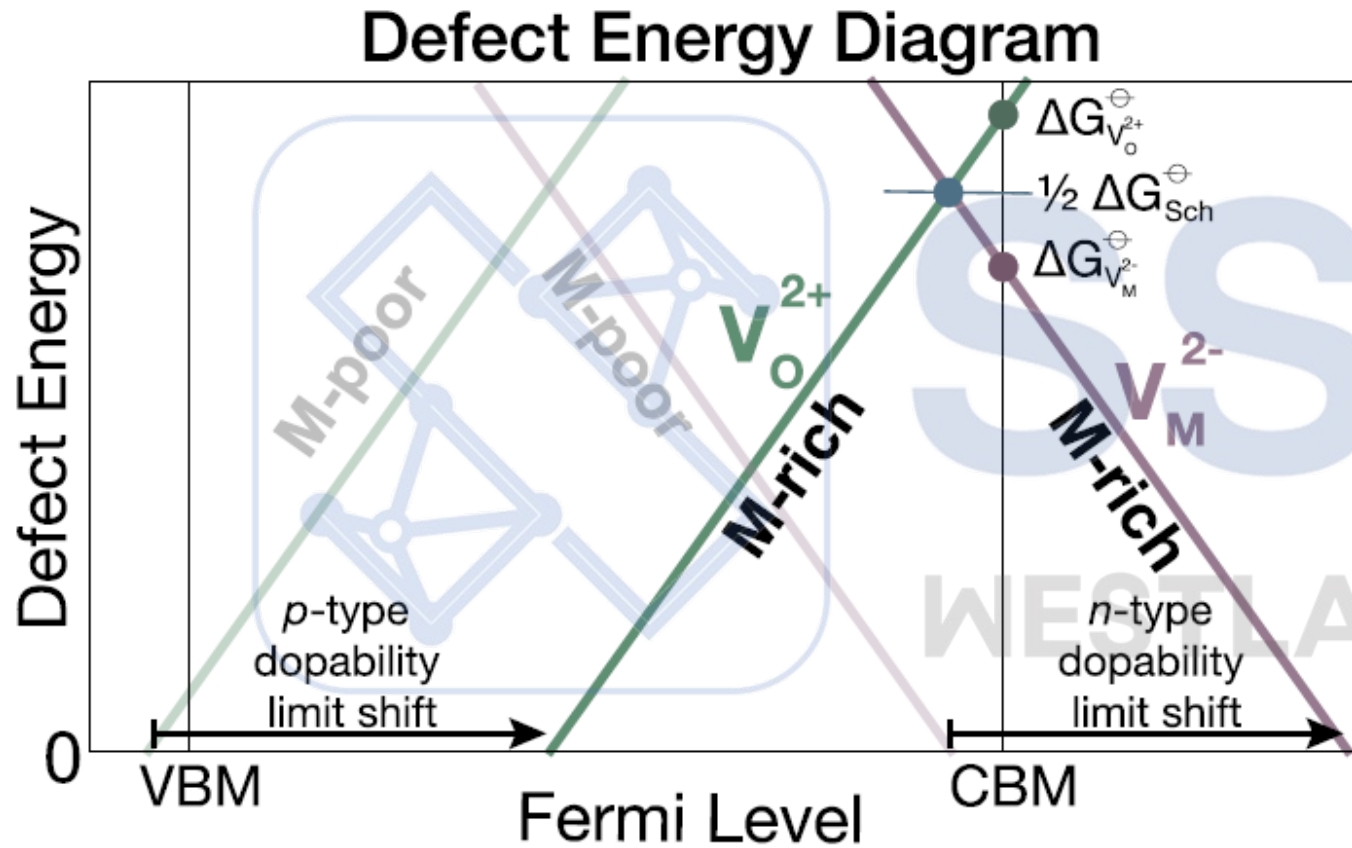
## Small Band-gap Semiconductor



$$\mu_O = 1/2 \mu_{O_2}$$

$$\mu_{O_2} = \mu_{O_2}^0 + RT \ln(pO_2/pO_2^0)$$

# Another way of expressing formation of defects



Anand, Haile, Snyder *et al.*, *Acc. Mater. Res.*, 2022

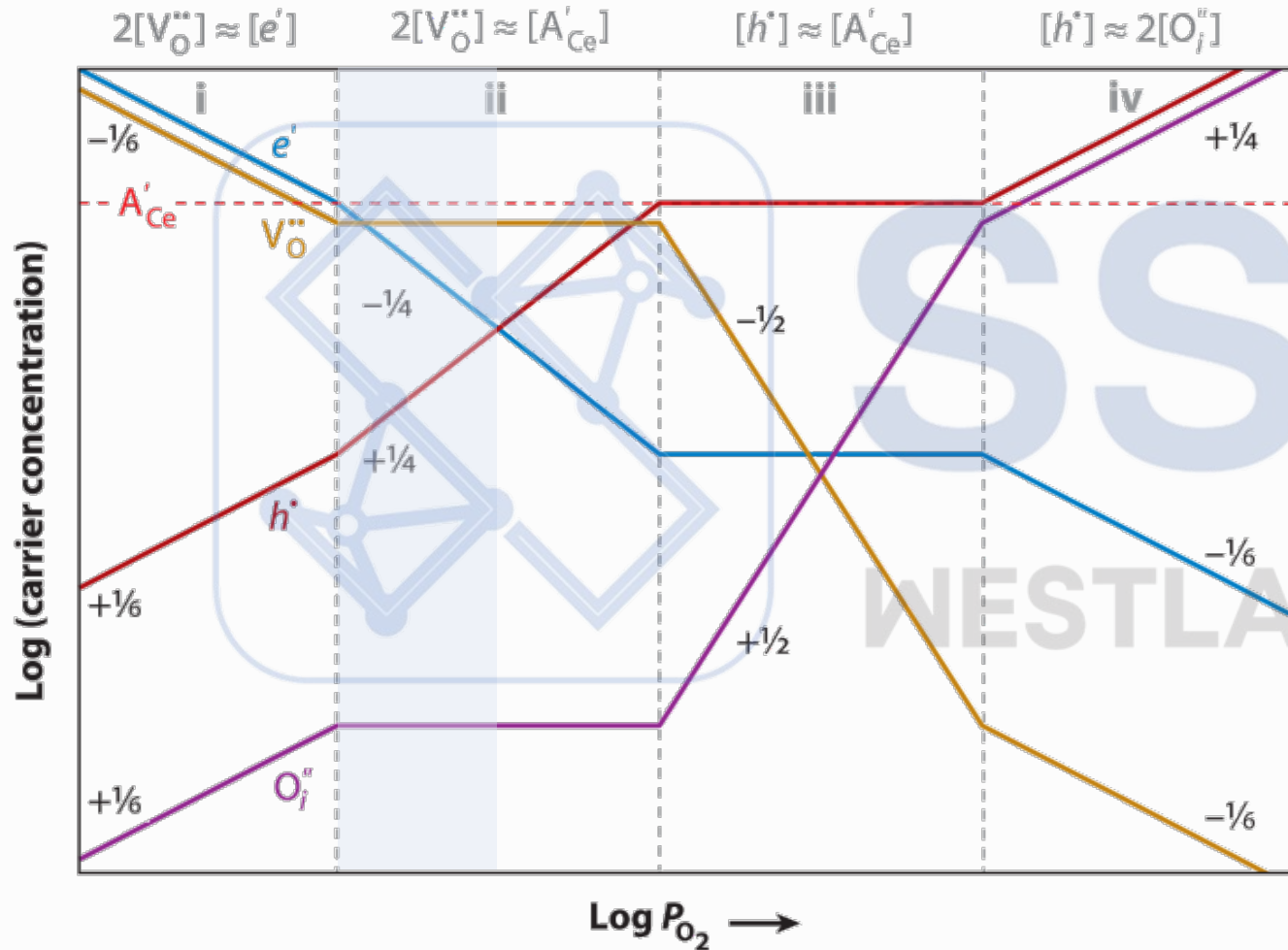
$$\Delta E_d = E_{def} - E_{perfect} + qE_F - \sum_i \Delta N_i \mu_i$$

Fermi level

- Very commonly used by semiconductor physicists, especially those who do DFT calculations, since  $E_F$  can be used as a tuning parameter;
- Can be used to **construct/convert to Brouwer diagrams**.



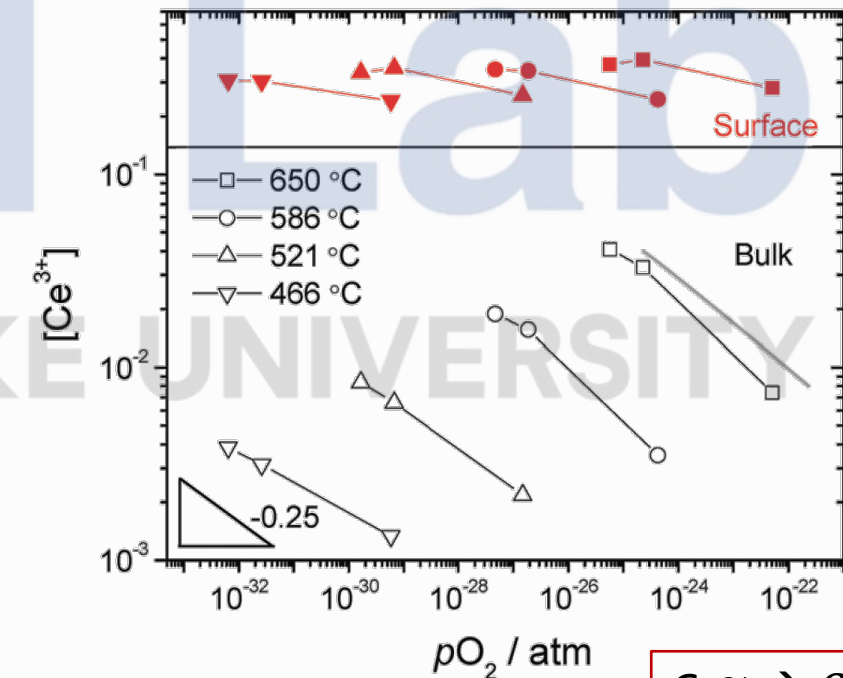
# More “realistic” Brouwer diagram w/ Schottky pairs or extrinsic ionic defects



Acceptor-doped  $CeO_{2-x-\delta}$

e.g.,  $Sm_{0.2}Ce_{0.8}O_{1.9-\delta}$

$Sm^{3+}$  replacing  $Ce^{4+} \rightarrow Sm'_{Ce}$



Chueh et al., *Chem. Mater.*, 2012

$Ce^{3+} \rightarrow Ce'_{Ce} \rightarrow e'$   
"small polaron"

- Direct measurement of defect concentration can be rather difficult;
- The most common practice is to measure the **conductivity** of materials.

Expression of **conductivity**:

$$\sigma = n \cdot ze \cdot M = cN_A \cdot ze \cdot M = czFM$$

$\sigma$  : Conductivity

$c$  : Charge carrier concentration;  $N_A$ : Avogadro constant;  $F$ : Faraday constant

$ze$  : Electric charge, for  $\text{Li}^+$ ,  $z = 1$

$M$ : Mobility (usually denoted as  $\mu$ , but not to be confused with chem. potential)

If we assume that the mobility stays constant within a range of chemical potential, then:

$$\sigma \propto c_{def}$$

Usually, what we measure is the **total** conductivity  $\sigma_{tot}$ :

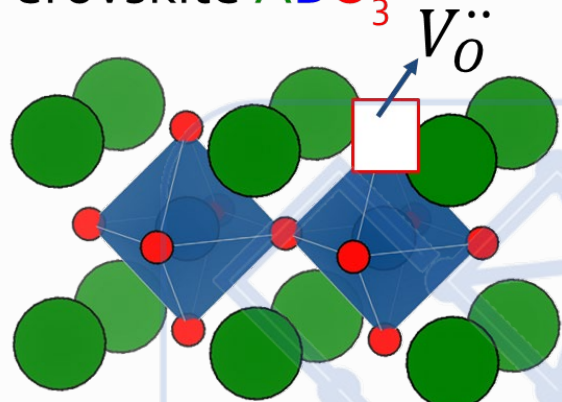
$$\sigma_{tot} = \sigma_{el} + \sigma_{ion}$$

$$\sigma_{el} = n_e e M_e + n_p e M_h$$

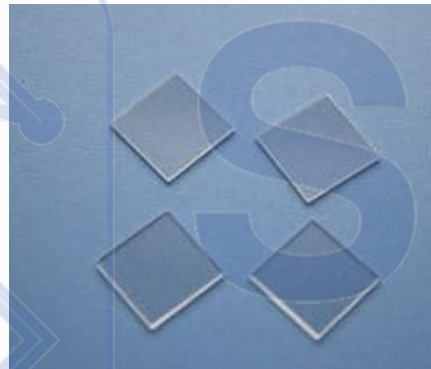
$$\sigma_{ion} = n_{ion} z_{ion} e M_{ion}$$

# SrTiO<sub>3</sub> serves as a perfect example to show the $\log \sigma \sim \log pO_2$ plot

Perovskite ABO<sub>3</sub>

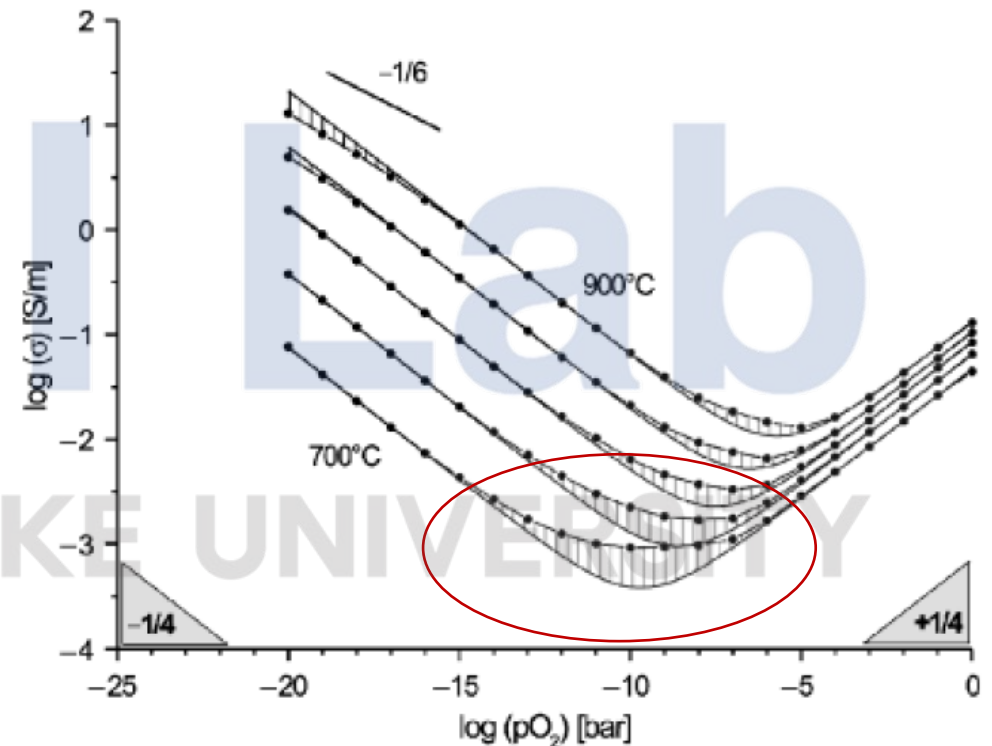


SrTiO<sub>3-δ</sub> (STO)



Perovskite ABO<sub>3</sub> structure; Cubic (ideal tolerance factor);  
("Si for perovskite oxides")

$\log \sigma_{tot} \sim \log pO_2$  plot

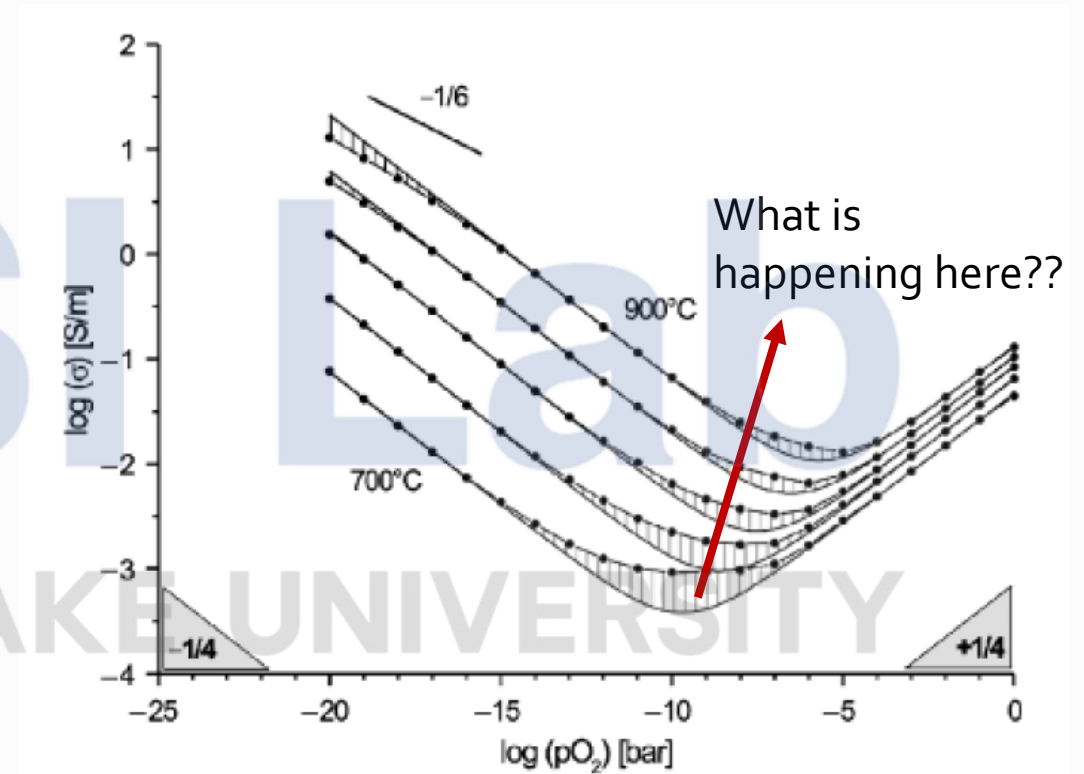
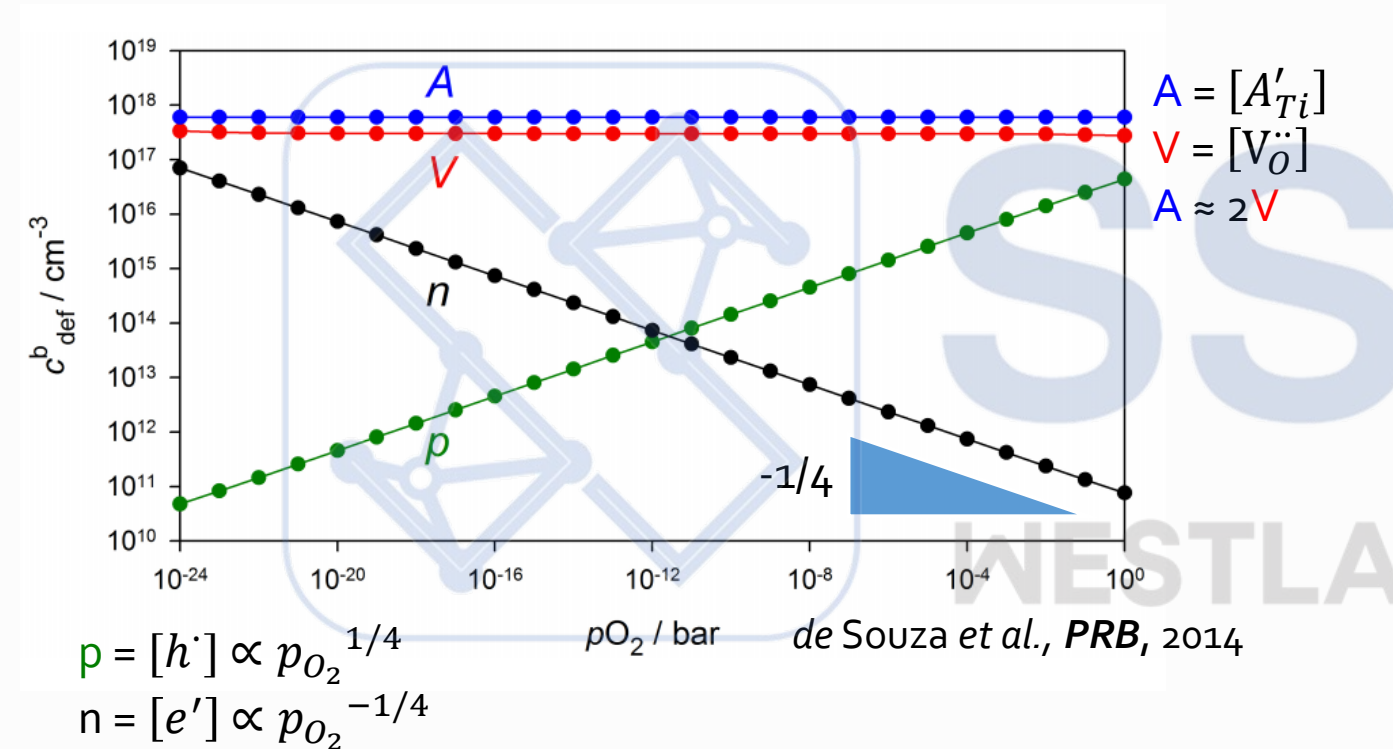


Ohly et al., J. Am. Ceram. Soc., 2006

**We want to understand:**

- What do the different slopes tell us about ionic and electronic defect equilibrium?
- What does the shaded area mean?

# Ionic and electronic conductivity of SrTiO<sub>3</sub>



Ohly et al., *J. Am. Ceram. Soc.*, 2006

$$\sigma_{tot} = ne\mu_e + pe\mu_h + V(Z_V e)\mu_V$$

At ~973 K for SrTiO<sub>3</sub>:

$\mu_e, \mu_h \sim 0.1 \text{ cm}^2/(\text{V}\cdot\text{s})$  vs.  $\mu_V \sim 4 \times 10^{-4} \text{ cm}^2/(\text{V}\cdot\text{s})$



# Brouwer diagram for lithium storage materials

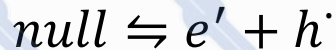
For a lithium-storage material **LiX** that can tolerate both **Li-excess** and **Li-deficiency**, we consider four point defects:  $Li'_i$ ,  $V_{Li}$ ,  $e'$ ,  $h\cdot$

Frenkel pair



$$K_F = [Li'_i][V_{Li}]$$

“Band-to-band”



$$K_{el} = [e'][h\cdot]$$

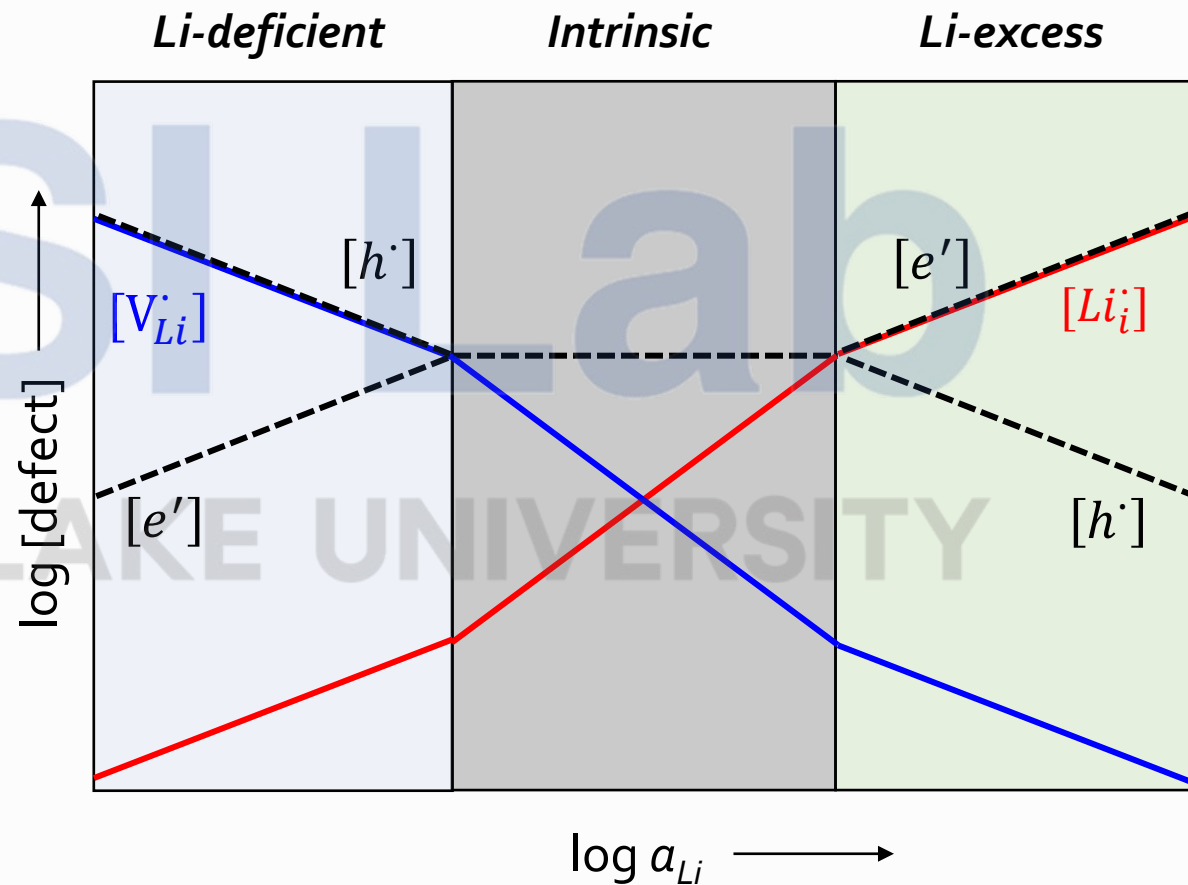
Li intercalation



$$K_{Li} = \frac{[Li_i][e']}{a_{Li}}$$

$$\mu_{Li} = \mu_{Li}^0 + RT \ln a_{Li}$$

We assume  $K_{el} \gg K_F$  (small bandgap)

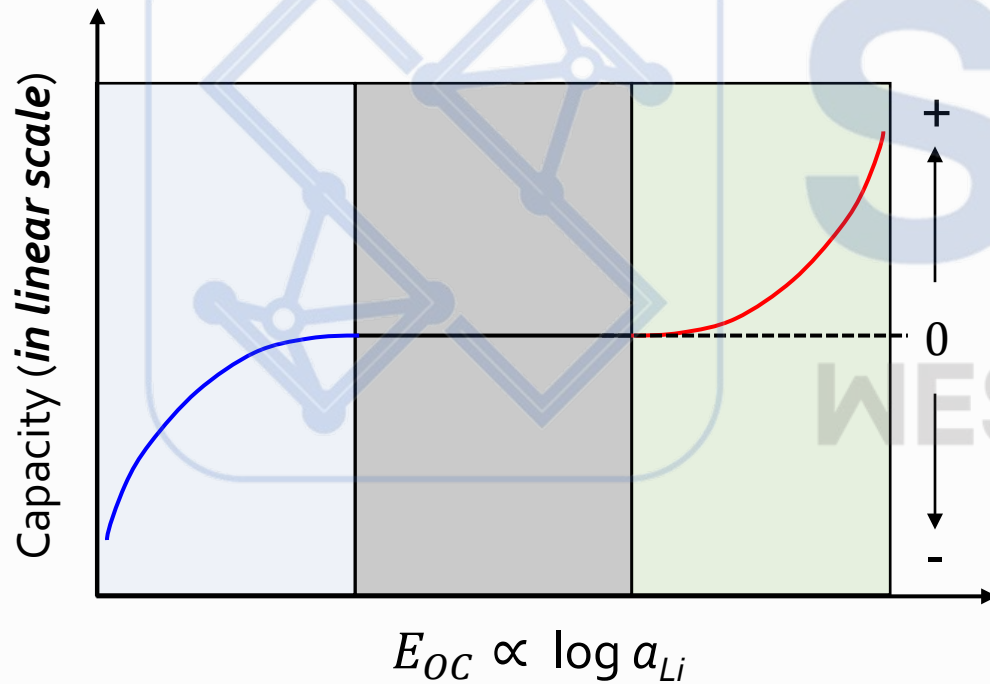


# What can we get from such a Brouwer diagram?

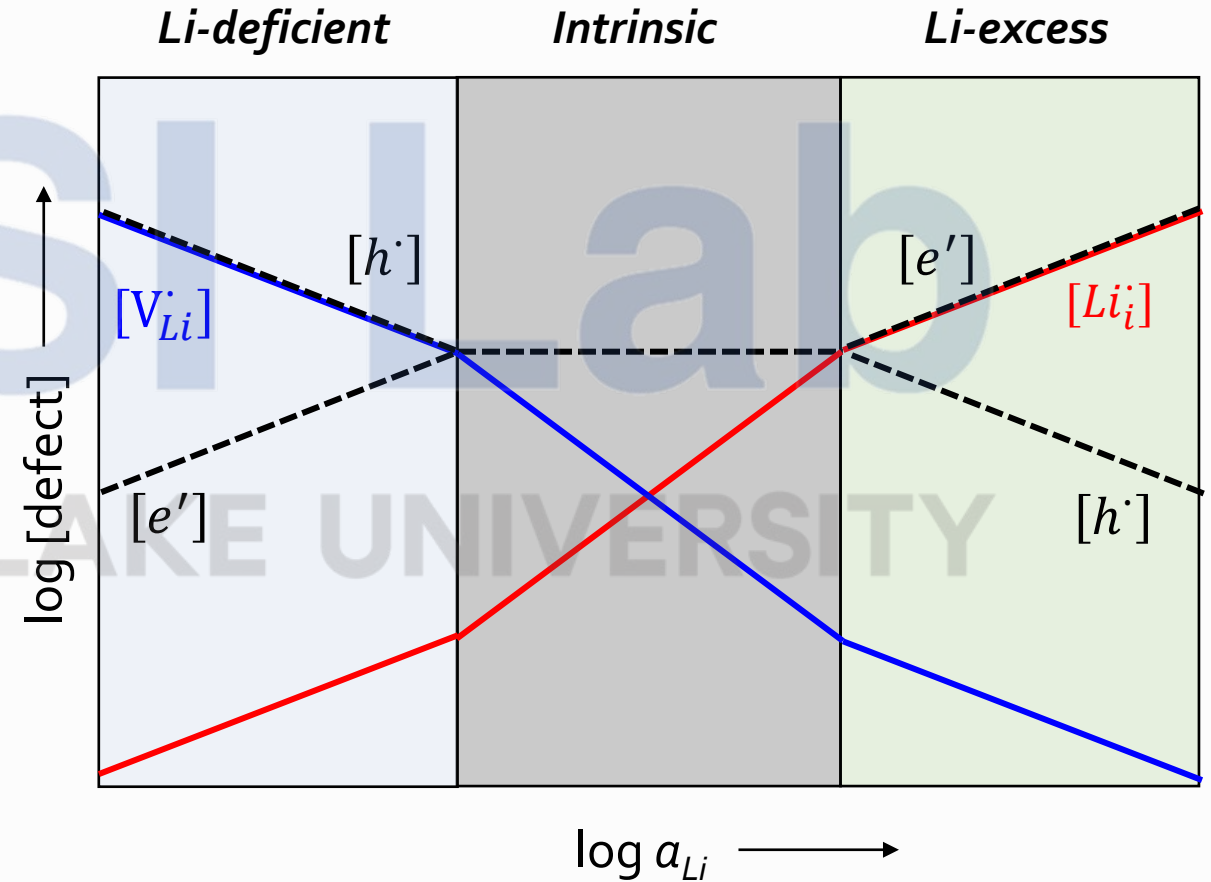
For Li-ion battery materials, we have:

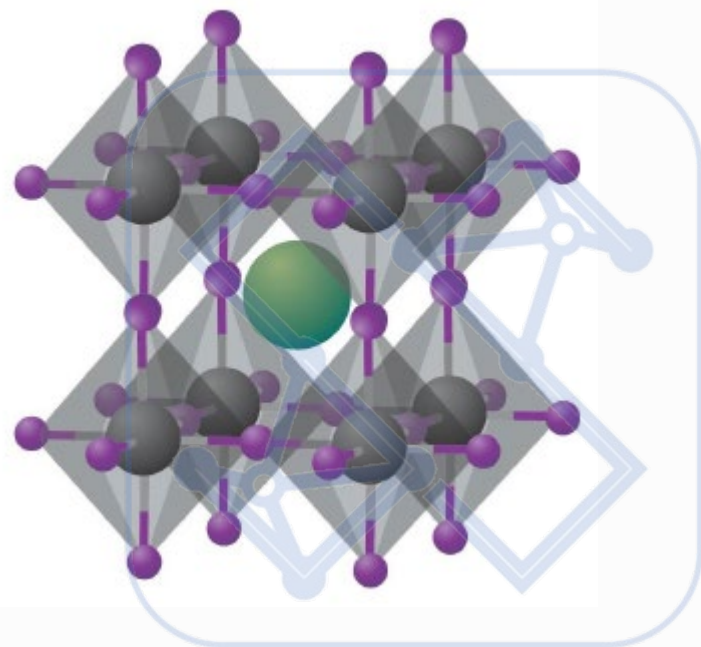
Charge stored:  $Q \propto |[Li_i] - [V_{Li}]|$

Open-circuit potential:  $E_{OC} \propto \mu_{Li} \propto \ln a_{Li}$



*Simulated battery charge/discharge curves (!)*





e.g. Methylammonium lead iodide  
 $\text{CH}_3\text{NH}_3\text{PbI}_3$  ( $\text{MAPbI}_3$ )

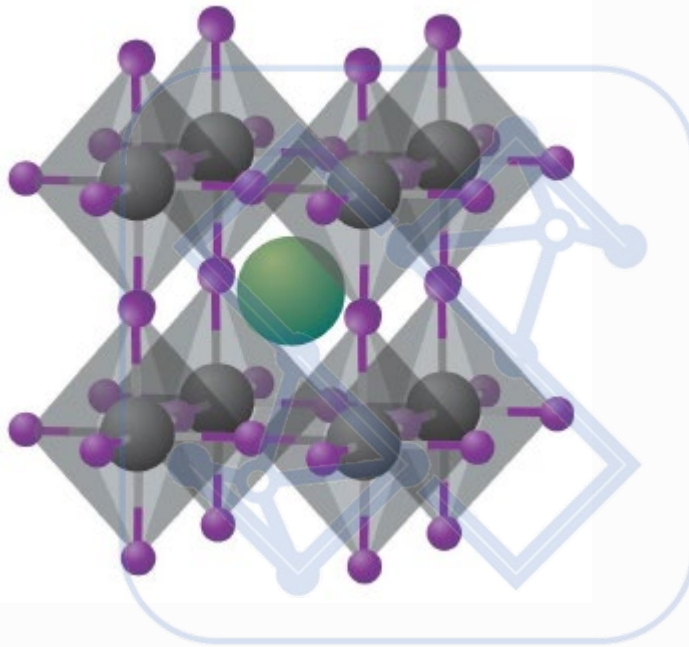
## Why do halide perovskites attract so much attention?

- Wide range in composition design: from inorganic compound (e.g.  $\text{CsPbI}_3$ ) to organo-metallic compound (e.g.  $\text{MAPbI}_3$ );
- High light absorption coefficient + very long diffusion length of electrons and holes (i.e., very low charge recombination rate)

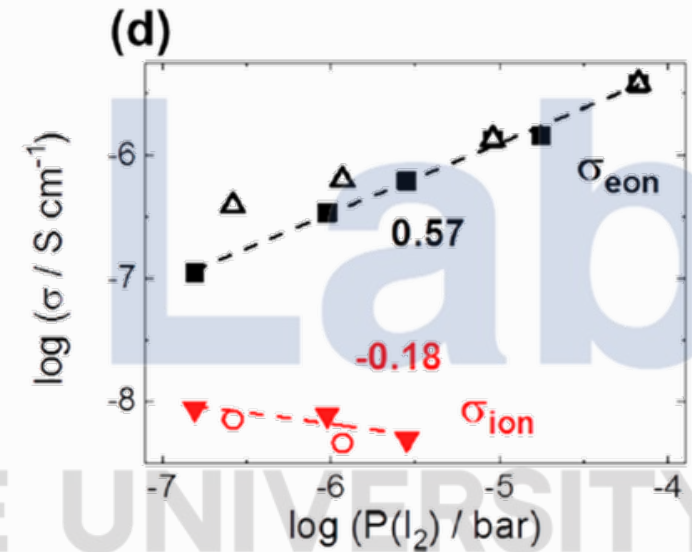
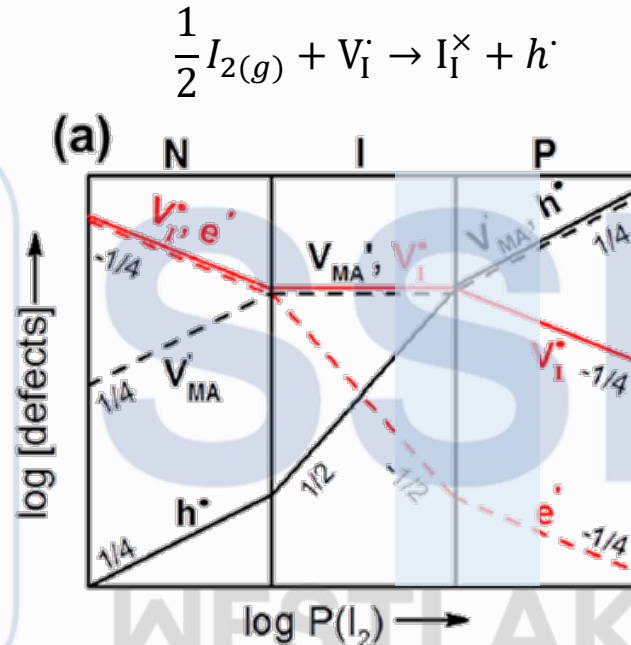
Since halide perovskites are called “*ionic semiconductors*”, we should be able to understand from the perspective of defect chemistry:

- Halide perovskites are ***mixed ionic-electronic conductors***. Therefore, it is important to differentiate ionic and electronic conductivities;
- Peculiar ion transport behavior under illumination. Ionic conductivity is shown higher under light compared with dark condition.

# Brouwer diagrams for halide perovskites: *Intrinsic*



Brouwer diagram of  $\text{MAPbI}_3$   
Measure **ionic** ( $\sigma_{\text{ion}}$ ) and electronic ( $\sigma_{\text{eon}}$ ) conductivity as a function of  $pI_2$



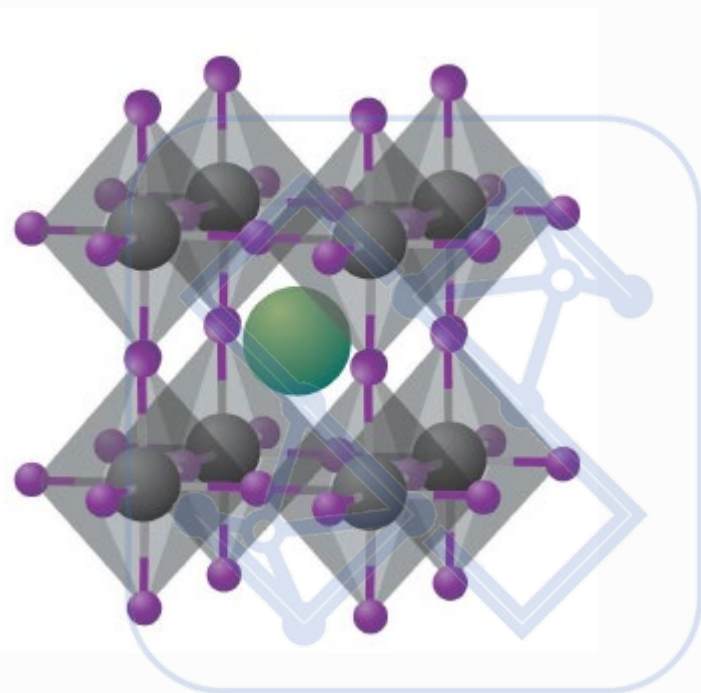
Senocrate and Maier, *J. Am. Chem. Soc.*, 2019

## Conclusions:

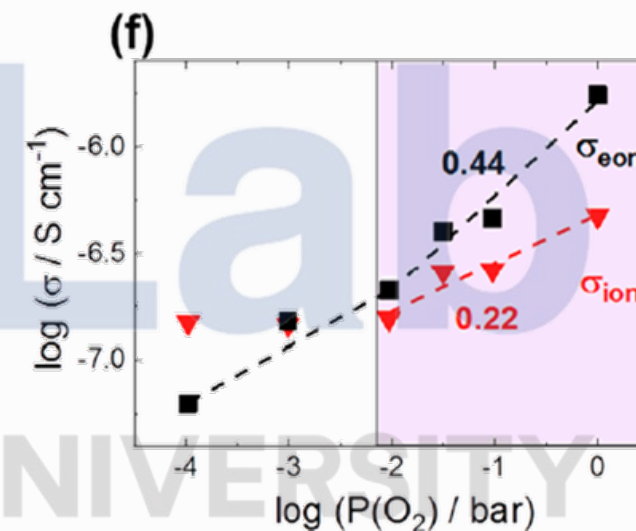
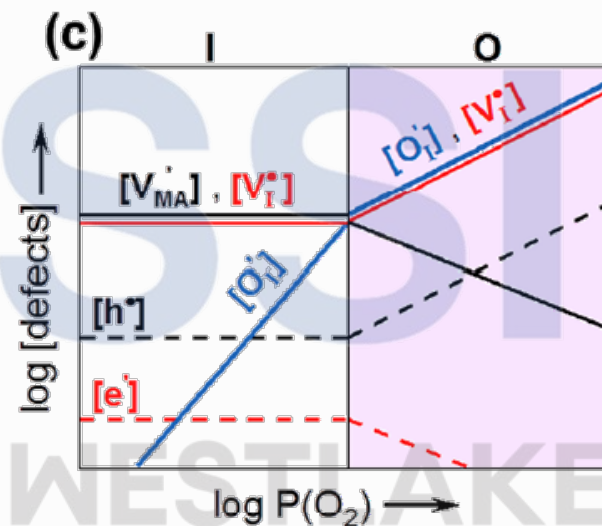
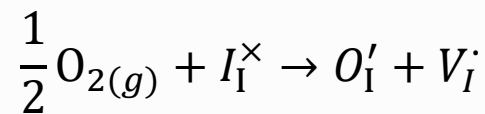
1. The tested  $pI_2$  range is within the **I-P** transition region in the Brouwer diagram;  
(The measured slope has some uncertainties)
2. The dominate ionic mobile defects are the **iodine vacancies**



# Brouwer diagrams for halide perovskites: *Intrinsic*



Brouwer diagram of  $\text{MAPbI}_3$   
Measure **ionic** ( $\sigma_{\text{ion}}$ ) and electronic ( $\sigma_{\text{eon}}$ ) conductivity as a function of  $p\text{O}_2$



Senocrate and Maier, *J. Am. Chem. Soc.*, 2019

## Conclusion:

Incorporated O ions act as accept dopants, which increases the electronic conductivity.

## Electronic defects:

- What is the difference between electronic and ionic defects?
- A brief review on basic semiconductor physics: how to write the charge neutrality equation by using Kröger-Vink notations?

## Brouwer diagrams:

- What question are we trying to solve by using the Brouwer diagram?
- What is the key assumption of Brouwer diagram (so-called Brouwer condition)?
- How does a typical Brouwer diagram for metal oxides look like? How can it be used to predict/understand the conductivity of materials?

**Goal of this lecture:** you should be able to answer the questions above now (hopefully) : )



# End of Lecture 2

## Solid State Ionics Fall 2023

西湖大學

固态离子学课题组网站



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